

1 **Experimental Investigation of Energy Storage Properties and Thermal Conductivity**
2 **of A Novel Organic Phase Change Material/MXene as A New Class of**
3 **Nanocomposites**

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25 **Abstract**

26 Energy storage is a global critical issue and important area of research as most of the renewable
27 sources of energy are intermittent. In this research work, recently emerged inorganic nanomaterial
28 (MXene) is used for the first time with paraffin wax as a phase change material (PCM) to improve
29 its thermo-physical properties. This paper focuses on preparation, characterization, thermal
30 properties and thermal stability of new class of nanocomposites induced with MXene nanoparticles
31 in three different concentrations. Acquired absorbance (UV-Vis) for nanocomposite with loading
32 concentration of 0.3 wt.% of MXene achieved ~39% enhancement in comparison with the pure

33 paraffin wax. Thermal conductivity measurement for nanocomposites in a solid state is performed
34 using a KD2 PRO decagon. The specific heat capacity (c_p) of PCM based MXene is improved by
35 introducing MXene. The improvement of c_p is found to be 43% with 0.3 wt.% of MXene loaded
36 in PCM. The highest thermal conductivity increment is found to be 16% at 0.3 wt.% concentration
37 of MXene in PCM. Decomposition temperature of this new class of nanocomposite with 0.3 wt.%
38 mass fraction is increased by ~6%. This improvement is beneficial in thermal energy storage and
39 heat transfer applications.

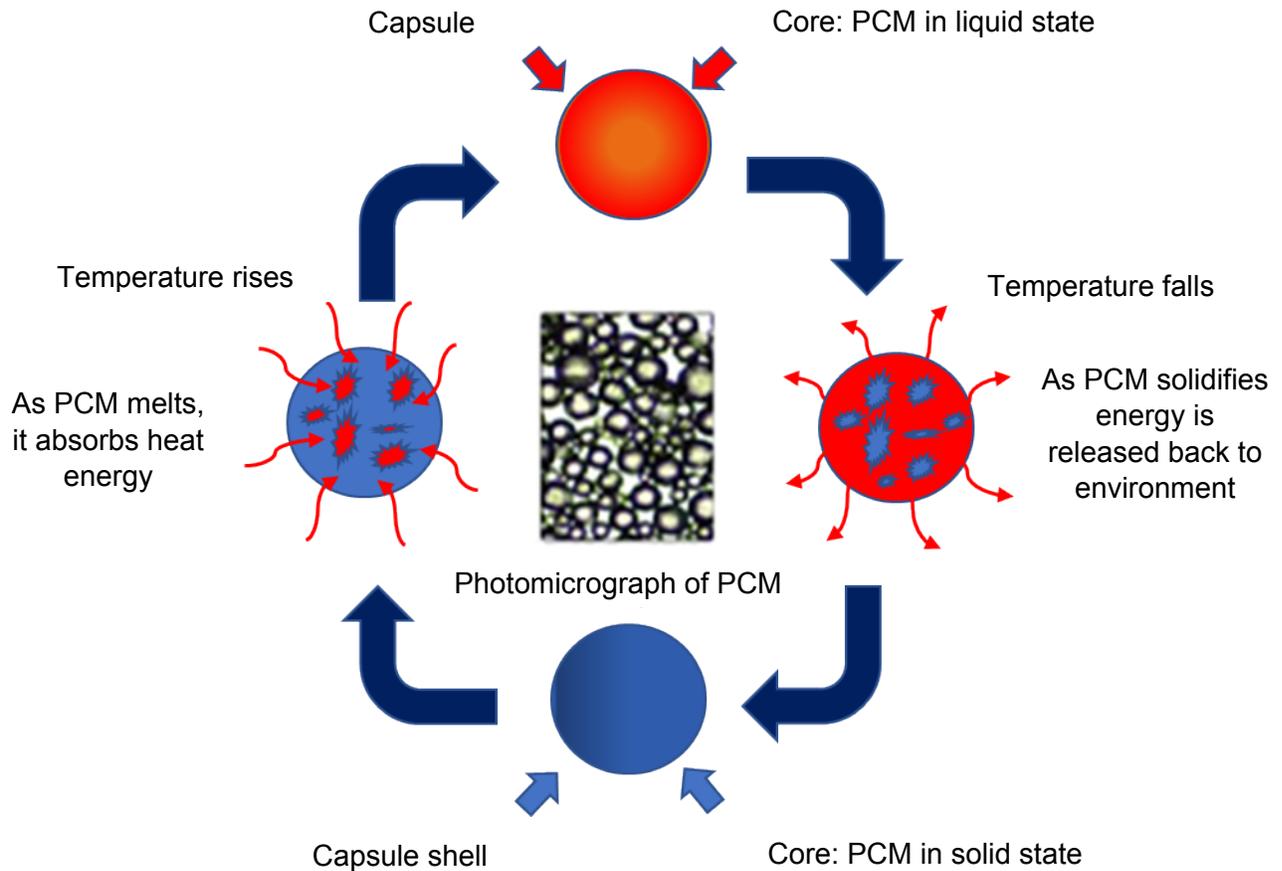
40 **Keywords.** MXene, PCM, Thermal energy storage, Thermal conductivity, Thermal stability

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42 **1. Introduction**

43 Using renewable energies is one of the solutions for many environmental challenges that we
44 face today. Solar energy is regarded as the most prominent source of renewable energy moving
45 forward as the hourly incident solar flux on the surface of the earth is more noteworthy than annual
46 global energy consumption [1]. Solar thermal harvesting was studied extensively over the last
47 decade and therefore it became one of the most efficient and reliable technologies for supplying
48 energy. To make thermal energy storage (TES) cost effective and efficient, further improvement
49 with the introduction of emerging nano-materials is needed. TES offered promising solutions to
50 the world's energy as well as environmental challenges [2, 3]. An effective way to store thermal
51 energy is employing a latent heat storage system with organic/inorganic phase change material
52 (PCM). PCMs can absorb and/or release a remarkable amount of latent heat as a result of a phase
53 transition when the phase transition temperature is within a specified temperature range. Phase
54 change phenomenon (encapsulated PCM) is explained in Figure 1.

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59 **Figure 1.** Performance of organic/inorganic phase change materials in solid/liquid states (encapsulated
60 PCM) [4]

61 PCMs were applied in the area of both thermal isolation and energy storage, such as air-
62 conditioning [5], solar thermal storage [6, 7], building energy management [8, 9]. Phase change
63 temperatures and phase transition enthalpies are dependent on materials [10]. The choice of a
64 correct PCM depends generally on proper thermal characteristics for a specific application, in
65 addition to its stability and cost. PCMs are classified into organic, inorganic, and eutectics
66 according to their chemical composition and structure. Their details are explained in [4, 11, 12].
67 Table 1 demonstrates a comprehensive comparison among different class of PCMs.

Table 1

Comparison of thermophysical properties of different PCMs [4]

Property of Characteristics	Paraffin Wax	Non-Paraffin Organics	Hydrates Salts	Metallic
Heat of Fusion	High	High	High	Medium
Thermal Conductivity	Very Low	Low	High	Very High
Melting Temperature (°C)	-20 to +100	5 to +120	0 to +100	150 to +800
Latent Heat (kJ/kg)	200 to 280	90 to 250	60 to 300	25 to 100
Corrosiveness	Non-Corrosive	Mildly Corrosive	Corrosive	Varies
Thermal Cycling	Stable	Elevated temperature can cause decomposition	Stability/instability depends on the corresponding solid-liquid phase diagram	Stable

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69 Paraffin, a mixture of mostly straight chain n-alkanes with general formula of $\text{CH}_3-(\text{CH}_2)_n-$
70 CH_3 is probably the most common and widely used organic PCMs in many TES applications.
71 Paraffin waxes which are chemically compatible with most metals owing to the characteristics of
72 large latent heat of fusion, high storage density, minimal tendency to super-cooling, desirable
73 thermal and chemical stability, low vapor pressure of the liquid phase, non-toxic, noncorrosive,
74 and relatively low price [10, 13]. These clearly merit paraffin wax's utilization in TES applications
75 for broad range of temperatures. In solar thermal systems, paraffin as a thermal energy reservoir
76 (TER) absorbs thermal energy as heat during the day time (called charging process) when solar
77 energy is accessible in sufficient amount and release the thermal energy over the night (called
78 discharging process) [14]. Despite all its advantages, low thermal conductivity of paraffin wax
79 (0.21–0.24 W/m K) can be a drawback in solar and high power applications that significantly
80 decelerates the energy charging/discharging rates [2, 13]. Researchers tried to improve the
81 effective thermal conductivity of paraffin and other popular PCMs using various approaches.

