1	Engineered/designer hierarchical porous carbon materials for organic
2	pollutant removal from water and wastewater: A critical review
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## 22 Graphical abstract



Organic contaminant removal in water

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## 25 Highlights

- 26 Hierarchical porous carbon materials (HPCs) are important in different disciplines.
- 27 HPCs can be synthesized from diverse carbon materials by various methods.
- 28 HPCs are highly effective in organic contaminant removal in waters.
- 29 The efficacy of organic contaminant removal can be enhanced by modification of HPCs.

#### 30 Abstract

Hierarchical porous carbon (HPC) materials have found advanced applications in energy 31 storage, adsorption, and catalysis in recent years. Since HPCs can be synthesized from a vast 32 range of inexpensive carbon precursors including waste biomasses, and contain unique 33 34 structural features, such as nano-scale dimension, high porosity, high surface area, and tunable pore surfaces, these materials hold an immense potential for removing contaminants 35 from water. However, currently this area is severely under-explored. In this review, we 36 37 discussed the recent advances of synthesis, modification, and application of HPCs for contaminated water cleanup, especially focusing on organic pollutants. Findings suggest that 38 HPCs can be synthesized using multiple methods (e.g., dual templating, hard-soft templating, 39 soft-soft templating, bio-templating, a combination of activation and templating) including 40 the advanced nanopore lithography technique. Owing to their intrinsic hydrophobic nature 41 and unique interconnected porous structure, HPCs demonstrate high affinity to hydrophobic 42 organic contaminants, which can be enhanced many folds by target-specific chemical 43 activation such as alkali and/or hydrothermal treatments. Successful high-performance 44 removal of water contaminants by pristine and modified HPCs include plastic-derived (e.g., 45 bisphenol A), pharmaceutical (e.g., antibiotics), dye (e.g., methylene blue) and pesticide 46 micro-pollutants. Besides, the easily tunable features of HPCs make them a promising 47 commercial filtration/membrane material for household and large-scale wastewater treatment 48 49 applications. Therefore, future research is warranted to find optimal and effective HPC synthesis and modification methods for further improving their ability to remove aqueous 50 organic contaminants as a low-cost and energy-inexpensive remediation technology. 51

- 53 Keywords: Carbon material; Contamination; Engineered carbon; Electrode material; Waste
- 54 water treatment.

#### 56 **1. Introduction**

Hierarchical porous carbon (HPC) materials have drawn increasing attention over the past twenty years [1–3]. HPCs with hierarchical pores can be synthesized from carbon precursors via chemical activation and templating with different materials. Scientists around the world are formulating, testing, and applying HPCs, derived from many different carbon precursors, such as kraft lignin [4], polyacrylonitrile [5], cotton stalk [6], chitosan [7], and polystyrene [8] in different applications.

HPCs have many tailored structural features, such as nanostructures, high porosity, high
surface area, unique pore surface chemistry, and high electrical conductivity [9–11].
Consequently, HPCs can be used in different real-world applications, such as, electrode
materials, electro-catalysts, energy storage, chromatography, adsorption, catalyst, sensing and
nanoreactors [2,12–16].

68 Water pollution by organic contaminants has been repeatedly reported globally [17]. For instance, discharge of pharmaceuticals and personal care products from untreated or poorly 69 treated wastewaters around the world has been identified as one of the emerging water 70 71 polluting problems [18–24]. Subsequently, the scarcity of clean water and high demand of water consumption in the world need effective remediation technologies and pollution control 72 measures [25–28]. With interconnected porous structures, tunable pore size and structures, 73 74 excellent flow-through permeability, high specific surface area (SSA), HPCs are one of the 75 best materials that can effectively be used in water remediation [2,29].

There are 3892 articles published in SCOPUS database (quoted on 15 January 2020) on HPCs when searched in article title, abstract and keywords. Among them, 1.7% are review articles, and >70% of the review articles are focused on advanced applications of HPCs, such

as super capacitor, energy storage and catalysis [30–32]. However, according to the SCOPUS 79 database and to the best of authors' knowledge, no review till date concentrated on the 80 removal of aqueous organic contaminants by HPCs. Applications of HPCs in water related 81 studies have increased only since 2010 (Fig. 1). Hence, understanding the synthesis 82 83 procedures of HPCs, their properties, and possible modifications are critical to promote their applications in water and wastewater treatment. Therefore, this work will provide the first 84 critical review on the designs and applications of HPCs for the removal of organic 85 contaminants from polluted waters. 86

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#### 88 2. Methods of synthesizing HPCs

#### 89 2.1. Dual templating method

90 Templating is the most commonly and effectively used method to design and control pore size distributions of HPCs (Table 1). Knox et al. [33] pioneered the method to synthesize 91 porous carbon materials by the templating method, and since then, it has been gradually 92 developed. Templating method refers to the process of first depositing related materials into 93 94 the holes or surface of the template by physical or chemical ways, and then removing the template to obtain nanomaterials with standard morphology and size of the template. The 95 process proceeds in the area of effective control, and thus, it is easy to tail and vary the 96 97 structural parameters of HPCs. Templating methods mainly include hard and soft template 98 methods (Fig. 2).

Hard templates are generally rigid forms, held together by stable inorganic solids [1]. The
inorganic solids can be different types of silica, such as silica monolith [34], silica spheres
[35], silica colloidal crystals [36,37], silica opal [38,39], etc. In addition, CaCO<sub>3</sub> [40] and

Na<sub>2</sub>CO<sub>3</sub> [41] are also commonly used as porogen of hard template. Soft templates are
typically organic polymers that can be thermally decomposed and removed [1,42,43]. These
thermally decomposed polymers can be polystyrene (PS) [44,45], polymethyl methacrylate
(PMMA) [46], polyurethane (PU) and surfactants [47].

106 Considering that HPCs are composed of different levels of pores, the dual templating 107 method is commonly used to synthesize HPCs of macro-mesoporous, meso-microporous, 108 macro-meso-microporous structures. The carbon source is guided by the double space 109 limiting action of two pore-forming agents to achieve the purpose of the graded structure.

- 110
- 111 2.2. Hard-soft templating method

When combining hard and soft templates, phenolic resin is often used as the carbon 112 precursor to allow the organic-organic self-assembly with triblock copolymers in the 113 interspaces between inorganic solids [36]. Silica is still one of the most popular hard template 114 to synthesize mesoporous or macroporous carbons with diameters of about 30-50 nm [48] or 115 200-500 nm [35,36] because it is easy to control the ordered structure. In addition, hard silica 116 117 template can restrict the shrinkage of the framework during the thermosetting and carbonization process [36,49]. For example, Yonghui et al. [36] assembled purified and 118 uniform silica microspheres into ordered colloidal crystal templates, then heated at 100 °C for 119 120 24 h to ensure the structural hierarchy and stability of templates. The obtained porous carbons 121 had a highly ordered face-centered cubic macrostructure with tunable pore sizes of 230-430 nm and interconnected windows with a size of 30-65 nm [36]. Besides, Li et al. [38] explored 122 123 the effect of silica to the restrain shrinkage of mesoporous polymers. The authors indicated

that the hydroxyl group interactions among the resin polymer and the surface of SiO<sub>2</sub> spheres
and 3D structure of SiO<sub>2</sub> template played important roles in restrain shrinkage [38].

