

Fluorene-Based Rib Waveguides with Optimized Geometry for Long-Term Amplified Spontaneous Emission Stability

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ABSTRACT: Amongst the different optoelectronic applications of conjugated polymers, waveguide amplifiers and optically-pumped lasers are those requiring larger photochemical stability, owing to the large irradiation conditions under operation. In this context, suitable waveguide optimization enabling the reduction of amplified spontaneous emission (ASE) threshold values appears as important as synthetic chemistry protocols to promote polymer robustness against photo-oxidation. In this work we develop rib waveguides with different geometries based on four different fluorene-based compounds and assess the influence of rib confinement on ASE properties. We observe ASE threshold values as low as $8.9 \cdot 10^{-4} \text{ mJcm}^{-2}$, being among the lowest ASE threshold values reported so far on blue emitting polymer / oligomer waveguides. We demonstrate that the increase in ASE efficiency on some of these rib waveguides leads to a five-fold increase in operation lifetime respect to spincoated slab waveguides, thus confirming the impact of waveguide geometry on ASE operation stability.

KEYWORDS: organic lasers, conjugated polymers, fluorescence, photolithography, oligofluorenes

INTRODUCTION

Conjugated polymers are materials which exhibit outstanding optical gain properties.[1-3] Processed as thin films they behave as slab waveguides, their large stimulated emission cross-section values and waveguiding properties enabling amplified spontaneous emission at very low pump thresholds, compared with organic dyes.[4-5] These assets

together with their semiconducting charge transport behavior open up prospects for the future development of polymer-based electrically pumped lasers.[6-7] Among conjugated polymers, poly(9,9-di-octyl-fluorene) (PFO) is one of the most representative optical amplifying polymers.[8-10] Its notable properties as optical gain

medium arise from a combination of efficient solid state blue emission[11] (c.a. 40%) and excellent thin film optical quality which

manifests into reduced scattering-induced losses and large waveguide performance.[12]