82 Dispersing high thermal conductive particles, such as carbon, metals, or graphite within PCMs
83 was one of the approaches. Graphite usually improves thermal conductivity of paraffin in various
84 forms such as expanded filler, or nanoparticle [15]. Earlier efforts investigated the enhancement
85 of the heat transfer between PCM and heat transfer fluid using stationary, highly thermal
86 conductive pieces like stainless steel and copper as well as graphite–PCM composite as an additive
87 into PCMs [16]. The results of this research demonstrated that graphite and copper both
88 successfully elevated the heat flux. However, stainless steel had only limited contribution. Mills
89 et al. [17] indicated that thermal conductivity of paraffin/expanded graphite composites was
90 enhanced by 20 to 60 times. Chen et al. [18] managed to analyze the energy storage process of a
91 solar collector with an integrated porous structure filled with paraffin as the phase change medium.
92 These solutions not only increase the cost of PCMs, but also solid particles loaded to enhance
93 thermal properties reduce the amount of PCM that can be used in a fixed volume [3]. There is a
94 recent tendency of employing various nanoparticles as thermal enhancement additives into PCMs
95 instead of bulky metal pieces and particles. Upon the review by Khodadadi et al. [19], a wide range
96 of nanostructure materials are utilized as thermal conductivity promoters. These include carbon-
97 based nanostructures (nanofibers, nanoplatelets and graphene flakes), carbon nanotubes, both
98 metallic (Ag, Al, C/Cu and Cu) and metal oxide (Al_2O_3 , CuO, MgO and TiO_2) nanoparticles and
99 silver nanowires. Authors concluded that carbon-based nanostructures exhibited far greater
100 thermal conductivity enhancement compared to metallic/metal oxide nanoparticles. In recent
101 decade, carbon nanotubes (CNTs) were utilized to increase effective thermal conductivities of
102 paraffin wax and other organic PCMs [20-22]. Multi-walled carbon nanotubes (MWCNTs)
103 showed excellent potential for heat transfer applications [22]. Wang et al. [23] added 2 wt.% of
104 carbon nanotubes to paraffin and showed that thermal conductivity was increased by 35 and 40%

105 in solid and liquid states, respectively. Increment of 13% of thermal energy storage capacity of
106 paraffin was obtained with adding 1% carbon nanotubes [3]. Reduction of melting temperature
107 and improved thermal energy storage were reported with the loadings of MWCNT in paraffin wax
108 [24]. Tang et al. [25] reported the enhancement of thermal storage by more than 10% for paraffin
109 wax by loading of 1 and 5 wt.% f-MWCNT, whereas the loading of 10% of f-MWCNT decreased
110 the thermal storage value. Some researchers reported that there is no change in thermal storage
111 value for organic phase change materials with the presence of CNTs and xGNP (graphene
112 nanoplatelets) [26, 27]. However, most of the studies reported a significant reduction in thermal
113 storage for organic phase change materials with the loading of the nanoparticles [28-31].

114 Recently, a new family of two-dimensional materials consist of transition-metal carbides and/or
115 carbonitrides so-called “MXene” was manufactured via selective etching out the “A” layers from
116 $M_{n+1}AX_n$ phases (where M is an early transition metal, A is an A-group element, X is C and/or N,
117 and $n = 1, 2, \text{ or } 3$) and was first introduced by Naguib et al. [32] in 2011. Due to outstanding
118 physicochemical properties such as hardness, hydrophilic nature, excellent oxidation resistance,
119 high thermal and electrical conductivity, high thermal stability and high surface area of MXene, it
120 has been applied in different applications [33]. MXenes have recently gained extreme attention in
121 many exciting applications including ion batteries [24], hydrogen storage [25], water splitting [26],
122 electronic devices [27], and supercapacitors [28]. However, MXene nanoparticle has not been
123 explored in thermal energy storage systems and heat transfer applications yet. Specifically, there
124 is no study on the specific heat capacity, thermal conductivity and thermal stability of PCM with
125 MXene. The present experimental work provides a systematic approach to demonstrate the energy
126 storage and thermal conductivity analysis of paraffin wax induced with MXene nanoparticles. This
127 study will be useful in thermal energy storage applications. Therefore, comprehensive

128 investigation should be carried in this area to evaluate its full potential in thermal energy and heat
129 transfer areas.

130 **2. Preparation of Nanocomposite (PCM/MXene)**

131 2.1. Materials

132 Organic phase change material (PLUSICE A70) with melting point of 70 °C, density of 890
133 kg/m³, volumetric heat capacity of 154 MJ/m³ and specific heat capacity of 2.2 kJ/kg K (at 25 °C)
134 was procured from Phase Change Materials Products Ltd Company [34]. PLUSICE A70 is also
135 known as Paraffin Wax, PW70.

136 2.2. Preparation of MXene (Ti₃C₂)

137 In the synthesis of Ti₃C₂, the following materials were used without any further purification: Ti
138 powder (-325 mesh, 99% purity, Alfa Aesar), aluminium powder (-100+325 mesh, 99.5% purity,
139 Alfa Aesar), titanium carbide powder (-325 mesh, 98% purity, Sigma Aldrich), sodium fluoride
140 (99% purity, Alfa Aesar), hydrochloric acid (37.5% wt., Sigma Aldrich), and sodium hydroxide
141 (97% purity, pellets, Sigma Aldrich). The precursor MAX phase Ti₃AlC₂ is synthesized using a
142 molar ratio of 1:1.2:2 of (Ti:Al:TiC) elemental powders, mixed with a pestle and mortar, followed
143 by a thermal treatment using a tube furnace under Ar atmosphere at 1,400 °C for 2 h (5 °C /min
144 heating/cooling rate). For the etching, 4.85 g of NaF is mixed in 30 ml HCl (6 M) until dissolved.
145 Then 3 g of Ti₃AlC₂ are slowly added to avoid overheating (it is an exothermic reaction), and the
146 solution is left to stir at 40 °C for 48 h. After etching is complete, a dilute solution of NaOH was
147 added slowly until the pH of the solution reached 6, and then filtered and rinsed several times with
148 deionized water. The materials were then treated in 0.5 NaOH solution for 3 h at room temperature
149 and filtered. The materials were then washed with deionized water, being subsequently dried at
150 60–70 °C overnight.

151 2.3. Preparation of PW70/MXene as a new class of nanocomposites

152 Three different concentrations of Ti_3C_2 (0.1, 0.2 and 0.3 wt.%) are loaded into the PW70. Mass
153 values of PW70 (59.940 g) and MXene flakes (0.06 g) are measured separately using microbalance
154 (Explorer series, EX224, Ohaus) with measuring uncertainty of ± 0.0001 . These amounts are used
155 to prepare the PW70/MXene with a 0.1 wt.% of MXene. The proposed amount of paraffin wax
156 (59.940 g) is heated in a beaker (150 ml volume) using a hot plate (RCT BASIC, IKA). The
157 temperature of hot plate is adjusted to 100 °C. PW70 is left for 15 minutes on the hot plate for
158 melting and homogenization purpose. Afterwards, 0.06 g of MXene flakes is added to the beaker
159 and stirred by a magnetic stirrer at 500 rpm. During the mixing process, the beaker is covered with
160 an aluminum foil to prevent the formation of air bubbles during magnetic stirring process. Stirring
161 process is conducted for 2 h continuously. The resultant PW70/MXene nanocomposite is poured
162 into a vial (14 ml NEST) to mold at room temperature. Same protocols are repeated to prepare
163 nanocomposites with concentrations of 0.2 and 0.3 wt.% MXene respectively by adding 0.12 and
164 0.18 g of MXene flakes as fillers to the melted organic phase change material (PW70) of 59.880 g
165 and 59.820 g, respectively.

166 3. Characterization techniques

167 3.1. Structural Characterization of pure MXene (Ti_3C_2)

168 3.1.1. Powder x-ray diffraction (XRD) of pure MXene (Ti_3C_2)

169 X-ray diffraction is carried out in a Smart lab diffractometer (Rigaku, Tokyo, Japan) using Cu
170 $\text{K}\alpha$ radiation operating in reflection mode with Bragg-Brentano geometry to investigate the crystal
171 structure. Prior to the XRD characterization, all samples are dried in a heated oven for 18 h at
172 80 °C. The black powders are then ground and placed on a silica sample holder and pressed flat
173 with a glass slide.

174 3.1.2. Raman spectroscopy of pure MXene (Ti_3C_2)

175 Raman spectroscopy is carried out on a Horiba Lab Raman Spectrometer (Horiba, Minami-ku
176 Kyoto, Japan) with an EM-cooled Synapse camera. For taking spectra, a 100x, 0.90 NA
177 microscope objective is used. The dried powder is sandwiched between two glass microscope
178 slides which are pressed together to give flat MXene particles. One of these slides is then discarded
179 with the other slide placed flat under the diode laser (532 nm, 200 μW) for measurements.