Furthermore, by using two diameter-sized colloidal crystal templates, 3D interconnected 126 ordered macroporous carbon with uniform mesoporous walls are fabricated. The process is 127 128 realized through physical mixing of polystyrene (PS) colloidal crystals and silica particles, which simplifies the template synthesis route. Chai et al. [44] prepared HPCs using nano-129 casting method. First, PS microspheres and small silica particles were mixed. During the 130 drying process of the mixture, PS microspheres self-assembled into an ordered array, while 131 SiO<sub>2</sub> particles with small size were closely arranged in the cracks of PS array, forming a 132 mixed dual-template. The carbonization and removal of PS template created templated 133 aggregate of the small silica particles, which was then impregnated with the carbon precursor 134 (divinylbenzene), and finally HPCs were obtained through the carbonization of the carbon 135 136 precursor and dissolution of silica (Chai et al., 2004).

In a different study, Woo et al. [45] used PS colloidal crystals as a pore-making agent as well as a carbon precursor. After heat treatment at 300 °C, the melted PS was first penetrated into the space between the colloidal silica. The penetrated PS was then carbonized with heat treatment to provide a very thin carbon layer on the colloidal silica, and the microporous structure corresponding to the PS particle size was formed simultaneously. This greatly simplified the process of the impregnation of carbon precursor [45].

143 Zhang et al. [50] developed a one-pot method to avoid pre-synthesis of the template and 144 additional infiltration. The interconnected macropores and mesopores were synthesized by in-145 situ self-assembly of colloidal polymer and SiO<sub>2</sub> particles with sucrose as the carbon source. 146 This procedure is simple and easy to operate. In addition, intermediate composite films are 147 compounded rather than powder due to soft polymer spheres (Tg = 21 °C) compared with 148 hard PS spheres. It is the surrounding nano-silica particles and sucrose crystallites that 149 restrain the deformation of these soft polymer spheres [50].

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### 2.3. Soft-soft templating method

Unlike the hard templates, the soft templates can be removed in the process of 152 carbonization, and the use of harmful reagents for etching the templates can be reduced to 153 154 some extent. For instance, two soft templates of poly(methyl methacrylate) (PMMA) colloidal crystals and triblock copolymer can be used to prepare HPCs [51]. In this process, 155 PMMA plays the same role as silica, while the PMMA template can be decomposed 156 completely during carbonization when the temperature reaches 900 °C to produce macropores 157 [51]. Lower concentration of amphiphilic triblock copolymer can produce surface-centered 158 cubic mesoporous structures, and a higher concentration can produce 2-D mesoporous 159 structures [51]. 160

Similarly, Xue et al. [52] used organic polyurethane (PU) foam scaffold as a sacrificial macroporous template on which solvent evaporation induced the self-assembly process of phenol/formaldehyde resol and triblock copolymer. Their hierarchical porous framework constructed by macropores was cable-like struts, and the carbon material exhibited relatively disordered macropores with diameters of about 100-450 µm [52].

- 166
- 167 2.4. Bio-templating method

Unlike previously discussed methods that use artificial carbon skeletons, bio-templating isa simple, sustainable, environment-friendly and suitable method for mass production of HPCs.

170 Biomass of natural inorganic/organic composites often contains nanostructures. For example, Huang and Doong [53] prepared HPCs by employing natural sugarcane bagasse. Surface 171 coating (i.e., triblock copolymer F127 and phenol-formaldehyde resin) and solvent 172 evaporation-induced self-assembly were employed. After carbonization at 1000 °C, the 173 174 carbon materials formed from sugarcane bagasse maintained a stable skeleton structure, and interconnected macropores based on the natural texture and 2-D hexagonal ordered 175 mesopores were clearly observed. The authors suggested that this might be due to the 176 177 hydroxyl functional groups of the bagasse which interacted with phenolic resin. The drawback was that the specific surface area (SSA) was not high enough, only 544 m<sup>2</sup> g<sup>-1</sup>, and 178 179 the microporosity was up to 66-67% [53].

180 Chen et al. [54] selected fish scales as a raw material to prepare HPCs, which was activated by KOH to produce micropores with SSA as high as 2273 m<sup>2</sup> g<sup>-1</sup>. First, the fish scales have an 181 overlapping plywood structure of stratified lamellae, which can be preserved to maintain 182 carbon skeleton after carbonization and activation. Second, the fish scales are composed of 183 organic components (mainly collagen fibers) and inorganic components (calcium-deficient 184 185 hydroxyapatite), which form macropores and mesopores, respectively. Hydroxyapatite disperses well in organic components helping to maintain the stability of the carbon skeleton 186 [54]. 187

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#### 189 **2.5.** Combination of activation and templating methods

190 In general, the relatively ordered HPCs, containing a large number of micropores, is 191 mainly prepared by the template method following post-activation step to produce micropore on the walls of mesopore or macropore [1]. Carbon dioxide, water vapor and KOH arecommonly used as the physiochemical activation agents [55,56].

Xia et al. [57] explored the effects of CO<sub>2</sub> activation on the pore structure of highly ordered 194 mesoporous carbon CMK-1 and CMK-3. The activated carbons showed variable porous 195 196 textures and remarkable enhancements in the volume and SSA of both mesopores and micropores. The authors indicated that this was probably due to the combined effects of the 197 closed pores, new narrow pores and extensive pre-existent pores. The great enhancement was 198 199 accompanied by the expansion of the ordered porous structures. With the extension of 200 activation time, the long-range ordering was lost; however, the interconnection of pores in 201 some parts still remained intact [57].

Compared with silica, commercial nano-CaCO<sub>3</sub> microspheres have been used as a 202 relatively environmentally-friendly nano-template to synthesize HPCs, and attracted wide 203 204 attention. Macro-mesoporous and micro-mesoporous carbon can be synthesized by choosing 205 different microsphere sizes of nano-CaCO<sub>3</sub>. Yang et al. [58] reported that nano-CaCO<sub>3</sub> could 206 be used not only as a template, but also as an activating agent by producing CO<sub>2</sub> from the 207 carbon precursor to produce micro- and mesopores [58]. Similarly, nano-CaCO<sub>3</sub> was used as a dual template to synthesize macro-mesoporous carbon nanofiber, and CaCO<sub>3</sub> could be 208 dissolved by acid to form macropores after the carbonization [38]. 209

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#### 2.6. New methods of synthesizing HPCs

In order to overcome the drawbacks of the above-mentioned methods, many novel strategies and raw materials have been introduced to synthesize HPCs. Martín-Jimeno et al. [59] successfully synthesized a 3-4 nm uniform mesoporous carbon material by using metal organic framework (MOF) as a template. They coated the MOF with graphene oxide nanosheets. At high temperatures (i.e., >800 °C), the incipient porosities of MOF served as the entry point for activation and created uniform mesoporosity on the graphene oxide nanosheets. This process can be termed as "nanopore lithography". The hierarchically micromesoporous structure relies on a strictly controlled amount of KOH (activating agent), and excessive activation with high amounts of KOH can cause a collapse of the carbon skeleton [59].