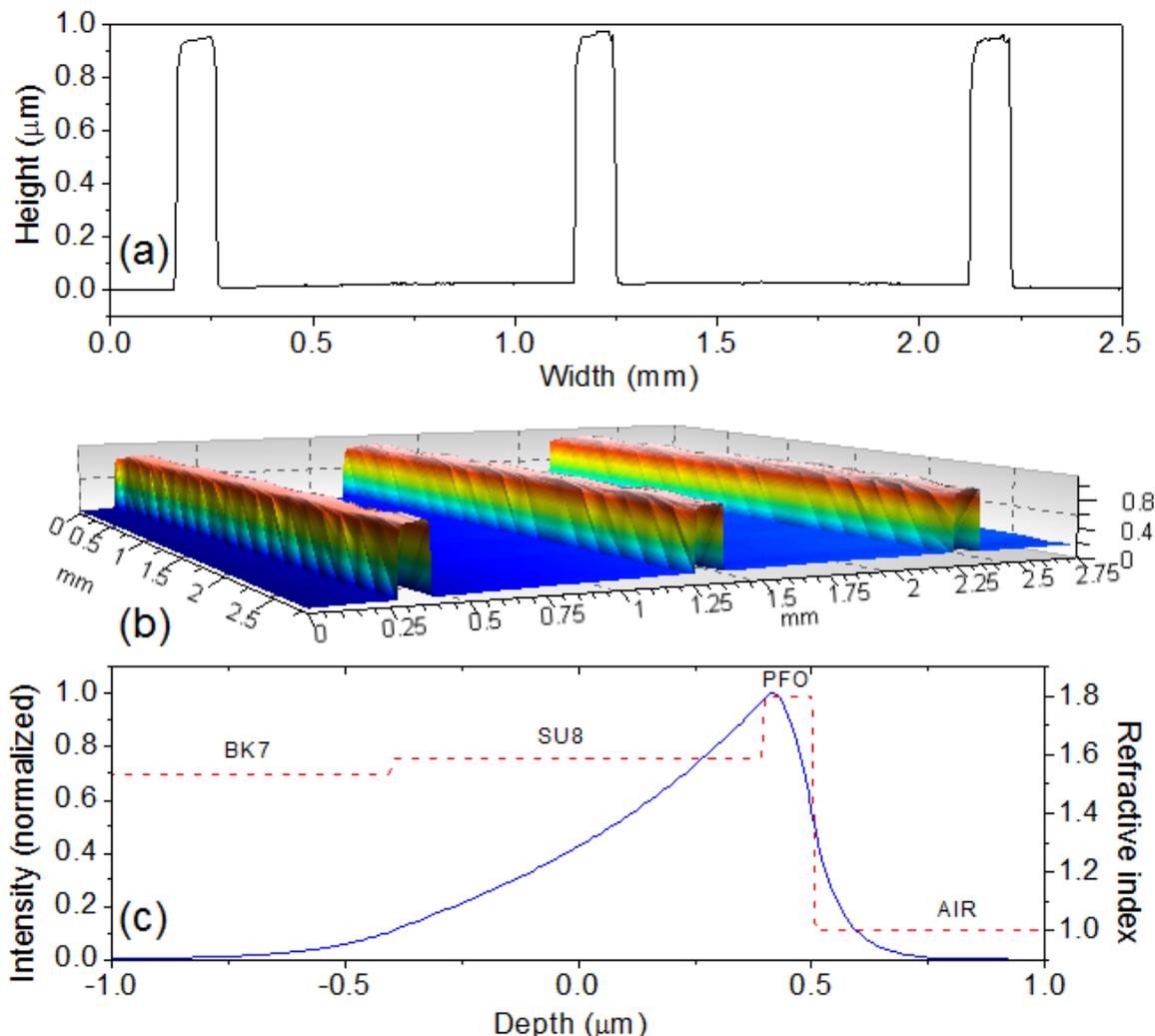


FIGURE 1. (a) Profile of the 100 μm waveguide. (b) 3D reconstruction of the waveguide, (c) in depth intensity distribution of the fundamental propagating mode.

Hitherto, its operation lifetime is compromised by its low emission stability associated to inter-chain aggregation-induced energy transfer to fluorenone defects formed upon photo-oxidation,[13] a behavior which is particularly pronounced under intense UV irradiation. This problem is partially circumvented by reducing inter-chain aggregation, a solution which has been addressed with supramolecular chemical strategies, such as backbone encapsulation with cyclodextrin rings,[14-15] or strategies involving the design of side-chain substituents which

interfere with backbone π - π stacking.[16-17] It has been shown in oligofluorenes that the replacement of alkyl- side-chains by di-aryl groups has led to strong changes in molecular packing inferred from the optical spectra.[18] In line with this second strategy, some of us have already reported on H-shape fluorene-based oligomers bearing one terfluorene arm with flexible alkyl chains and a carbazole-bridged rigid and highly non planar diarylfluorene arm.[19] The large steric hindrance induced by the diaryl-fluorene substitution was manifested

on enhanced emission stability and absence of fluorenone emission upon deliberate aging. Additionally, enhanced thermodynamical stability was observed, as manifested by large glassy morphologies associated ($T_g \sim 250$ °C) and unaltered ASE behavior upon air annealing at temperatures as high as 200 °C. These results suggest that H-shape oligofluorenes are promising compounds for applications involving optical amplification. In this work we develop rib waveguides based on these compounds as well as other three fluorene-based polymers with different side-chain substituents. Optical confinement induced by the rib leads to lowering of ASE thresholds by one order of magnitude respect to spincoated films. The width of the rib is found to have a strong influence on ASE threshold. Our results highlight the impact of chemical structure and waveguide geometry on long term stability of fluorene-based waveguides.