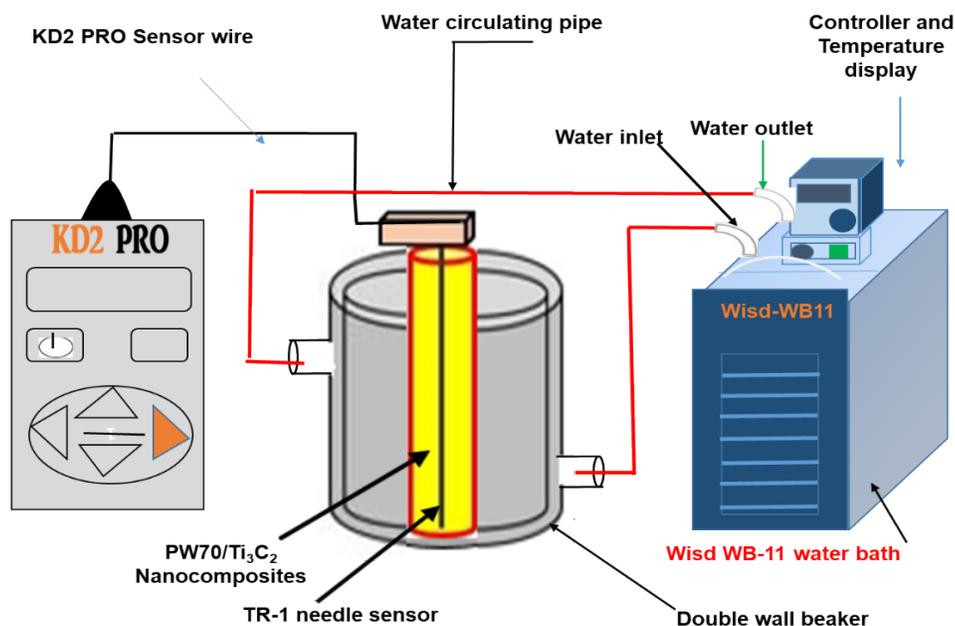
180 3.1.3. Scanning Electron Microscopy (SEM) of pure MXene (Ti_3C_2)

181 Scanning electron microscopy is done in a JEOL JSM-7800F (JEOL, Tokyo, Japan), using an
182 accelerating voltage of 10 kV and a working distance of 10 mm, which are used to study the
183 morphology and particle size. The dried powder samples are dry cast onto a carbon tape support,
184 which is placed onto a copper stub for analysis.

185 3.2. Thermal Conductivity Measurement

186 Thermal conductivity of the pure PW70 and PW70/MXene samples are measured using a KD2
187 Pro thermal properties analyzer (Decagon, USA, version 5). This probe works with the principle
188 of a transient hot wire technique. The TR-1 sensor (100 mm length, 2.4 mm diameter) is used for
189 the thermal conductivity measurement. A programmable digital water bath (Wisd, 11Lit, WB-11)
190 with temperature accuracy of ± 0.1 $^\circ\text{C}$ and heating power of 1 kW is used for controlling the
191 temperature within ± 1 $^\circ\text{C}$. Thermal conductivity of the nanocomposites (PW70/MXene) is
192 investigated at average temperature of ~ 25 $^\circ\text{C}$. The 14 ml vial consisting of the nanocomposite
193 and sensor is placed inside a double wall beaker (1000 ml volume) covered with Styrofoam to
194 prevent the exchange of heat between sample and atmosphere. Those are wrapped using an
195 aluminum foil to prevent the formation of air bubbles during the measurements. A schematic for
196 the measurement set up for thermal conductivity is shown in Figure 2. The deviation of the

197 measured temperature is considered around ± 0.5 °C. Each measurement is repeated three times to
198 get the precise results. Since the temperature uniformity is highly important for the KD2 Pro,
199 sample and sensor are allowed to achieve the temperature equilibrium for 30 minutes before any
200 measurement is made. Reading time of the measurements is raised to 10 minutes for each
201 measurement to ensure the accuracy of thermal conductivity results as well as reducing the error
202 points indicated by decagon instrument.



203

204

Figure 2. Thermal conductivity measurement set up for nanocomposites

205 3.3. Specific heat capacity (c_p) measurement

206 In this study, the specific heat capacity (c_p) measurements of PW70/MXene nanocomposite
207 samples are performed using a differential scanning calorimetry (DSC). DSC-1000/C (Linseis,
208 Germany) is a high resolution ($0.03\mu\text{W}$) instrument and the measurements are conducted using an
209 aluminum crucible of $40\mu\text{l}$. The temperature range for measurements is between $25\text{-}250$ °C with
210 the heating rate of 10 °C/min. The nanocomposite samples are tightly sealed in a regular aluminum
211 crucible with the capacity of $40\mu\text{l}$ under a nitrogen atmosphere with a flow rate of 20 ml/min. The

212 temperature repeatability and calorimetric precision are $\pm 0.1^\circ\text{C}$ and $\pm 1\%$, respectively.
213 Temperature and enthalpy calibrations for this DSC are carried out employing four standard
214 reference samples (Indium, Tin, Lead and Zinc) provided by the supplier. The experimentally
215 achieved data for specific heat capacity value of the pure paraffin wax is found in good agreement
216 with c_p value provided by the supplier. One uniform protocol for c_p measurements of the
217 PW70/MXene nanocomposites are adjusted in order to ensure the accuracy of results. All DSC
218 measurements are performed using the heating rate of $10^\circ\text{C}/\text{min}$ under nitrogen atmosphere.

219 3.4. Thermal stability test

220 Thermogravimetric analysis (TGA) of the PW70/MXene nanocomposites are conducted using
221 Perkin Elmer TGA 4000. A $180\ \mu\text{l}$ alumina crucible (withstanding temperature $\sim 1750^\circ\text{C}$) under
222 an ultra-high pure nitrogen gas flow of $19.8\ \text{ml}/\text{min}$ with the gas pressure of $2.6\ \text{bar}$ is selected to
223 examine the samples. The utilized heating rate is $10^\circ\text{C}/\text{min}$ for raising the temperature from 30 to
224 800°C . About $15\ \text{mg}$ of PW70/MXene nanocomposite sample is used for the decomposition
225 temperature measurement. The obtained data is analysed using Pyris Software.

226 3.5. Fourier transform infrared spectrum (FT-IR)

227 The Fourier Transform Infrared spectrum (FTIR) of the PW70/MXene nanocomposites is
228 determined using a Perkin Elmer Spectrum Two-UATR. Spectra is detected using the integrated
229 detector of MIR TGS ($15000 - 370\ \text{cm}^{-1}$). Scanning speed is $0.2\ \text{cm}/\text{s}$ within the optimum scan
230 range of $500 - 4000\ \text{cm}^{-1}$.

231 3.6. Ultraviolet-visible (UV-Vis) spectroscopy

232 Optical absorbance measurements are performed using Ultraviolet-visible (UV-Vis). Spectra is
233 acquired using Perkin Elmer Lambda 750. Data is collected at room temperature within the

234 wavelength from 800 to 200 nm. The adjusted scan speed is 266.75 nm/min with the 860 nm
235 monochromator.