Pomelo peel, an environment-friendly biomass, was used to synthesize hierarchical mesomicroporous carbon [11]. The dual-activating agent of  $NH_4H_2PO_4$  and  $KHCO_3$  suggestively increased the SSA (2726 m<sup>2</sup> g<sup>-1</sup>) and percentage of mesopores (52%) of HPCs. At an elevated temperature (e.g., 800 °C), the  $NH_3$ , CO and CO<sub>2</sub> gases generated from the activating agents were released to create pores in the carbon precursor. The difference between the two activating agents was that they promoted the expansion of the micropores and the conversion of micropore to mesopore, respectively [11].

229 Siyasukh et al. [60] reported that HPC monolith could be synthesized using high intensity 230 ultrasonic wave to generate macropores, and Ca(NO<sub>3</sub>)<sub>2</sub> impregnation and CO<sub>2</sub> activation could generate mesopores on the wall of macroporous monolith (Siyasukh et al., 2008). Zou 231 et al. [61] reported that HPC could be compounded through the reactions of linear 232 233 polystyrene resin, carbon tetrachloride and anhydrous aluminum chloride. It was the -CO-234 group that connected the polystyrene chains and made up the whole carbon structure. The mesopore and macropore were caused by the gap of different sizes formed by the non-235 236 uniform arrangement of polystyrene nanoparticles [61].

#### 238 **3. HPC properties**

The unique properties of HPCs include uniform macropores, interconnected meso- and/or 239 micropores, and an overall well-defined pore system, which allow them to have an excellent 240 mass transfer performance associated with the larger pores, and abundant adsorption sites 241 242 associated with the smaller pores. Furthermore, high SSA and pore volume in general, are also crucial to promote mass transfer and adsorption processes. HPCs containing multiple 243 levels of pores have higher SSA and pore volume than the carbon materials with single-size 244 pores due to the fact that the space of HPCs is fully utilized. Li et al. [49] indicated that 245 macro-mesoporous carbon (Pluronic P123 and silica as pore-making agent) has a higher SSA 246  $(803 \text{ m}^2 \text{ g}^{-1})$  and pore volume  $(0.86 \text{ m}^3 \text{ g}^{-1})$  than that of pure mesoporous carbon  $(350 \text{ m}^2 \text{ g}^{-1})$ 247 and 0.35 m<sup>3</sup> g<sup>-1</sup>, respectively). Meanwhile, Zhang et al. [37] reported that macro-mesoporous 248 carbon also had a higher SSA (1290 m<sup>2</sup> g<sup>-1</sup>) and pore volume (1.35 m<sup>3</sup> g<sup>-1</sup>) than that of pure 249 macroporous carbon (473 m<sup>2</sup> g<sup>-1</sup> and 0.82 m<sup>3</sup> g<sup>-1</sup>, respectively) in the absence of Poloxamer 250 251 407, used as a mesoporous template.

Through self-thermal polymerization of phenolic resin, a highly crosslinked and stable 252 polymer can be formed, leading to the monolithic feature of cm in size (Li et al., 2016; Meng 253 et al., 2005). In the absence of surfactant to build a stable structure, the synthesized thin films 254 or powders might have macroscopic morphology. However, in practical water treatment 255 256 process, these hierarchically porous carbon monoliths (HPCM) would show great advantages 257 over the thin films or powders. Since the use of monolith might effectively reduce the highpressure drop, this would avoid adsorbent loss and secondary pollution during an adsorption 258 operation. Moreover, the monolith can be used in continuous flow microreactor instead of 259 conventional packed-bed reactors [62,63]. The HPCM showed the combined properties of 260

both mesoporous HPC (i.e., high specific surface area, uniform pore size, large pore volume,
interconnected pore channels for adsorption, good chemical inertness and stability), and
macroporous HPC (i.e., mass transport, fast accessibility of the bulky reagents/adsorbates and
high storage capacity) [52,63–65].

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#### 266 4. Engineered HPCs

Additional micropores and mesopores can be developed through HPC modification, and it 267 enhances the adsorption of organic contaminants [66]. Some strategies as explained in the 268 previous sections have been recommended to synthesize HPCs, and a combination of 269 template carbonization and chemical activation might improve the porous structure 270 development in HPCs (Figure 3). Generally, it includes two steps. Firstly, carbon materials 271 incorporated with hard templates (e.g., silica, nano-CaCO<sub>3</sub>, nano-MgO, nano-Fe<sub>2</sub>O<sub>3</sub>, and 272 nano-ZnO) are carbonized at high temperature under an inert atmosphere. This process 273 creates meso- and macropores. Secondly, chemical activation (i.e., KOH or NaOH) is used to 274 develop the micropores [67]. The SSA of HPCs following template carbonization and 275 chemical activation can reach beyond 2000 m<sup>2</sup> g<sup>-1</sup>, and generally comprises micropores of >2276 nm size [68]. 277

Yu et al. [67] studied the production of HPCs activated by KOH, having different concentrations (i.e., 1, 2 and 3 M). Enteromorpha, a sea weed, was used as the precursor for the synthesis of HPCs. Contrasting to the conventional activation of carbonized materials, carbonization and chemical activation were simultaneously carried out at 800 °C. The scanning electron microscopy (SEM) images showed the development of highly porous HPCs with interconnected macropores (i.e., 200-1000 nm) with an increase of KOH concentration. Moreover, X-ray photoelectron spectroscopy (XPS) analysis indicated the
 considerably high heteroatom (i.e., N and O) doping in HPCs.

Hu et al. [68] developed a new method by coupling the in situ template with a NaOH 286 activation to enhance the porous structure of HPCs. They used lotus seed shell as the biomass 287 288 and sodium phytate as the hard template precursor. During the carbonization, sodium phytate converted to nano-Na<sub>5</sub>P<sub>3</sub>O<sub>10</sub> and reacted with NaOH. During this reaction, nano-Na<sub>2</sub>CO<sub>3</sub> and 289 nano-Na<sub>3</sub>PO<sub>4</sub> particles were generated and homogeneously dispersed in the biomass creating 290 large mesopores and macropores after washing with HCl. Moreover, NaOH created 291 micropores, and ultimately produced a well-developed hollow nest-like structure of HPC. It 292 showed very high SSA of 3188 m<sup>2</sup> g<sup>-1</sup> and total pore volume of 3.2 cm<sup>3</sup> g<sup>-1</sup>. Conventional 293 294 NaOH activation is well known to produce micropores. However, the authors clearly showed that NaOH activation with the addition of sodium phytate could create HPCs with micro-, 295 meso- and macropores. Similarly, Chen et al. [69] carried out carbonization and chemical 296 297 activation by KOH concurrently, and developed HPCs with high micro-, meso- and macropores. They were able to produce N self-doped 3D porous HPCs from waste cottonseed husk 298 299 by a one-step chemical activation.