EXPERIMENTAL

Rib waveguides were fabricated by spincoating an epoxy-based negative SU8 photoresist (Gersteltec GmbH), previously heated at 65 °C, at 4500 rpm on top of a 10 nm SiO₂ coated glass substrate. For this purpose, glass substrates were previously cleaned in several ultrasonic baths and dried in a hot stage, before being coated with SiO₂ by electro gun

evaporation method at a fixed deposition rate of 1 Ås⁻¹. The SiO₂ coating was deposited in order to improve the adhesion of SU8 to the substrate. The resulting SU8 layer (c.a. 800 nm thick) was then subsequently baked in two stages: 65°C for 5 minutes and 90°C for 10 minutes. Rib geometries were developed using the standard photolithography process upon UV exposition of SU8 through a shadow mask (365 nm, 100 mWcm⁻² for 25 s) followed by subsequent baking (same parameters as before) and rinsing in developer and methanol. Accordingly, ribs of 20, 50, 100, 200 and 400 μm width, 800 nm height and 8 mm length were developed, (see Figure 1).

Toluene solutions of conjugated polymers were spin-coated at 2500 rpm on top of SU8 ribs. The polymer concentration was adjusted for each material in order to yield approximately 100 nm thickness. Profilometry performed on coated waveguides indicated similar homogeneity and topography than uncoated ones, (not shown). The chemical structures of polyfluorene derivatives employed are shown in Figure 2(a). Poly(9,9-dioctylfluorene) (PFO) was purchased from American Dye Source whereas PODPF, PCzODPF, and Ph-Cz-FOH were synthesized elsewhere.[19-20] The set-up used for optical characterization of the waveguides comprised a Nd:YAG laser (355 nm) with 300 ps temporal width, and 23 Hz repetition rate as excitation

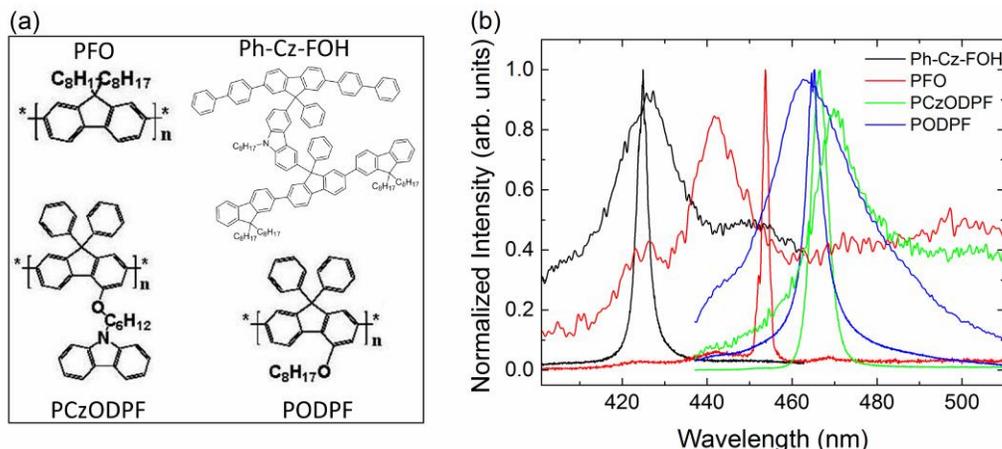


FIGURE 2. (a) Chemical structures of the conjugated polymers used in this study. (b) Corresponding ASE and PL spectra of conjugated polymers deposited on SU8 ribs.

source. The laser beam was focused with a 100 mm focal length cylindrical lens to achieve a narrow stripe beam, (3 mm long and 40 μm width). The length of the beam was adjusted using a tunable slit. Photoluminescence guided along the rib was collected at the edge and focused in free-space configuration on a SP-2558 Princeton Instruments (Acton Research) spectrometer equipped with a liquid N_2 cooled deep-depleted CCD. Excitation fluence was modified with a set of neutral density filters. Losses were calculated upon moving the excitation stripe along the rib and detecting the Amplified Spontaneous Emission (ASE) spectrum for each position. Threshold values were instead calculated from the dependencies of FWHM and output intensity with fluence, being the final result and average between both values. Final losses and threshold values were the result of averaging among 3-4 analogous waveguides (same width and same conjugated compound). Measurements were carried out in air at room temperature. The topographies of waveguides were visualized with an Alpha Step 120 D Tencor profilometer, able to realize 1D and 2D measurements as well as 3D reconstructions.