236 3.7. HRTEM and elemental analysis of MXene nanoparticles

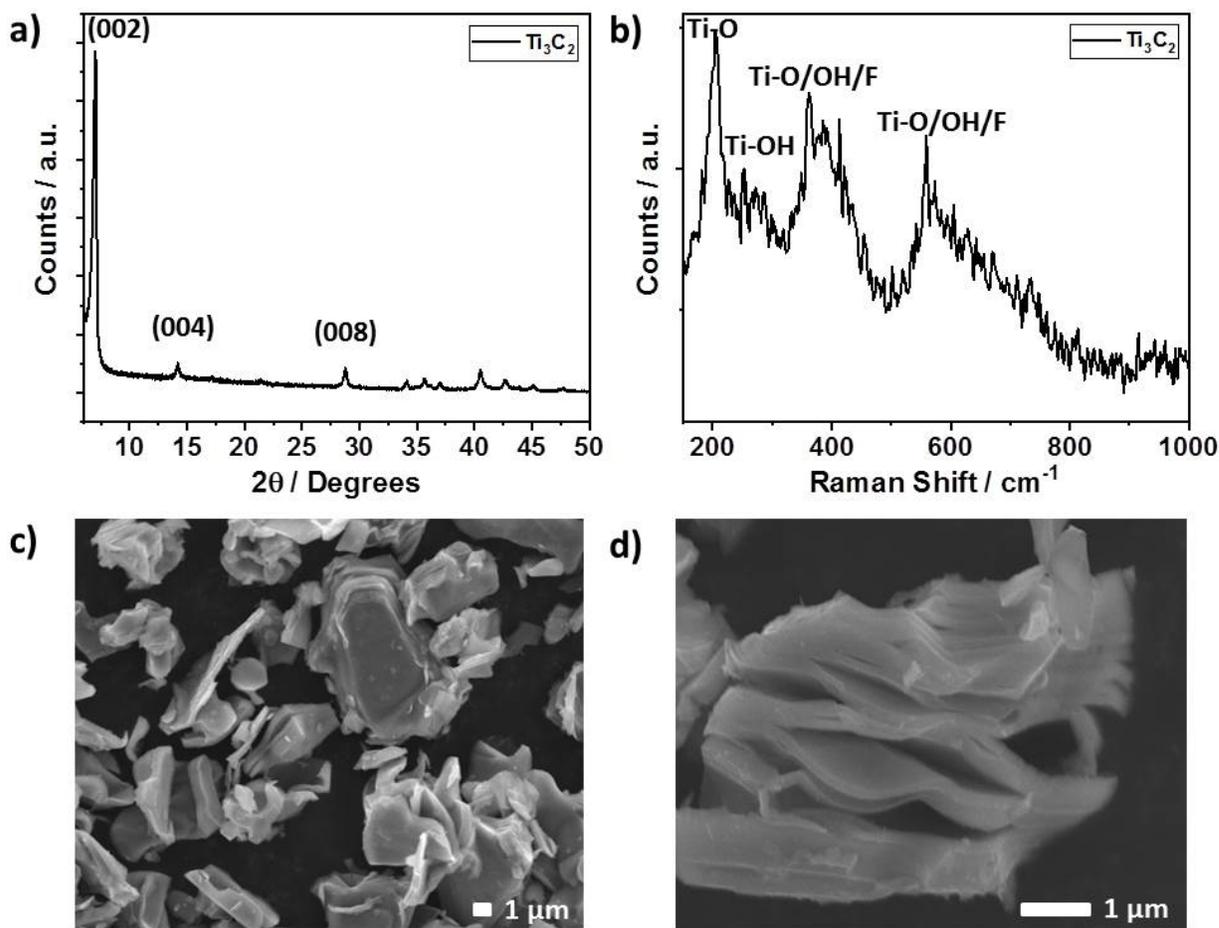
237 Morphology of the MXene flakes is investigated using HRTEM (JEOL JEM-ARM 200F)
238 imaging at an accelerating voltage of 200 kV. About 1 mg of MXene flake is added into ~ 4 ml of
239 ethanol in a vial. The mixture is remained on the hot plate at 60 °C and stirred using a magnetic
240 stirrer for half an hour. About 4 μ l of diluted sample is taken by micropipette and dripped onto the
241 carbon-coated copper grid. Energy dispersive X-ray spectroscopy (EDX, OXFORD
242 INSTRUMENT) is used to confirm the elemental map imaging and observation of spatial
243 distribution of all the elements. The spot analysis is carried out to detect the quantitative
244 distribution of elements in different portions of MXene.

245 4. Results and Discussion

246 4.1. Structural characterization of pure MXene (Ti_3C_2)

247 To confirm the structure of the as-synthesized MXene, XRD, Raman spectroscopy and SEM
248 studies are carried out and results are presented in Figure 3. The XRD data shows a sharp
249 diffraction peak at $7.1^\circ 2\theta$, corresponding to the (002) diffraction plane (Figure 3b). This gives an
250 interlayer spacing of 14.5 Å, consistent with previous reports of HCl-fluoride salt etched Ti_3C_2
251 [35]. Other (001) diffraction peaks such as (004) and (008) are also present, confirming the
252 successful synthesis of MXene (Ti_3C_2). Raman spectroscopy also confirms the formation of Ti_3C_2
253 with the Raman Spectra matching previous reports [36, 37]. The peaks at 207 cm^{-1} and 270 cm^{-1}
254 correspond to E_g vibrations resulting from the presence of Ti-O and Ti-OH surface groups,
255 respectively. The broad peaks which center around 390 cm^{-1} and 580 cm^{-1} are known to result from
256 vibrations relating to the presence of Ti-O, Ti-OH and Ti-F groups, all of which are common

257 surface terminations for MXene (Ti_3C_2) [36, 37]. SEM characterization reveals that the as-
258 synthesized MXene flakes are multilayered, with particle sizes ranging from 1-10 μm . The layered
259 nature of the MXene is also noticed, and matches previous reports [35].



260

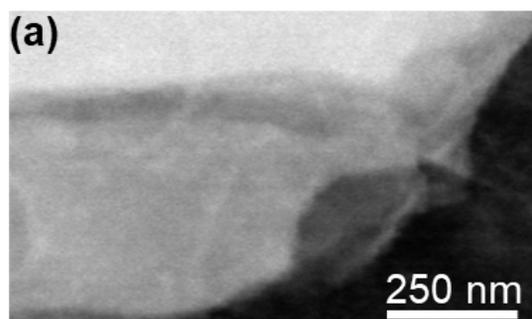
261 **Figure 3.** Structural characterization of the as-synthesized MXene (Ti_3C_2). a) XRD diffractogram b)
262 Raman spectrum c-d) SEM micrographs at different magnifications.

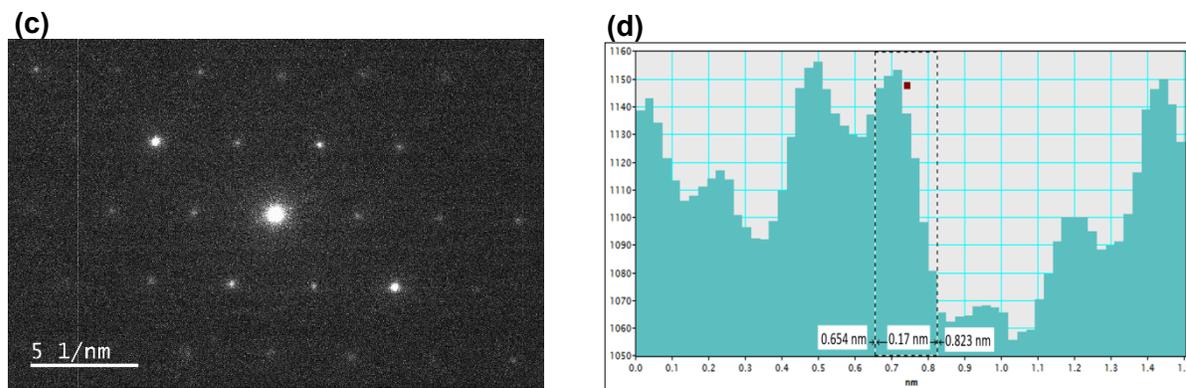
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264 4.2. Morphological characterization of MXene (Ti_3C_2)

265 An overview of emerging layered structure of MXene (Ti_3C_2) can be seen in Figure 4 where a
266 high-resolution TEM (HRTEM) image of few layered Ti_3C_2 flake is shown covering in a whole
267 carbon TEM grid. A low magnification image in Figure 4(a) depicting the electron-transparent

268 thin morphology and a few layer structures. Moreover, the thin MXene nanosheet is presented to
269 be flexible and foldable which is similar to those of graphene and two-dimensional MXenes [32,
270 38]. Its Fast Fourier Transform (FFT) (Figure 4.b) reveals a hexagonal-based crystal with chain-
271 like features of the MXene nanosheets [38]. The delaminated layers are found to be transparent to
272 the electron beam in TEM. Selected area electron diffraction (SAED) of delaminated MXene (in
273 Figure 4.c) image clearly shows that the atomic arrangement in the basal planes is identical to that
274 in the parent MAX phase [39]. These results provide further convincing evidence for the three-
275 dimensional to two-dimensional conversion of the material. Furthermore, it is clear from TEM
276 observation that there is no evidence for carbide amorphization. Ti_3C_2 sheets are found to be more
277 stable than graphene sheets under a 200 kV electron beam in the TEM [39, 40]. EDX results of the
278 same MXene flake confirm the presence of Ti, C with minor amounts of F, and O from the MXene
279 surface and very small amount of Al. Analysis of the intensity profile along the right dashed line
280 (Figure 4.d) gives the Ti-C bond length of $\sim 2.11 \text{ \AA}$ close to the expected value of 2.089 \AA [41].