A hydrochar material produced from eucalyptus sawdust was chemically activated by the mixture of potassium oxalate monohydrate ( $K_2C_2O_4$ ) and powdered melamine ( $C_3H_6N_6$ ) to enhance the porous structure [70]. Authors reported the presence of randomly distributed pores in HPCs via high-resolution transmission electron microscopic observations. Moreover, the XRD and Raman spectroscopy analyses revealed the presence of amorphous-like structure. Nitrogen adsorption-desorption isotherms results of HPCs showed an enlargement of pore size corresponding to increased melamine/hydrochar ratio, however, HPCs had >70% of micro pore volume from total pore volume. The optimum activation of hydrochar by  $K_2C_2O_4$  and  $C_3H_6N_6$  (i.e.,  $K_2C_2O_4$ /hydrochar = 3.6–6 and  $C_3H_6N_6$ /hydrochar = 2) generated HPC showed 3000 m<sup>2</sup> g<sup>-1</sup> of SSA which was similar to HPC obtained under harsh KOH activation conditions [71,72]. Hence,  $K_2C_2O_4$  and  $C_3H_6N_6$  can be substituted as promising chemicals instead of KOH to improve the porous structure of HPCs. In addition, these two chemicals are less corrosive compared to the KOH, causing less technical constraints [70].

The HPCs produced with synthetic carbon precursors are also modified with different chemicals. For instance, Wu et al. [73] produced HPCs with o-phenylenediamine and modified with ammonium persulfate ( $(NH_4)_2S_2O_8$ ), and potassium ferricyanide ( $K_3$ [Fe(CN)<sub>6</sub>]) to create Fe active sites, and N and S doping in HPCs. Authors did the carbonization and chemical modification simultaneously at 600 °C for 3 h after several sample preparation steps. Scanning transmission electron microscopy (STEM) coupled with energy-dispersive spectroscopy (EDS) clearly visualized the doping of Fe, N and S in HPCs.

Górka and Jaroniec [74] produced different HPCs using polymeric carbon precursors and 320 block copolymer template in acidic conditions with tetraethyl orthosilicate (TEOS) and 321 322 colloidal silica. They developed cylindrical and spherical mesopores having dimensions of 12 nm and 20-50 nm, respectively. The thermal decomposition of the soft template created the 323 cylindrical mesopores, dissolution of colloidal silica resulted the spherical mesopores, and the 324 325 dissolution of TEOS created the fine pores in HPCs. In addition, post activation of HPCs with carbon dioxide and water vapour further increased the fine pores, and the surface area 326 increased up to 2800 m<sup>2</sup> g<sup>-1</sup>. Similarly, Lee et al. [75] synthesized HPCs with high surface 327 area (i.e., 1625–1796 m<sup>2</sup> g<sup>-1</sup>) and porosity from polyacrylonitrile fibers, silica template, and 328

via post activation of KOH. They developed a porous structure with < 2 nm micropores, 2-5</li>
and 10-50 nm mesopores, and > 50 nm macropores.

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#### 332 5. Removal of aqueous organic contaminants by HPCs and associated mechanisms

In recent years, with the advancement of industrial activities, organic contaminants including dyes, pharmaceuticals, pesticides, hydrocarbons and other emerging contaminants are being continuously released into the aquatic environment, and causing the environmental problems, which ultimately affect humans' health. To control and limit the impact of organic contaminants on the environment and human health, the urge for exploring novel adsorbents with excellent adsorption performance to remove the pollutants from contaminated water is growing steadily (Table 2).

Traditionally, activated carbons have played an important role in the adsorption field 340 because of their very high SSA and porosities. Activated carbons, however, suffer from 341 limitations, such as the limited interconnectivity between defective and irregular microporous 342 structures that restricts the contaminant molecule's access to the adsorbent surface. Ji et al. 343 344 [76] found that the pore structure of template-synthesized carbon adsorbent primarily supported the adsorption of antibiotics (i.e., sulfamethoxazole, tetracycline, and tylosin), 345 while the highly disordered and closed pore structure of activated carbons may lead to the 346 size-exclusion effect or slow adsorption kinetics. Therefore, it is necessary to use the 347 348 templating method to synthesize carbon materials to make up the structural weakness of activated carbons and promote the adsorption performance of the carbon materials. 349

350 Nowadays, HPCs possessing uniform macropores and interconnected meso- and/or 351 micropores for adsorption have received considerable research attention. This is mainly

because of the unique properties of HPCs such as fine and well-defined pore systems, 352 excellent performance of mass transport via larger pores, and the high amount of adsorption 353 sites at smaller pores. There are only few reports about the adsorption of organic 354 contaminants by HPCs, and the studied organic contaminants included dyes (methylene blue), 355 antibiotics (sulfamethazin, tetracycline, chloramphenicol), hydrocarbons (phenol, bisphenol 356 A), oil, bilirubin, etc. The molecular size of the studied organic contaminants was diverse 357 including bulky (e.g., oil, bilirubin, dyes) as well as small (e.g., antibiotic, hydrocarbon) 358 359 molecules, which showed the extensive application potentials of HPCs in the field of contaminant adsorption. 360

The HPC monoliths (HPCMs) with three-dimensionally connected macroporous and 361 ordered hexagonal mesoporous structures were developed via an optimized hydrothermal 362 process followed by a nanocasting pathway [63]. Because of strong hydrophobicity (water 363 contact angle at 140  $\pm$ 3 °) and extremely low density (0.017 $\pm$ 0.002 g cm<sup>-3</sup>), HPCMs 364 performed as a superior adsorbent compared with conventional mesoporous carbon CMK-3 365 and traditional activated carbon. The oil adsorption capacity of HPCMs reached to 45 mg g<sup>-1</sup> 366 within a few seconds. The extraordinarily high SSA (1354  $m^2 g^{-1}$ ), macroporous cumulative 367 volume (48.6 cm<sup>3</sup> g<sup>-1</sup>), and ordered mesoporous and 3D connected microporous structures 368 contributed to the high rate of oil adsorption by the material. In addition, bilirubin adsorption 369 on HPCMs took place within 2 h, accounting for 86.8% removal (equilibrium concentration = 370 700 mg  $L^{-1}$ ). The bilirubin adsorption capacity of HPCMs was 613 mg  $g^{-1}$ , which was 371 remarkably higher than single mesoporous or microporous carbon materials [63]. 372

Liu et al. (2012) developed a macro-meso-micro HPC using two types of diatomites (i.e., Dt(JL) and Dt(SD) as the template and catalyst) for methylene blue (MB) adsorption. The 375 structure of the HPC was dependent on the original form of the template, and the SSA of the 376 templated carbon materials was 270 m<sup>2</sup> g<sup>-1</sup> and 335 m<sup>2</sup> g<sup>-1</sup>, respectively. Within 20 min, the 377 adsorption of MB on both the diatomite-templated carbon materials reached the equilibrium 378 state, showing a fast adsorption process. The maximum monolayer adsorption capacity ( $Q_m$ ) 379 values of the template materials were 333 mg g<sup>-1</sup> and 250 mg g<sup>-1</sup>, respectively, indicating a 380 higher adsorption capacity than commercial activated carbon (CAC) [77].