RESULTS AND DISCUSSION

The emission spectra of conjugated polymers are shown in Figure 2(b). PFO is a widely characterized blue emitting polymer whereas Ph-Cz-FOH, PODPF and PCzODPF are novel polyfluorene compounds with emission in the deep blue spectral region. These four compounds exhibit ASE on rib waveguides upon pumping with high fluences, being their ASE peaks located at 454, 425, 465 and 467 nm

respectively. The spectral location of the ASE peaks is in correspondence with the 0-2 photoluminescence (PL) vibronic peak for the four compounds, thus suggesting that they behave as a four-level amplifying medium, in agreement with previous findings in conjugated polymer spincoated films.[21] **The 300 ps pulse duration employed as excitation source is close to the PL lifetime values of most the compounds employed in this work.[] [] [] Thus, ASE is likely to arise from a population inversion build-up involving generation and recombination rates, rather than being of impulsive nature.** Figure 3 displays the Output.vs.Input characteristics of four materials on 200 μm width ribs. In all cases a sudden drop in the FWHM of emission accompanied by a dramatic emission enhancement with increasing fluence is observed. Note that a collapse of FWHM in the 10^{-4} fluence range is observed for Ph-Cz-FOH in contrast with 10^{-2} – 10^{-1} for the other three compounds, thus anticipating superior amplifying performance of the former compound. Figure 4 displays the dependence of the ASE threshold values on waveguide width for the four polymers. In order to understand the effect of rib confinement on ASE threshold we also performed control measurements in conjugated polymers spincoated on top of SU8 films. In all cases, ASE values in ribs were considerably lower. If we focus on PFO for instance we observed almost a nine-fold decrease in ASE threshold when coated on 100 μm ribs (from 0.09 mJcm^{-2} in SU8 film to 0.013 mJcm^{-2} on ribs).

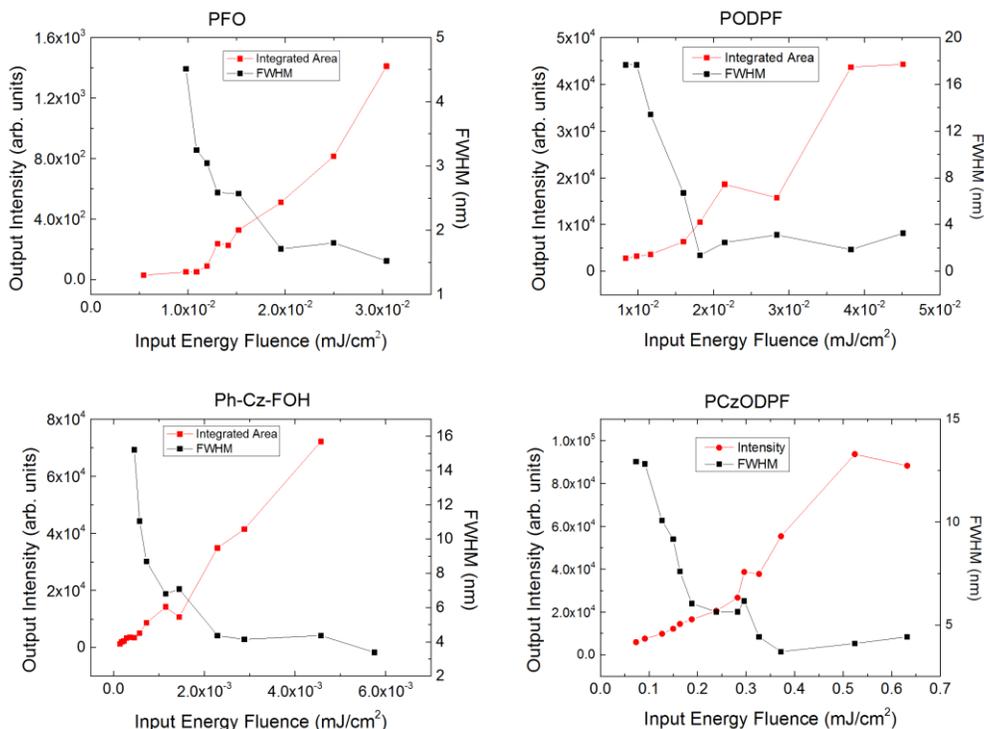


FIGURE 3. Examples of the Input - Output emission characteristics of conjugated polymers on 200 μm wide ribs displaying intensity and FWHM as function of fluence.