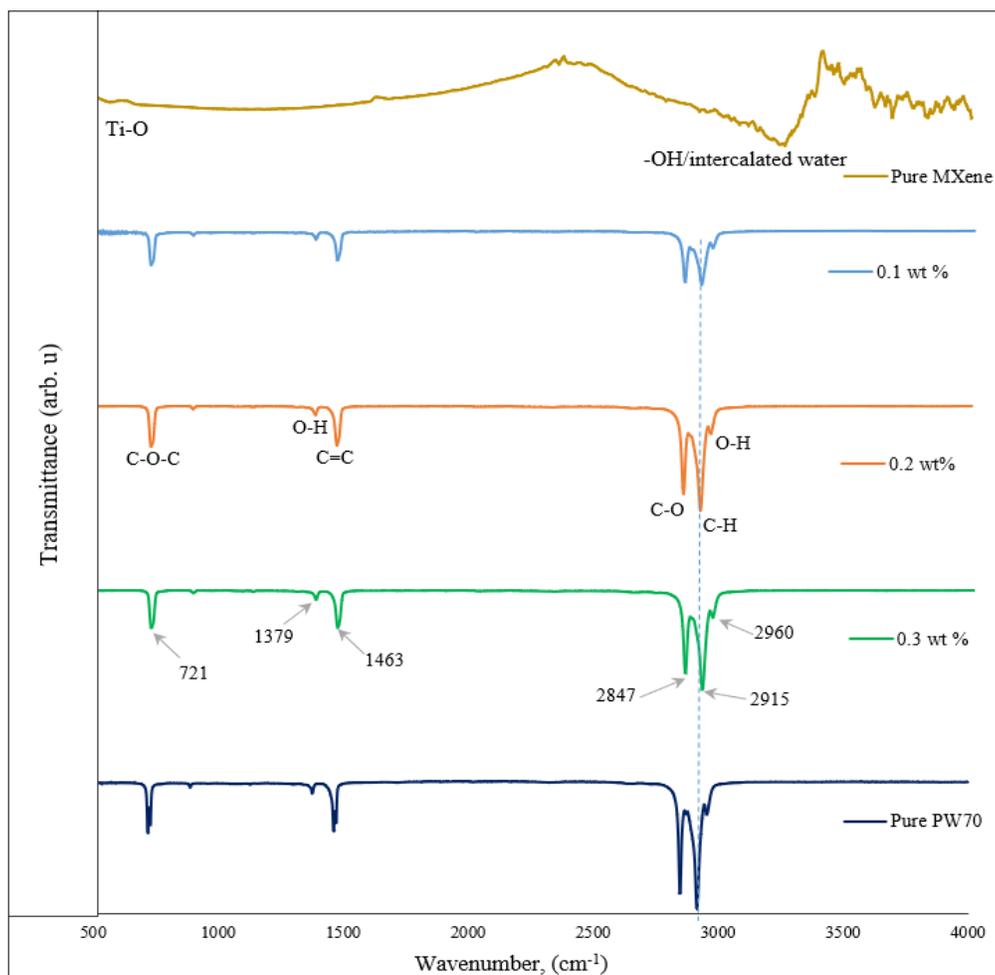
281 **Figure 4.** Top view of typical bright-field HRTEM images of the of a MXene (Ti_3C_2) flake. (a) low-
 282 magnification image of few layered MXene flake, (b) higher magnification HRTEM image of edge-on
 283 MXene (Ti_3C_2), (c) corresponding to the selected area electron diffraction pattern, (d) intensity analysis
 284 along the white dashed line shows Ti-C bond length of 2.11 Å.

285

286 4.3. Chemical structure characterization (FTIR) results

287 The chemical structures of the pure organic paraffin wax and the synthesized PW70/MXene
 288 nanocomposites are investigated using FTIR spectroscopy analysis. The achieved FTIR spectrum
 289 is presented in Figure 5. The FTIR spectrum for PW70/MXene nanocomposites and pure PW70 is
 290 shown separately. Inset of the Figure 5 presents the FTIR spectrum of pure MXene used in this
 291 work showing the peaks for Ti-O at $\sim 530 \text{ cm}^{-1}$ and prominent peak for O-H group at $\sim 3250 \text{ cm}^{-1}$
 292 from the intercalated water in the MXene. On the other hand, shifts in the characteristic peaks of
 293 PCM/MXene nanocomposites revealed the presence of C-O-C, O-H, C=C, C-O and C-H groups
 294 in the samples. The peak at $2924 - 2960 \text{ cm}^{-1}$ corresponds to the skeletal vibrations from C-O, C-
 295 H and O-H groups in the MXene structure. The presence of oxygen functionalities in the samples
 296 is detected in peaks at 1380 cm^{-1} due to O-H stretching vibrations. There is no O-H band detection
 297 in the wavenumber of 3400 cm^{-1} [42, 43]. Besides, it's observed peaks for C-C-C and C=C at ~ 721
 298 and $\sim 1466 \text{ cm}^{-1}$ respectively, is allocated due to the stretching deformation vibration of intercalated

299 water in the MXene sample [44]. Clearly, it is seen that the wavenumbers (cm^{-1}) of characteristic
300 peaks of all the composite samples with varying MXene loading mass fraction is similar. It reveals
301 that due to the increase of MXene in this low amount (wt.%) in the PCM, negligible chemical
302 changes occurred in the PCM. A few extra peaks have been observed at ~ 721 and ~ 1466 cm^{-1} in
303 the characteristic FTIR spectrum of the pure PW70 beyond the effect of MXene addition indicating
304 the purity of the PCM (PW70).

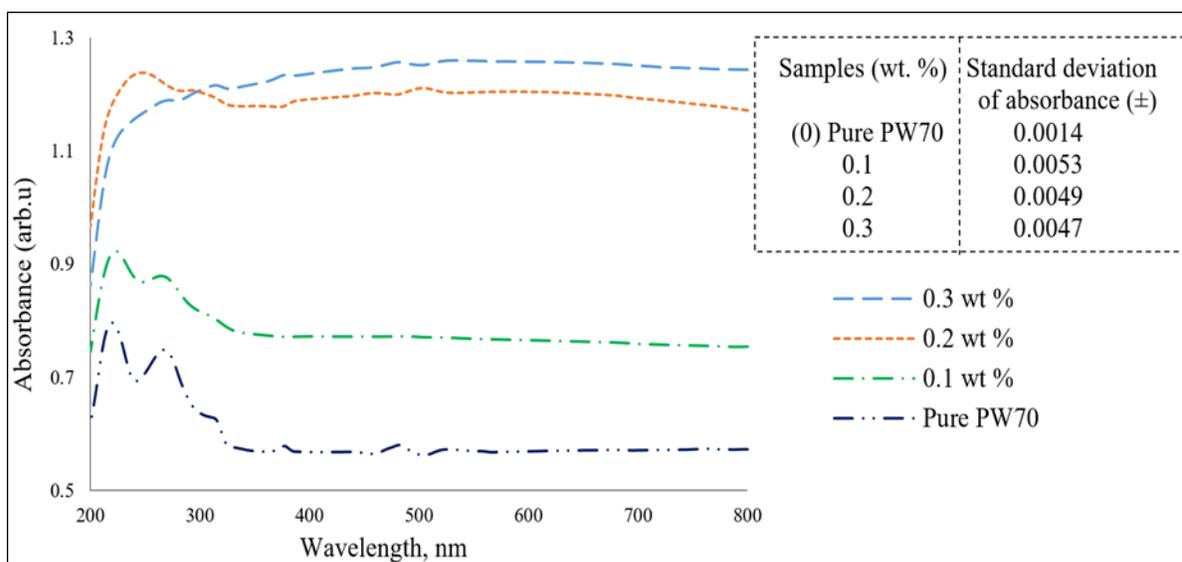


305
306 **Figure 5.** Full range (500 - 4000 cm^{-1}) FTIR spectra of pure MXene and PW70/ Ti_3C_2 nanocomposites
307 with varying MXene loading mass fraction and base material of pure PW70 PCM.
308