Dai et al. [78] reported adsorptive removal of sulfamethazine by lignin-based HPCs with 3-381 D interconnected macroporous and meso-/microporous structures. The meso-/micropores 382 produced by the activation of KOH was less than 4 nm in diameter, and the macropore 383 diameter was about 200 nm. The  $Q_m$  of sulfamethazine by the HPC was 869.6 mg g<sup>-1</sup> at 308 384 K. The strong adsorption affinity of the HPC to sulfamethazine was due to its high SSA (2784 385 m<sup>2</sup> g<sup>-1</sup>) and pore volume (1.382 cm<sup>3</sup> g<sup>-1</sup>). Owing to the well-defined 3-D interconnected 386 hierarchical porous structure, the adsorption kinetics of the HPC was fast in the first 30 min 387 [79]. The same research group used carbon nanotubes as hard templates to synthesize HPCs, 388 which had excellent adsorption capacities for chloramphenicol and tetracycline (1297 mg  $g^{-1}$ 389 and 1067.2 mg g<sup>-1</sup>, respectively), far higher than previously reported values. The sample had 390 a wide pore size distribution, ranging from < 2 to 100 nm [80]. 391

Tripathi et al. [81] reported bisphenol A (BPA) removal by hierarchically ordered micromesoporous carbon with an ultra-high adsorption capacity of 1106 mg g<sup>-1</sup>, which is three times larger than that activated carbons of a previous study of Liu et al. [82]. In particular, the sizes of the micropores and mesopores were tailored and enlarged to 1.3 nm and 9.0 nm, respectively, to accommodate the molecular dimensions of BPA for achieving an optimal adsorption performance. This was achieved by controlling the condensation behavior of phloroglucinol-terephthalaldehyde resin. Besides, kinetic studies revealed that the mesopores were the key to promote adsorbate diffusion through the pore channels, and the smaller secondary micropores contributed to the high adsorption capacity, achieving a removal rate of 86% (743 mg g<sup>-1</sup>), which was higher than that of the single mesoporous material. The authors demonstrated that controlling the porosity and structural features of ordered carbon materials were effective to promote the adsorption of BPA [81].

In order to explore the effect of porosities (of all sizes) and to demonstrate the relative role 404 of each pore size class on the adsorption process in a liquid media, Bulavová et al. [83] 405 prepared single microporous, micro-mesoporous and micro-meso-macroporous carbon 406 materials by changing the synthesis conditions. Methylene blue and phenol (molecular 407 diameters vary greatly; 1.3 nm and 0.75 nm, respectively) were chosen as molecular probes 408 for testing the adsorbents. Both adsorption rate and adsorption capacity of micro-mesoporous 409 carbon were far more favorable than those of the single microporous carbon. Meanwhile, the 410 presence of macropores further promoted the adsorption. The functional groups of 411 micropores were different from those of the micro-mesoporous and micro-meso-macroporous 412 413 carbons. However, the authors suggested that the functional groups were not the decisive factor for adsorption because the diffusion restriction posed by the microporous system made 414 it impossible for the adsorbates to reach the adsorbent active sites [83]. 415

For the adsorption of super-large molecules (e.g., gasoline), it is important to mention that the macropore volume/mesopore volume needs to be relatively high. It is reasonable to consider that macropores play a predominant role in adsorbing capacious gasoline, while mesopores can induce roughness on the surface of the macropores and high surface area, which can benefit the dispersive interactions between carbon basal planes and the adsorbate[63].

The analysis of adsorption mechanisms of sulfamethazine antibiotics showed higher adsorption capacities of HPCs, which might be attributed to its high SSA (i.e., 2784 m<sup>2</sup> g<sup>-1</sup>) and large pore volume (i.e., 1.382 cm<sup>3</sup> g<sup>-1</sup>). In addition, the van der Waals forces between the antibiotic molecules and the adsorbent, and the  $\pi$ - $\pi$  electron-donor-acceptor interactions of the antibiotic molecules at the plane of benzene rings might also affect the adsorption [79]. Furthermore, hydrophilic antibiotics had a higher adsorption on the studied HPC, indicating that the hydrophilic process promoted the adsorption capacity [80].

429

#### 430 **6.** Conclusions and future prospects

HPCs are emerging adsorbents that can be promisingly applied for the removal of 431 organic pollutants from contaminated water. The highly porous structure and easily tunable 432 structural arrangements provide advantages in designing highly effective HPCs for organic 433 contaminants removal. Moreover, waste biomass such as crop residues, agricultural wastes 434 435 and food wastes can be used to produce HPCs, introducing an additional benefit of waste management and low-cost. The use of HPCs in water treatment related studies has increased 436 just recently, but applications of these materials for treating organic contaminants in 437 wastewater are not yet at a satisfactory level. According to SCOPUS database, the first 438 439 publication on HPC application in removing aqueous organic contaminants appeared in 2012 [84]. Since then, there has been no extensive growth in this area of research. This might be 440 due to the strong focus of HPCs concerning super capacitor and energy storage applications. 441 Hence, there is a huge research scope in the use of HPCs for water remediation. Optimal and 442

effective HPC modification methods to enhance their ability to remove aqueous organic contaminants are also not adequately available. Since HPCs are easily tunable, they have a promising capacity to be used in commercial filtration materials, such as membranes and filters for household and large-scale wastewater treatments. Hence, more research is essential to improve and implement the HPCs for water remediation as an energy- and costinexpensive technology.

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Hierarchical porous carbon materials	Image (SEM, TEM)	Synthesis method	Properties	Application	Reference
Hierarchically macro- mesoporous carbon monolithic	10 µm 10 µm 10 µm 10 µm 10 µm 0 5 µm	Dual-templating approach(nanocasting) with tri-block polymer F127 as the soft template, monolithic silica as hard template and resol as precursor	SSA=1036 m <sup>2</sup> /g V (pore volume)=2.9cm <sup>3</sup> g <sup>-1</sup> Relatively disordered macro- mesopore size Macropore size about 2 μm Mesopore size about 7.08 nm	As super-capacitor materials	[34]
Hierarchically/bimodal macro-mesoporous carbon	SEM 211 (1) 10 (	Dual-templating approach with tri-block polymer F127 as the soft template, monodispersed silica colloidal crystal as the hard template and resols as precursor	SSA =760 m <sup>2</sup> /g V (pore volume)=1.25cm <sup>3</sup> g <sup>-1</sup> Ordered macro-mesopore structure Macropore size about 230-430 nm Mesopore size about 11 nm Highly ordered face-centered cubic macrostructure Interconnected windows with a size of 30-65 nm		[36]
Hierarchically carbon spheres with hollow macroporous core and mesoporous shell	TEM SEM	Dual-templating approach (aqueous route) with monodisperse silica spheres as hard template, F127 as soft template and resol as precursor	SSA =972 m <sup>2</sup> g <sup>-1</sup> V (pore volume)=1.27 cm <sup>3</sup> g <sup>-1</sup> Macropore size about 250-300 nm Mesopore size 6–9 nm Thin shell thickness of 5 nm	As super-capacitors Highest capacitance (125 F g <sup>-1</sup> )	[35]