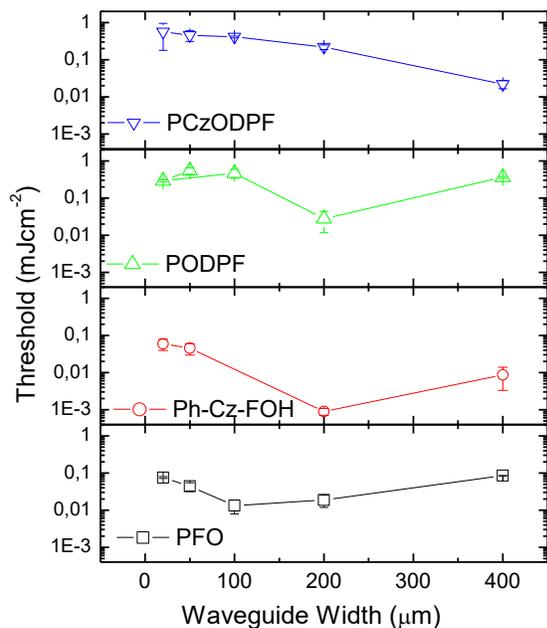


FIGURE 4. ASE threshold values for different polymers and different rib widths. The error

bars were obtained from the standard deviation resulting from values measured on four different waveguides.

On account of the highly asymmetric distribution of the fundamental mode in ribs displayed in Figure 1(c), we infer that ASE efficiency enhancement is associated to in-plane confinement of the emission induced by the rib geometry, rather than to a displacement of the TE₀ mode from the substrate to the polymer film. This second strategy has been previously adopted by other authors in symmetric waveguides.[11] Concerning the width dependence of ASE, at first glance the ASE threshold values diminish upon decreasing the widths down to 100 - 200 μm in all cases except for PCzDOPF. This result was reproduced over time and we attribute it to

the low solubility of this latter compound in toluene and the interplay of aggregate-induced scattering leading to poor amplifying properties of the same. As the rib width gets narrower (50 and 20 μm) the limited resolution at the edges induced by the contact with the shadow mask becomes more important, leading to an overall decrease in planarity of the rib surface. Under these circumstances, larger inhomogeneity of conjugated polymer films deposited on top is expected which can explain the increase in threshold on waveguides with the lowest widths. Comparing the four materials, Ph-Cz-FOH displays the lower thresholds, with a lowest value of $8.9 \cdot 10^{-4} \text{ mJcm}^{-2}$ found on 200 μm width ribs. This value is more than one order of magnitude lower than that reported by Qian et al. in spin-coated films on top of quartz substrates, ($8.3 \cdot 10^{-3} \text{ mJcm}^{-2}$).^[19] The outstanding optical amplification properties of Ph-Cz-FOH likely arise from the bulky and rigid nature of its carbazole-diaryluorene side-chain substituents, leading to large steric hindrance and reduced inter-chain interaction. Accordingly, detrimental effects such as aggregation [16] as well as absorption from excited states stabilized between two polymer chains [22] would be partially hindered. Figure 5(a) depicts the dependence of ASE intensity as a function of distance between the excitation stripe and the waveguide edge on a Ph-Cz-FOH 200 μm rib. From the exponential dependence we extract a loss coefficient value of 15 cm^{-1} . Comparison between loss coefficients found in waveguides with different widths reflects again slightly upper values for 50 and 20 μm ribs (26 and 31 cm^{-1} respectively), in line with the upper threshold values already shown for same widths. The use of rib waveguides with very low threshold associated allows their ASE operation under low pumping energy regimes.