309 4.4. UV-Vis absorption spectrum characterization

310 The conversion of abundant solar energy into usable energy is an important point regarding
311 phase change materials which makes the possibility of storing energy during exposure to sunlight
312 and utilize at night. This kind of energy storing is called adsorptive energy storage that can be
313 measured through Ultraviolet-visible spectrometry (UV-Vis) in different wavelengths. UV-Vis
314 spectrometry of the PW70/MXene as a new class of nanocomposites is carried out by using
315 Lambda 750 Perkin Elmer. Absorbance measurements are achieved through optical resolution of
316 0.4 nm and over wavelength of 200-800 nm. The typical optical absorption spectra of pure organic
317 paraffin wax and PW70/MXene nanocomposite samples with different mass fractions of 0.1, 0.2
318 and 0.3 wt.% are measured and presented in Figure 6. Standard deviation of absorbance's of the
319 three repeated measurements of the pure PW70 and PW70/MXene nanocomposite samples with
320 varying wt.% of MXene are presented at the inset of the Figure 6. Pure PW70 spectrum has the
321 characteristic maximum at 200-300 nm region which is consistent with the literature data [45]. The
322 peak that appears at 230 nm in the UV-Vis spectrum of PW70 contributed to the degree of the
323 remaining conjugation (π - π^* transition), and the shoulder around 300 nm should be ascribed to
324 the n - π^* transition of carbonyl groups. The absorbance in the 270-350 nm regions is thought to be
325 caused by the conjugated aromatic domains [46]. Pure organic paraffin wax shows two absorption
326 peaks in the UV light range. However, almost no absorption peak is seen in the visible light range
327 (350-800 nm). The high background adsorption in the visible light range is induced by the
328 scattering of crystalline paraffin [46]. The UV-Vis spectra of PW70/MXene nanocomposites
329 exhibit typical absorption peaks at 220 and 290 nm which is consistent with the literature data [47].
330 It is seen that pure PW70 and PW70/MXene nanocomposite sample with 0.1 wt.% show similar
331 patterns with slight shifting of wavelength. The absorbance of the prominent peaks is found to be

332 increased about ~14% over pure PW70. On the other hand, PW70/MXene nanocomposite samples
 333 with 0.2 and 0.3 wt.% show the prominent peaks more consistent with MXene [46]. However,
 334 absorbance for these compositions are increased about ~38 and ~39% over pure organic paraffin
 335 wax. Thus, through the experimentally achieved data, it is evident that using MXene nanoparticles
 336 as additives to conventional PCMs can affect the intermolecular dynamics. The intermolecular
 337 interactions between atoms in surface area of MXene nanoparticles and C-H bonds of the organic
 338 phase change materials may cause to enhance the light-to-heat energy conversion [48]. The
 339 achieved enhancements in the absorbance of the prepared novel PW70/Ti₃C₂ nanocomposites
 340 might attribute to improve the efficiency of the solar systems by incorporating nanocomposites
 341 integrated with optical layers during direct exposure to sunlight



342
 343 **Figure 6.** UV-Vis spectra of pure PW70 and PW70/MXene nanocomposite samples. Inset: standard
 344 deviation of absorbance

345

346

347

348 4.5. Thermal conductivity of organic PCM dotted with MXene (Ti_3C_2)

349 Thermal conductivity is an important determinant for PCMs, influencing the rate of thermal
350 energy charging/discharging throughout the day/night. Table 3 illustrates experimentally achieved
351 data of thermal conductivity for nanocomposites with different loadings of MXene (Ti_3C_2) flakes
352 including the average value of thermal conductivity for three data points. The experimentally
353 obtained data reveals that the thermal conductivity of pure paraffin wax is found to be ~ 0.2 W/mK
354 which is in good agreement with literature value of 0.21 W/mK [49]. The thermal conductivity of
355 PCM/MXene nanocomposites increases by approximately 6.1, 12.4 and 16% for a loading
356 concentration of 0.1, 0.2 and 0.3 wt.%, respectively. The achieved average error point for three
357 different readings of pure PW70 is ± 0.007 . Low error points proved the accuracy of the thermal
358 conductivity measurements. It is observed that the addition of MXene into PW70 remarkably
359 improved the nanocomposite's thermal conductivity. The enhancement can be attributed to the
360 greater thermal conductivity of MXene at higher mass fractions in addition to the high surface
361 area-to-volume ratio of MXene compared to that of paraffin wax. Nevertheless, the thermal
362 resistance at the interface between MXene and PW70 could not be neglected. The previous studies
363 have proved that the thermal boundary resistance rises up remarkably as the nanoparticle
364 dimensionality escalates based on the results for two-dimensional materials included graphene
365 nano platelets [50]. Enhancement in thermal conductivity in the PW70/MXene nanocomposites is
366 more likely due to the high surface area and negligible thickness of MXene flakes. This reason
367 might contribute to create a percolation conducting network in the PCM matrix.

368

369

Table 3

Experimentally achieved data for thermal conductivity measurement using KD2 Pro Decagon

Nanocomposite (PW70/MXene)	Thermal Conductivity (W/mK)		Temperature (°C)	
	Average	Standard	Average	Standard
	value	deviation	value	deviation
Pure PW70	0.197	0.004	25.42	0.42
PW70/MXene (0.1 wt.%)	0.209	0.007	25.19	0.19
PW70/MXene (0.2 wt.%)	0.221	0.010	25.27	0.27
PW70/MXene (0.3 wt.%)	0.228	0.013	25.26	0.26

370

371 According to Yang et al. [50], 1 wt.% Nano-Si₃N₄ additives escalated the effective thermal
372 conductivity of paraffin wax by ~10% at room temperature. Li et al. [51] utilized MWCNTs in
373 synthesizing phase change microcapsules (MicroPCMs) compromised paraffin wax in order to
374 enhance the thermal conductivities of the microcapsules. The results indicated that thermal
375 conductivity of MicroPCMs/CNTs-SA with 1% of CNTs increased by ~19% compared with that
376 of MicroPCMs. In another study, Wang et al. [24] investigated the influence of adding MWCNTs
377 to the thermal properties of paraffin wax (PW). The PW/MWCNT composites showed no
378 lamination after settled for 96 h at liquid state. It was observed that for the PW/MWCNTs binary
379 nanocomposites with a weight fraction of 0.2, 0.5 and 2%, the thermal conductivity enhancement
380 reached ~11, 16 and 35% in solid state at 30 °C, respectively. The superior performance of our
381 synthesized MXene in raising the thermal conductivity of PCMs compared to above-mentioned
382 past results may be attributed to their particular planar structure and two-dimensional geometry.
383 As proposed by the molecular dynamics simulation data of Tang et al. [25], the improvement in

384 effective thermal conductivity is not only owing to the presence of the highly-conductive
385 nanofillers, but is also owing to the filler-induced alignment of the paraffin molecules that
386 inherently raises the thermal conductivity of the matrix.

387 4.6. Thermal energy storage characteristics of the pure paraffin and PW70/MXene 388 (Ti_3C_2)

389 For the purpose of assessment, calculation and design of thermal systems, specific heat capacity
390 is considered as one of the major parameters [52]. Thermal energy performance of the pure organic
391 paraffin wax and PW70/MXene nanocomposites are examined using DSC. Distinct solid to liquid
392 phase change peaks can be observed in Figure 7. Table 4 presents the all achieved data for c_p
393 measurements of pure organic PCM and PW70/MXene nanocomposites.

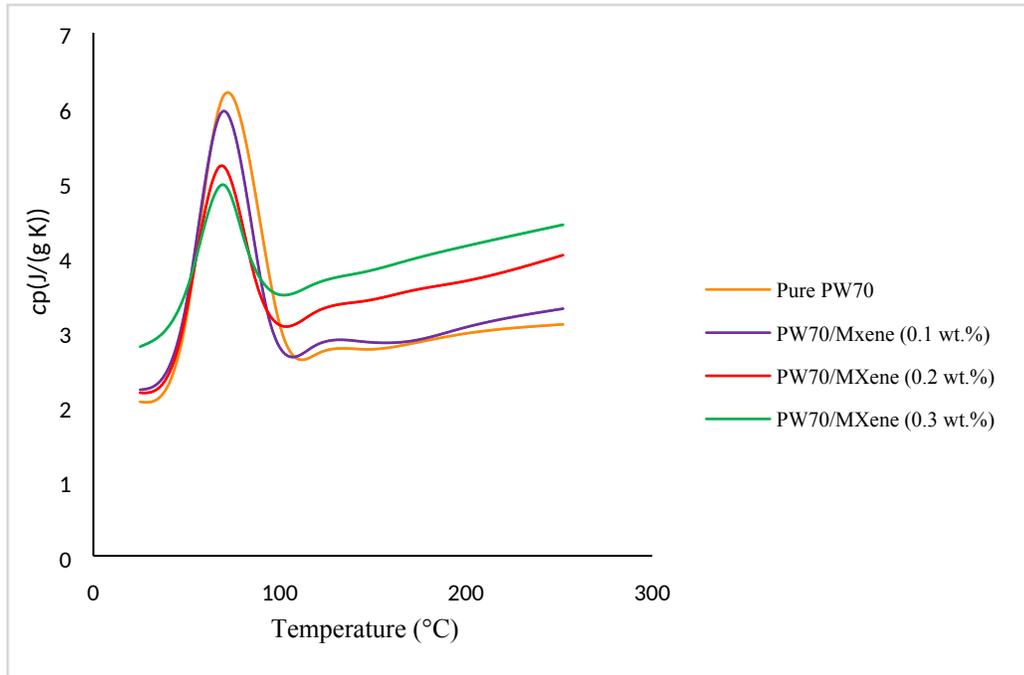
Table 4

Experimentally acquired data for c_p measurements of pure paraffin wax and PW70/MXene nanocomposites

Samples	Specific heat capacity (J/g K)	
	Temperature (°C)	
	Low temperature (~25 °C)	High temperature (~250 °C)
Pure paraffin wax	2.1	3.11
PW70/MXene (0.1wt.%)	2.2	3.31
PW70/MXene (0.2wt.%)	2.2	4.03
PW70/MXene (0.3wt.%)	2.8	4.43

394

395 The experimentally achieved DSC results of pure paraffin wax and PW70/MXene
396 nanocomposites for specific heat capacity measurements are presented in Figure 7.