# Table 1: Synthesis, properties and applications of HPCs.

Dimensionally ordered macroporous carbon with mesoporous walls	a 05 200 nm d 200 nm d 200 nm SEM	Dual-templating approach with PS latex (macropore former and carbon source) and colloidal silica (mesopore former)	SSA =1500 m <sup>2</sup> g <sup>-1</sup> V (pore volume)=2.62 cm <sup>3</sup> g <sup>-1</sup> A bimodal porous structure Mesopore size 5 nm Macropore size 190 nm	As electric double- layer capacitors Showed highest capacitance of 120 F g <sup>-</sup>	[45]
	b 100 mm Merry Spinlage Dri 100 Eq 1 Jun Merry Spinlage Dri 100 Eq 1 Jun SEM	Dual-templating approach one-pot method via in situ self-assembly of colloidal polymer (PMMA macropore former) and silica spheres (mesopore former) with sucrose as the carbon source	SSA =818 m <sup>2</sup> g <sup>-1</sup> V (pore volume)=1.29 cm <sup>3</sup> g <sup>-1</sup> A bimodal porous structure Mesopore size 9 nm Macropore size 370 nm Small windows with a size of 120 nm	As a Catalyst Support They show very good support of the Pt-Ru alloy catalyst in a direct methanol fuel cell.	[50]
Hierarchically macro/mesoporous carbon monoliths	SEM TEM	Nanocasting pathway by synthesizing parent SiO <sub>2</sub> monolith (hydrothermal treatment, removal of P123 template and immerse into furfuryl alcohol solution) and carbon monoliths	SSA =1354 m <sup>2</sup> g <sup>-1</sup> Hexagonal mesoporosity and three-dimensionally connected macroporosity Mesopore size 3.3-4 nm Macropore size 32.46 $\mu$ m High macroporous cumulative volume is 48.6 cm <sup>3</sup> g <sup>-1</sup> Regular cylindrical shape, strong hydrophobicity and low densities	As adsorbent high adsorption capacity of 613 mg /g for bilirubin and oil with weight 23–48 times of their own in a few seconds.	[63]

Hierarchically macro/mesoporous carbon	for the second s	Dual-templating approach with $SiO_2$ opal as a hard template, (F127 or P123) as a soft template, and phenolic resin as a precursor	SSA =1667 m <sup>2</sup> g <sup>-1</sup> V (pore volume) = 4.80 cm <sup>3</sup> g <sup>-1</sup> Highly periodic arrays of uniform and bimodal macropores with surrounding quasi-2D hexagonal mesoporous walls Mesopore size 18.1 nm Macropore size about 290 nm Interconnected windows with sizes of about 30 nm		[38]
Hierarchically ordered macro-mesoporous carbon monoliths	A B B C C C C C C C C C C C C C C C C C	dual-templating technique with poly (methyl methacrylate) (PMMA) colloidal crystals, F127 as soft templates and	SSA = 505 and 464 m <sup>2</sup> g <sup>-1</sup> V (pore volume) = 0.37 and $0.34 \text{ cm}^3 \text{ g}^{-1}$ Face-centered cubic and 2D- hexagonal mesoporous structure Mesopore size 2.7 nm and 3 nm Macropore size about 340 nm and 350 nm macropore window sizes of 150 nm and 130 nm		[51]
The 1D hierarchical macro-mesoporous carbon nanofiber	(a) <u>June</u> TEM	Electrospinning pathway with polyacrylonitrile as carbon precursor and nano-CaCO <sub>3</sub> as dual purpose template. Nano-CaCO <sub>3</sub> template decomposed to released $CO_2$ to develop mesopores, and CaO nanoparticles to develop	$SSA = 123 \text{ m}^2 \text{ g}^{-1}$ A total macropore volume of 18.7 cm <sup>3</sup> g <sup>-1</sup> and a porosity degree of 50.4% Continuous and uniform nanofibers structure with winding mesopores The diameter of the nanofibers is 900 nm Pore size mainly distributed at	As catalyst support	[85]

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	C I I I I I I I I I I I I I I I I I I I	macropores	4.2 nm		
Hierarchically disorderd Micro-mesoporous carbon powder	(b) (c) (c) (c) (c) (c) (c) (c) (c	Carbonization and KOH activation by using corn cobs as the carbon resource with ultra- attenuation and purification by coupled ultrasonication-milling	SSA =2288 m <sup>2</sup> g <sup>-1</sup> V (pore volume) = $1.21 \text{ cm}^3 \text{ g}^-$ With a high mesopore ratio of about 44.6% Average pore diameter 2.1 nm	As high-performance super-capacitor	[86]
Hierarchical macroporous carbon foam with bicontinuous micro- mesoporous	(c) (c) (c) (c) (c) (c) (c) (c)	Carbonization and KOH activation by using glucose as the carbon resource	SSA =3106.7 m <sup>2</sup> g <sup>-1</sup> V (pore volume) = $1.82 \text{ cm}^3 \text{ g}^-$ Typical amorphous structure and a block-like structure in micrometer scale The thickness of 5–10 mm Super-large macropores (0.5–5 µm) Bicontinuous large micropores (1.14 nm) Small mesopores (3.12 nm) With a high mesopore and micropore ratio of about 66.5% and 26.9%	Super-capacitor	[87]

Hierarchically mesoporous monoliths	micro- carbon	(c) (d) (e) (e) (f) (f) (f) (f) (f) (f) (f) (f	Dual-templating approach with sugarcane bagasse as the scaffold, F127 as the soft template, and resol as carbon precursor	$\begin{array}{l} SSA = 544 \ m^2 \ g^{-1} \\ V \ (pore \ volume) = \ 0.283 \ cm^3 \\ g^{-1} \\ 2-D \ hexagonal \ mesostructures \\ and \qquad micrometer-sized \\ lignocellulosic \qquad macroporous \\ texture \\ Enhanced \qquad degree  of \\ graphitization \\ Mesopore \ size \ about \ 3 \ nm \\ Micropore \ size \ about \ 0.5 \ nm \\ Micropore \ percentages \ of \ 66- \\ 67\%. \end{array}$	As capacitances Show good electrochemical property and the specific capacitances are in the range 190– 234 F g <sup>-1</sup>	[53]
Hierarchically macroporous and mesoporous monoliths	micro- d micro- carbon	PCM-4,15-0.04 2 µm SEM	Phase separation induced by polymerization of resorcinol and formaldehyde to produce macropores or mesopores, and CO <sub>2</sub> activation to produce micropores	$\begin{array}{l} SSA = 1000 - 2400 \ m^2 \ g^{-1} \\ V \ (pore \ volume) = \ 0.4 - 0.92 \\ cm^3/g \\ No \ regularly \ arranged \\ structures in the mesopore size \\ range. \\ Macro- \ and \ mesopores \ size \\ were \ tailored \ from \\ approximately \ 10 \ \mu m \ to \ 2 \ mm \\ With \ a \ low \ degree \ of \\ graphitization \end{array}$	As CO <sub>2</sub> adsorption capacity	[88]
Hierarchical mesoporous monolith	macro- carbon	e 100nm TEM	Dual-templating approach with P123 as soft template, mesoporous silica (KIT- 6) as hard template and lignin as carbon precursor	SSA = 803 m <sup>2</sup> g <sup>-1</sup> V (pore volume) = 0.86 cm <sup>3</sup> g <sup>-1</sup> A well-interconnected 3D porous network Mesopore size about 3.4, 6.4 and about 50 nm Macropore size about 100 nm The thick of carbon monolith is 310 $\mu$ m	As the high- performance electrode for electric double layer supercapacitor.	[49]