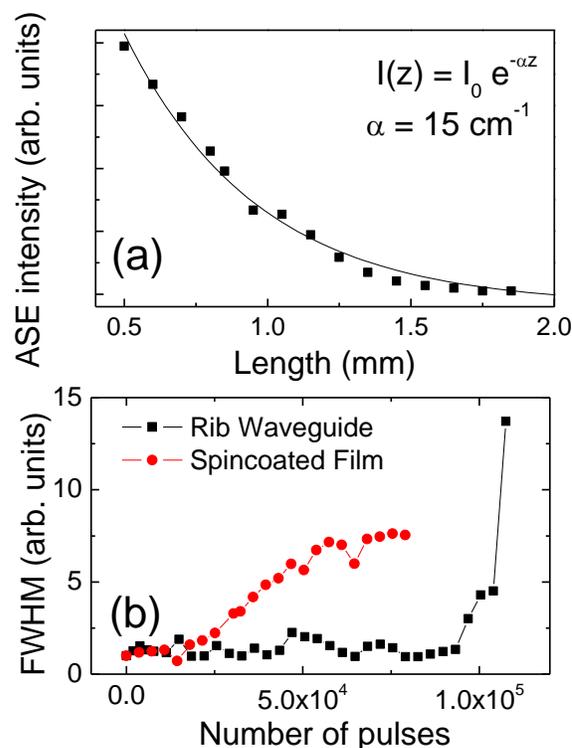


FIGURE 5. (a) Dependence of ASE on distance between stripe and waveguide edge in a 200 μm Ph-Cz-FOH rib depicting an exponential type behavior. (b) Evolution of FWHM of PFO on a 100 μm rib upon continuous pulse irradiation.

This is of particular importance since one of the main drawbacks of PFO based optical amplifiers is their poor environmental stability. In order to illustrate this, we compared the ASE degradation of PFO on SU8 rib and film respectively upon continuous photoexcitation. For each sample we tune the pump fluence just above their ASE threshold, (1.09 and 0.26 mJcm^{-2} for film and waveguide rib respectively), and photoexcite continuously for several minutes recording one spectrum every 1-2 minutes. Figure 5(b) shows the evolution of FWHM upon continuous pulse excitation. For comparative purposes, normalization to the initial values is shown. The FWHM of a PFO film starts to evolve already after photoexcitation with $2 \cdot 10^4$ pulses. **PFO photo-oxidation leads to formation**

of keto defects which act as antennae, funneling PFO singlet excitons and causing a change in the amplifying exciton density. Furthermore, it has been suggested that additional pitfalls arise from losses of waveguiding properties associated to keto defect re-absorption.[Anni] As photo-oxidation becomes more severe the ratio of PL to ASE increases leading to a rise in FWHM. Conversely, the FWHM of PFO rib waveguide remains nearly unchanged upon photoexcitation with 9×10^4 pulses, implying that a much lower photodamage is produced owing to the much lower density of incident photons required for ASE. These observations indicate that the operation lifetime of conjugated polymer amplifiers can be enhanced not only by promoting polymer robustness against photo-oxidation by synthetic chemistry routes, but also upon appropriate waveguide optimization.

CONCLUSIONS

Fluorene-based rib waveguides with very low ASE thresholds were developed. Our work suggests that the rib confinement enhances ASE efficiency up to a certain width in the 100-200 μm range. Waveguiding losses on sub 100 μm width ribs are likely influenced by the mask contact and shadow effects induced during the photolithography process. We demonstrate that the low ASE threshold values of rib waveguides implies a significant enhancement in long term ASE operation under continuous pulse excitation.

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GRAPHICAL ABSTRACT

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Fluorene-Based Rib Waveguides with Optimized Geometry for Long-Term Amplified Spontaneous Emission Stability

Long-term operation of blue emitting polyfluorene-based optical amplifiers is compromised by their relatively low photostability. In this work we develop rib waveguides based on different fluorene-based compounds with different side-chain substituents. We observe a nine-fold reduction in energy threshold values for amplified spontaneous emission respect to thin films which is associated to confinement and enhanced waveguiding properties. We demonstrate how appropriate rib width optimization allows for a significant enhancement in amplified spontaneous emission stability during continuous pulsed

excitation.