397

398 **Figure 7.** DSC diagrams of pure PCM and PW70/MXene nanocomposites: c_p measurement of pure
 399 paraffin wax and PW70/MXene nanocomposites.

400 Figure 7 demonstrates the phase transformations for pure organic PCM and PW70/MXene
 401 nanocomposites. DSC diagrams indicate that c_p values for pure organic paraffin wax in the
 402 temperatures of ~ 25 and ~ 250 °C are 2.1 and 3.10 (J/g K), respectively (Table 4). Acquired
 403 experimentally data for c_p value (2.1 J/g K) of the pure paraffin wax at 25 °C is in good agreement
 404 with the provided data by supplier (2.2 J/g K) [34]. In a research work conducted by Christoph
 405 Hilgert et al. [53] under a comprehensive project (SFERA II project), the authors investigated the
 406 uncertainty of c_p measurements. They reported uncertainty of the c_p measurement in the value of
 407 1.3 % at mean operation temperatures between 25 and 60 °C. The uncertainty measurement in the
 408 present work is in the range of ~ 3 % which is based on the higher operating temperature (up to
 409 ~ 250 °C). In another research work, Bingham [54] reported uncertainty measurement of c_p value
 410 in the range of 4.5-6 %. As it is clear from the achieved data that the c_p value increases for pure
 411 PCM at elevated temperatures. The mechanism for increasing specific heat capacity at elevated

412 temperatures can be described due to vibrational, rotational and translation of molecules at higher
413 temperatures. Heating up a substance will cause to increase the average energy of molecules which
414 will lead to store more amount of energy [48]. Acquired data from Figure 7. illustrates that the
415 changes of c_p value for PW70/MXene (0.1 wt.%) in low temperature (~ 25 °C) is negligible ($c_p=2.2$
416 J/g K)). The increment of c_p value at high temperature (~ 250 °C) is $\sim 7\%$. The DSC results for
417 PW70/MXene (0.2 wt.%) indicate small changes of c_p value ($c_p= 2.2$ J/g K)) in low temperature
418 (25 °C). However, enhancement of $\sim 29.5\%$ was achieved in high temperature (~ 250 °C) for the
419 nanocomposite with 0.2 wt.% loading of MXene flakes. The achieved data for c_p values of
420 PW70/MXene (0.3 wt.%) reveals $\sim 34\%$ enhancement of c_p value at low temperature (25 °C) and
421 $\sim 43\%$ increment of c_p value at high temperature (~ 250 °C).

422 Literature reports express that specific heat capacity of nanocomposites can be increased up to
423 25% compared with the base substance due to the presence of nanoparticles [55]. Wang et al. [24]
424 applied a theoretical model to report that the c_p value of nanoparticles can be enhanced when the
425 size of nanoparticles is decreased. Less number of bonds in the surface area of the lattice of the
426 nanoparticles causes the atoms to be less constrained. Lower natural frequency and higher
427 amplitude of the vibrations of the atoms in the surface area can allow the bonds to act like spring.
428 This phenomenon can be the cause of creation of interface interaction of atoms in surface area of
429 two-dimensional materials (because of high surface area) with high surface energy atoms of the
430 organic pure paraffin (C-H bonds). The created interactions between atoms can be the reason of
431 the enhanced specific heat capacity of PW70/MXene nanocomposites. This mechanism is
432 applicable in high temperatures as the temperature elevates the vibration of atoms. Enhancement
433 of vibration will increase specific heat capacity, which is found experimentally in this study for
434 new class of nanocomposites (PW70/MXene) with different mass fraction loadings of MXene

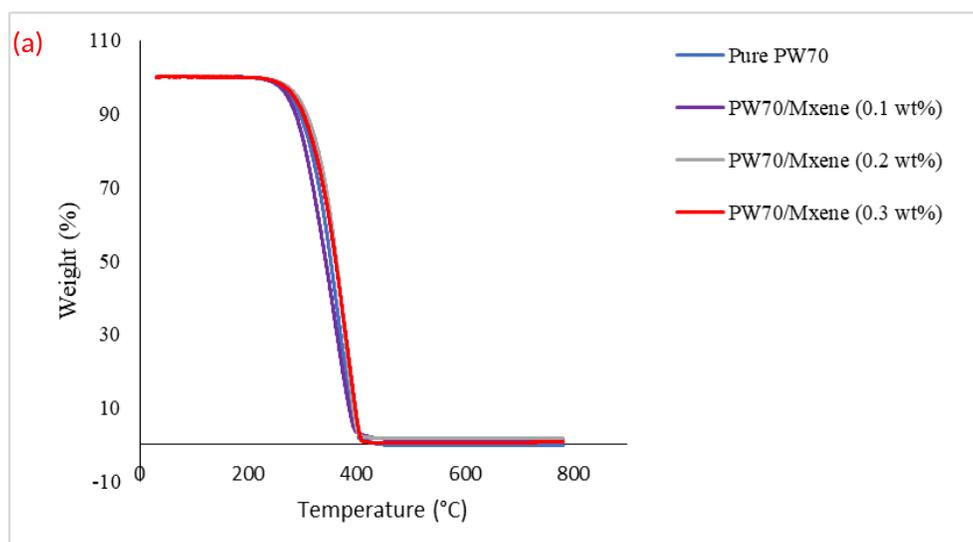
435 flakes. Due to the C-H bond in the structure of organic phase change materials and high surface
436 area of two-dimensional inorganic compound (MXene) it can be proposed that there is an
437 electrostatic interaction between the induced MXene in PW70. The aforementioned interface
438 interaction will cause to crystallize the particles of paraffin wax on the specific surface area of
439 MXene. Large specific surface area and specific surface energy will contribute to the enhancement
440 of the specific heat capacity. The acquired promising thermodynamic efficiencies at higher
441 operating temperatures may reduce the final cost of the thermal systems which are working in the
442 basis of thermal storage and thermal conductivity. An inexpensive method to reduce the cost of a
443 system is to enhance the thermophysical properties of organic/inorganic phase change materials
444 by doping them with minute concentration of nanoparticles. In a comprehensive research study
445 performed by Malik [56], the preliminary cost models for inorganic phase change materials (alkali-
446 salt eutectics) were investigated. In the above-mentioned study, the developed model for inorganic
447 phase change material induced with alumina nanoparticles revealed cost-effectiveness by
448 enhancement in the specific heat capacity value. The author demonstrated that a 30% enhancement
449 in specific heat capacity would offer a 15% saving in solar thermal power systems. The above-
450 mentioned model can be extended for organic phase change materials as well. Thus, based on the
451 developed model by Malik [56], the cost-effectiveness of the PW70/MXene nanocomposites for
452 solar thermal power systems might be expected according to the experimentally achieved
453 improvement specific heat capacity value. The required amount of thermal energy storage media
454 in concentrated solar power plants can be reduced significantly decrease with enhancement of the
455 specific heat capacity value, followed by a consequent reduction in the cost of electricity.