Hierarchically micro/mesoporous carbons	a 400 nm b 20 nm SEM	"Nanopore lithography" strategy by graphene oxide (GO) nanosheets enclosing ZIF-8 particles as templates to produce mesoporous structure	SSA = 1300 -1350 m <sup>2</sup> g <sup>-1</sup> V (pore volume) = 1.1-1.2 cm <sup>3</sup> $g^{-1}$ Mesopore size about 3-4 nm	As electrochemical energy storage	[59]
Hierarchical macro- meso-microporous carbon monolith	C30-Ca850 C30-Ca850 C30-Ca850 SEM	Template-free approach by ultrasonic irradiation to produce macropores, $Ca(NO_3)_2$ impregnation followed by $CO_2$ activation to produce mesopores and resorcinol-formaldehyde as carbon precursor	$\begin{array}{l} SSA = 624 \ m^2 \ g^{-1} \\ Slit-pore \ shape \ of \ mesopore \\ and \ not \ well-defined \\ mesopore \ size \ distribution \\ Mean \ pore \ size \ diameter \ of \\ 1.2 \ lm \\ Mesopore \ volume \ of \ 0.38 \ cm^3 \\ g^{-1} \\ Micropore \ volume \ of \ 0.22 \ cm^3 \\ g^{-1} \end{array}$		[60]
Three-dimensionally disordered interconnected micro-, meso- and macropores carbon	n (1) SEM (1) (1) (1) (1) (1) (1) (1) (1)	Template-free approach by constructing carbonyl crosslinking bridges between polystyrene chains	SSA = $679 \text{ m}^2 \text{ g}^{-1}$ S (micropore surface area) = $390 \text{ m}^2 \text{ g}^{-1}$ V (pore volume) = $0.66 \text{ cm}^3 \text{ g}^{-1}$ Micro-, meso-, macropore volumes account for 32%, 56% and 12% Mesopore size < 2 nm Mesopore size about 2-50 nm Macropore size about 50-400 nm		[61]

Ordered, uniform, macroporous carbons with mesoporous walls	SEM	Dual-templating approach by mixing monodisperse polystyrene spheres and silica particles with divinylbenzene as carbon precursor	SSA =465 m <sup>2</sup> g <sup>-1</sup> V (pore volume) =1.32 cm <sup>3</sup> g <sup>-1</sup> Periodically ordered, bimodal porous carbon Mesopore size about 10 nm Macropore size about 317 nm Interconnected windows with a size of 110 nm	As Catalyst Supports in Direct Methanol Fuel Cells	[44]
Hierarchical micro- mesoporous carbons	с 20 пт f	Template carbonization and CO <sub>2</sub> activation by using CMK-3 and CMK- 1 as mesoporous template	SSA =2749 and 2696 m <sup>2</sup> g <sup>-1</sup> V (pore volume) =2.09 and 1.87 cm <sup>3</sup> g <sup>-1</sup> V (micropore volume) =0.96 and 0.94 cm <sup>3</sup> g <sup>-1</sup> V (mesopore volume) =1.13 and 0.93 cm <sup>3</sup> g <sup>-1</sup> Mesopore size about 3-4 nm	As super-capacitor electrode materials	[57]
Hierarchical micro- /meso-macroporous carbon aerogel monoliths	(b) 5 µm SEM	Sol–gel polymerization to produce macroporous carbon aerogel and CO <sub>2</sub> activation to produce mesopores or micropores	SSA =1500-3000 m <sup>2</sup> g <sup>-1</sup> V (pore volume) =0.78-1.88 cm <sup>3</sup> g <sup>-1</sup> V (micropore volume) = 0.583 cm <sup>3</sup> g <sup>-1</sup> Carbon aerogel monoliths with densities about 0.55 g cm <sup>-3</sup>		[89]

Hierarchical 3-D (e) (d) (d) (d) (d) (d) (d) (d) (d) (d) (d	Template carbonization by using silica nanospheres as	$SSA = 2784 \text{ m}^2 \text{ g}^{-1}$ V (pore volume) = 1.382 cm <sup>3</sup> g <sup>-1</sup>	As adsorbent sulfamethazine	for [79]
carbon SEM Official official o	macroporous template and KOH activation to produce mesopores and micropores	A three-dimensional continuous cellular-like structure V (micropore volume) =0.678 cm <sup>3</sup> g <sup>-1</sup> D (average size) = 2.155 nm Mesopore size about 3-4 nm Macroporous diameter of about 200 nm The presence of a few graphite structure in the carbon framework		

Hierarchical porous carbon material	Carbon precurs or	Modific ation	Organic contami nant	Contami nant concentr ation	Background solution		Adsorption capacity	Adsorption mechanism	Reference
					Ionic strength	рН			
Hierarchically macro/mesopo rous carbon monoliths	Tetraetho xysilane	-	Bilirubin	200 mg L <sup>-1</sup>	0.2 M NaOH solution	7.4	613 mg g <sup>-1</sup>	Hierarchically porous structure	[63]
Hierarchical 3D interconnected micro-meso- macroporous carbon	sodium lignin sulfonate	KOH activatio n	Sulfamet hazine	Initial concentr ations of 50, 80, 120, 160, 200, 250, and 300 mg L <sup>-1</sup>	2.0 mg of 3DLHPC was mixed with 10 mL of SMZ solution $(C_0 = 160 \text{ mg } \text{L}^{-1})$ with the humic acid concentration of 10, 25, 50, and 100 mg L <sup>-1</sup> , respectively.	3.0-9.0	869.6 mg g <sup>-1</sup>	Mainly due to the higher specific surface area and pore volume, Partly van der Waals force, $\pi$ – $\pi$ EDA interaction, electronic interaction, hydrophobic interaction, and hydrogen bonding interaction	[79]
Hierarchical porous carbons well-defined hollow tubular structures in nano-size with micro- mesporous	sodium lignin sulfonate	KOH activatio n	Tetracycl	Initial concentr ations in a range of 50– 400 mg L <sup>-1</sup>	Na <sup>+</sup> ,K <sup>+</sup> ,Ca <sup>2+</sup> ,Fe <sup>3+</sup> ,Co <sup>2</sup> <sup>+</sup> (0.01 M), and humic acid (5–50 mg L <sup>-1</sup> )	3–8 for TC (250 mg L <sup>-1</sup> )	1297.0 mg g <sup>-1</sup>	oxygen containing functional groups	[80]

Table 2: Use of HPCs and modified HPCs in organic contaminant removal from water and wastewater.

Hierarchical porous carbons well-defined hollow tubular structures in nano-size with micro- mesporous	sodium lignin sulfonate	KOH activatio n	Chloram phenicol	Initial concentr ations in a range of 20– 200 mg L <sup>-1</sup>	Na <sup>+</sup> ,K <sup>+</sup> ,Ca <sup>2+</sup> ,Fe <sup>3+</sup> ,Co <sup>2</sup> <sup>+</sup> (0.01 M), and humic acid (5–50 mg L <sup>-1</sup> )	3–11 for CAP (200 mg L <sup>-1</sup> )	1067.2 mg g <sup>-1</sup>	oxygen containing functional groups	[80]
Ferromagnetic hierarchical porous carbon	phenolic resin	-	Methyle ne orange	1.5 x 10 <sup>4</sup> M	water		0.16 mg m <sup>-2</sup>	the strong pore surface-molecule electrostatic force and volume filling effect	[90]
Macro-meso- micro hierarchical porous carbon	bovine bone	KOH activatio n	Sulfamet hazine	200 mg L <sup>-1</sup>	water	-	$1194 \text{ mg g}^{-1}$		[78]
Macro-meso- micro hierarchical porous carbon	bovine bone	KOH activatio n	Chloram phenicol	200 mg L <sup>-1</sup>	water	-	$1240 \text{ mg g}^{-1}$		[78]
Hierarchical macroporous carbon with the mesopores and size- uniform wormhole-like micropores	Furfuryl alcohol	-	Methyle ne blue	500 mg L <sup>-1</sup>	water	-	250 mg g <sup>-1</sup>	hierarchically porous structure	[77]
Microporous or micro- mesoporous or micro-meso- macroporous carbon	prepared by sol- gel process from Resorcin	-	Phenol	1mM		-		hierarchically porous structure	[83]

materials	ol and								
	formalde								
Hierarchically	Tetraethy	-	Bispheno	345 mg	ethanol	-	1106 mg g <sup>-1</sup>	hierarchically	[81]
ordered micro-	1		1 A	$L^{-1}$				porous structure	
mesoporous	orthosilic								
carbon	ate								
Hierarchical	pomelo	-	Rhodami	$20 \text{ mg } \text{L}^{-1}$	Water	-	249.8 mg g <sup>-1</sup>	superhydrophilic	[91]
porous N-	peel		ne B	1				surfaces	
doped carbon									
<u>Iliananahiaal</u>	nomala		Mathril	20 mg I -	Watan		416.2 mg g-1	ann amhridean bili a	[01]
nierarchicai	pomeio	-	orange	$\frac{20}{1}$ mg L	water	-	410.3 mg g	supernyuropinite	[91]
doned carbon	peer		orange					surraces	
microspheres									
Hierarchical	pomelo	-	Rhodami	20 mg L <sup>-</sup>	Water	-	120 mg g <sup>-1</sup>	superhydrophilic	[91]
porous N-	peel		ne 6G	1			66	surfaces	
doped carbon	1								
microspheres									
Heteroatom	Glucose	-	Sulfachl	20 mg		7	$220 \mathrm{~mg~g^{-1}}$	well-defined pore	[25]
(nitrogen and			oropyrid	$L^{-1}$				structure	
sulfur)-			azine					Synergetic effect of	
codoped								N and S	
porous carbons									
(N-SPCs)									
N-S-PC-2									
(15% thiourea)	Clucosa		Slfachlor	20 mg		7	$100 \text{ mg s}^{-1}$	wall defined nora	[25]
(nitrogen and	Olucose	-	opyridazi	$I^{-1}$		/	190 mg g	structure	[23]
sulfur)-			ne	L				structure	
codoned			ne						
porous carbons									
(N-SPCs)									
N-S-PC-1 (5%									
thiourea)									

Heteroatom (nitrogen and sulfur)- codoped porous carbons (N-SPCs) N-S-PC-3	Glucose	-	Sulfachl oropyrid azine	20 L <sup>-1</sup>	mg	7	126 mg g <sup>-1</sup>	well-defined pore structure	[25]
(25% thiourea) Nitrogen doped activated porous carbon fibers (FeCo <sub>2</sub> @APC Fs)	polyacryl onitrile/p olybenzo xazine	KOH activatio n	Methyle ne blue	20 L <sup>-1</sup>	mg	-	99.08% remova efficiency within 25 min	<ul> <li>High surface area,</li> <li>Activation of peroxymonosulfate with reactive oxygen radicals (SO<sub>4</sub><sup>-</sup>-, <sup>•</sup>OH, and <sup>1</sup>O<sub>2</sub>)</li> </ul>	[92]
Nitrogen doped activated porous carbon fibers (FeCo <sub>2</sub> @APC Fs)	polyacryl onitrile/p olybenzo xazine	KOH activatio n	Sulfamet hoxazole	20 L <sup>-1</sup>	mg		68.8% remova efficiency within 25 min	<ul> <li>High surface area,</li> <li>Activation of peroxymonosulfate with reactive oxygen radicals (SO<sub>4</sub><sup>-</sup>-, <sup>•</sup>OH, and <sup>1</sup>O<sub>2</sub>)</li> </ul>	[92]
Nitrogen doped activated porous carbon fibers (FeCo <sub>2</sub> @APC Fs)	polyacryl onitrile/p olybenzo xazine	KOH activatio n	Phenol	20 L <sup>-1</sup>	mg		62.28% remova efficiency within 25 min	<ul> <li>High surface area,</li> <li>Activation of peroxymonosulfate with reactive oxygen radicals (SO<sub>4</sub><sup>-</sup>-, <sup>•</sup>OH, and <sup>1</sup>O<sub>2</sub>)</li> </ul>	[92]
Nitrogen doped activated porous carbon fibers (FeCo <sub>2</sub> @APC	polyacryl onitrile/p olybenzo xazine	KOH activatio n	Bispheno l-A	20 L <sup>-1</sup>	mg		53.76% remova efficiency within 25 min	<ul> <li>High surface area,</li> <li>Activation of peroxymonosulfate with reactive oxygen radicals (SO<sub>4</sub><sup>-</sup>-, <sup>•</sup>OH, and</li> </ul>	[92]

Fs)								<sup>1</sup> O <sub>2</sub> )	
Nitrogen doped activated porous carbon fibers (FeCo <sub>2</sub> @APC Fs)	polyacryl onitrile/p olybenzo xazine	KOH activatio n	Chloroph enol	20 L <sup>-1</sup>	mg	63.96% efficiency 25 min	removal within	High surface area, Activation of peroxymonosulfate with reactive oxygen radicals (SO <sub>4</sub> <sup>-</sup> -, <sup>•</sup> OH, and <sup>1</sup> O <sub>2</sub> )	[92]



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<sup>810 (</sup>quoted on 15 January 2020)



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Fig. 2: Synthesis routes for HPCs using a) soft template and dual template [93], b) soft template [47], c) hard template [94], and d) multiple templates fabrication [95].



Fig. 3: Schematic illustration of a) KOH activation [96], b) KOH activation with increasing
concentrations [97], c) KOH and hydrothermal activations and corresponding SEM images
[98], d) CO<sub>2</sub> and H<sub>2</sub>O activations [74], and e) KOH activation, temperature treatments and
nitrogen doping [99].