456

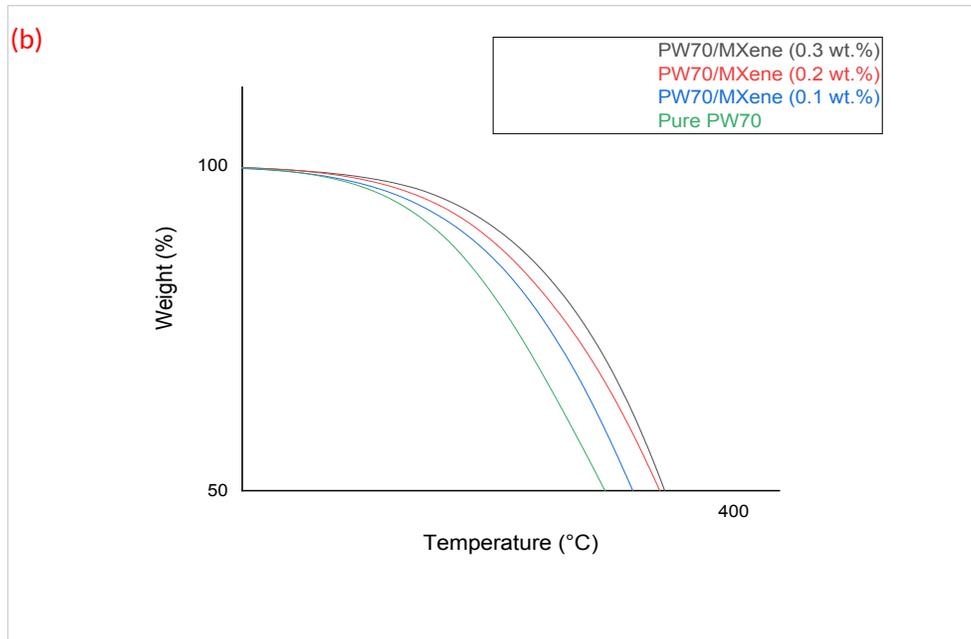
457

458 4.7. Thermal stability of pure PW70 and PW70/MXene nanocomposites

459 Thermal durability of the pure organic phase change material and PW70/MXene
460 nanocomposites has been evaluated by thermogravimetric analysis (TGA). Acquired TGA curves
461 are presented in Figure 8. TGA analysis for all samples is followed with a unique protocol and
462 same condition (mass value of ~14.5 mg and heating rate of 10 °C/min) to ensure the accuracy of
463 achieved results. It was observed that there is a slight shift in initial and final degradation
464 temperatures of the pure paraffin wax and PW70/MXene nanocomposites. The achieved data
465 reveals that the focal degradation temperature for pure organic phase change material is found to
466 be 362.5 °C. The presented data show that there is a slight increment in final degradation
467 temperature of nanocomposite induced with MXene flakes. The focal degradation temperature for
468 the nanocomposites of PW70/MXene with different loadings (0.1, 0.2 and 0.3 wt.%) is achieved
469 363.9, 380.9 and 384.0 °C. From figure 8. (b), it is clear that with the increased concentration of
470 Ti_3C_2 , the thermal stability of the nanocomposites is increased. The acquired data confirms that
471 with extending the loading amount of MXene nanoparticles, the thermal durability of the
472 PW70/MXene nanocomposites can be increased consequently.



473

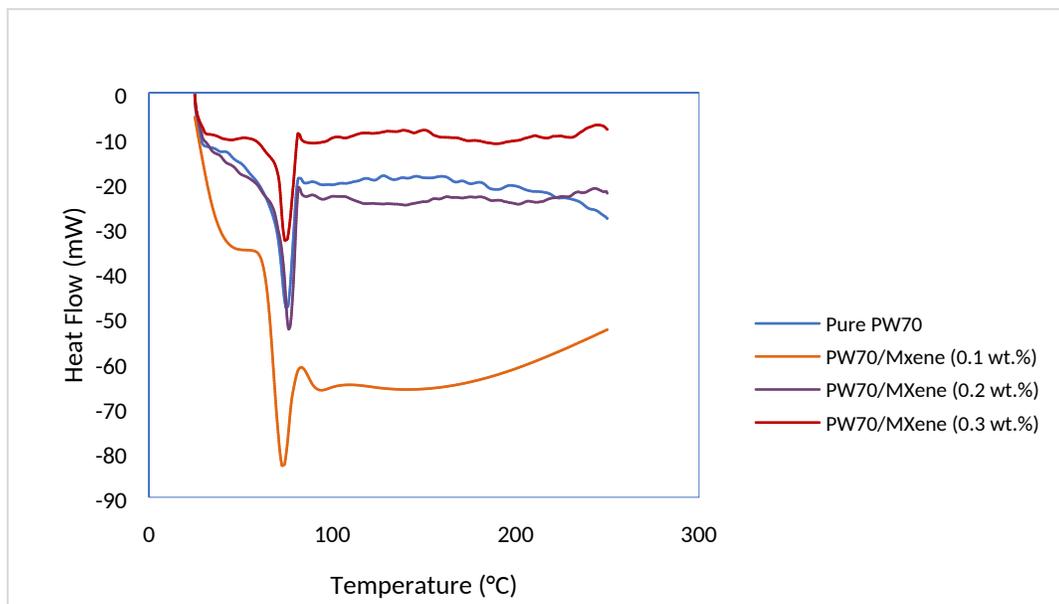


474 **Figure 8.** Thermogravimetric analysis of (a) pure organic PCM and PW70/Ti₃C₂ nanocomposites (b)
 475 enlarged area of the decomposition temperature region

476 4.8. Characterization of enthalpy and melting point of the PW70/Ti₃C₂

477 Enthalpy and melting point characterization of the synthesized PW70/MXene nanocomposites
 478 and pure PCM are performed using the DSC. Figure 9. presents the enthalpy results for pure
 479 paraffin wax and nanocomposites with different mass fractions of MXene flakes. The
 480 experimentally achieved data for melting point of pure PW70 is reported to be 69.8 °C which is in
 481 good agreement with the provided data by supplier of the pure PCM [34]. The uncertainty
 482 measurement in the value of the melting temperature is ~ 0.3%. It is observed that with adding
 483 MXene (Ti₃C₂) nanoparticles to pure paraffin wax, the melting point is increased slightly from
 484 69.8 to 71.7 °C for the highest concentration of MXene flakes (0.3 wt.%). The slight increment
 485 can be observed because of the interface interaction between high surface energy atoms of pure
 486 paraffin wax and high surface area of MXene nanoparticles. The increment of melting point
 487 confirms the explained mechanism of enhancement in the specific heat capacity. The achieved

488 data for melting point of PW70/MXene with loading of 0.1 and 0.2 wt.% are 70.7 and 71.4 °C. It
489 is observed from the DSC curves that the endothermic enthalpy value decreases with MXene flakes
490 as additive. This shows that the releasing energy in melting point occurs and heat transfer enhances
491 in this crucial point. The experimentally acquired data for enthalpy of pure PCM is found to be
492 110.68 J/g (endothermic) which is in agreement with the provided data by the supplier (107 kJ/kg)
493 [34]. The standard deviation for the enthalpy measurement is estimated to be ~ 3%. However, the
494 achieved data for enthalpy of PW70/MXene for the concentrations of 0.1, 0.2 and 0.3 wt.% are
495 93.86, 95.15 and 99.53 J/g, respectively. The results show that lower concentration of MXene
496 flakes (0.1 wt.%) has caused to less endothermic enthalpy. This performance shows that higher
497 concentration of MXene flakes causes to create stronger crystallinity which will lead to have more
498 thermal storage at this crucial point.



499
500 **Figure 9.** Enthalpy and melting point characterization of PW70/MXene nanocomposites and pure PCM.
501 (DSC curves)

502

503 **5. Conclusion**

504 In this paper, a new class of nanocomposite using MXene (Ti_3C_2) nanoparticles as additive to
505 a pure organic phase change material has been successfully synthesized. Structural analysis of the
506 synthesized MXene flakes has been investigated by using x-ray diffraction, Raman spectroscopy
507 and SEM characterizations. XRD and Raman spectroscopy analysis confirm the purity of the
508 synthesized MXene flakes. SEM characterization reveals that the synthesized MXene flakes are
509 multilayered, with particle sizes ranging from 1-10 μm . FITR results show good formation of the
510 synthesized nanocomposites. Thermal energy storage capability of the PW70/MXene and pure
511 paraffin wax has been measured at elevated temperatures considering decomposition temperature
512 as maximum operating temperature. It is revealed that the specific heat capacity of the
513 PW70/MXene nanocomposite with mass fraction of 0.3 wt.% enhanced by $\sim 43\%$ compared to the
514 pure PW70. Thermal conductivity of the PW70/MXene nanocomposite is found to have
515 enhancement of 16 % with mass fraction of 0.3 wt.%. The unique two-dimensional and well-
516 formed layers of planar structure of MXene nanomaterials leads to obtain high promising
517 thermophysical properties. The melting point of the synthesized nanocomposite is shown to be
518 slightly increased at elevated loadings of MXene nanomaterials. The melting point for pure
519 paraffin is found to be 69.8 $^\circ\text{C}$. Melting point of PW70/MXene with loading of 0.3 wt.% is
520 measured to be 71.7 $^\circ\text{C}$. The final degradation temperature of PW70/MXene nanocomposite in
521 volume fraction of 0.3 wt.% is found to be 384 $^\circ\text{C}$. Upon realizing the superior performance of the
522 new emerging nanomaterials (MXene) in serving as nanofillers for preparation of PCM
523 nanocomposites with enhanced thermal storage and thermal conductivity, the effects of the number
524 of layers, thickness and the size are of great interest for future studies. Thermal conductivity at
525 elevated temperature also need to be carried to investigate its variation with temperature.

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