

Graphene Contact Engineering in VP/MoS₂ Van der Waals Heterostructures for High Performance, Durable and Multispectral Photodetection

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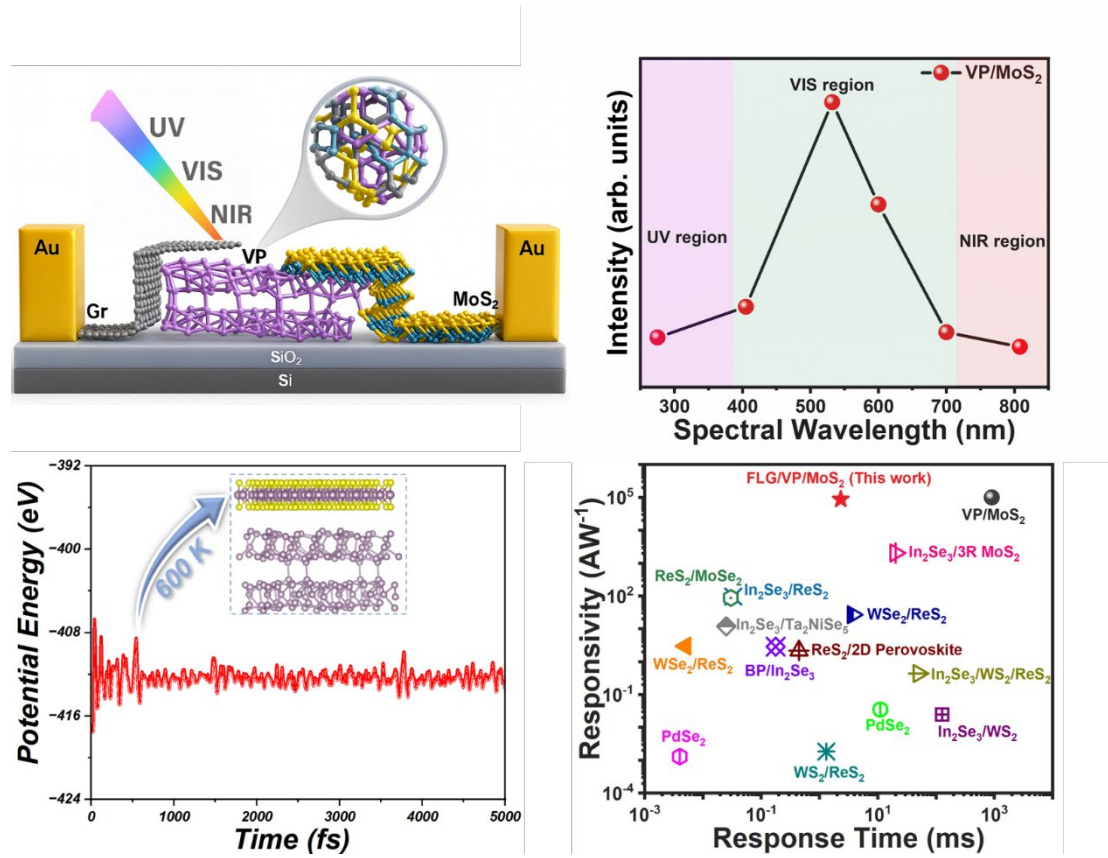
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Abstract

Efficient photocarrier separation and collection are essential for high-performance photodetectors. In van der Waals (vdWs) heterostructures, carefully engineering on the interfaces and built-in electric fields is an effective strategy to maximize device performance. Here, we present a unique photodetector based on a vertically stacked few-layer graphene/violet phosphorus/MoS₂ (FLG/VP/MoS₂) vdWs heterostructure, which exhibits significantly enhanced performance across a broad spectral range from ultraviolet (275 nm) to near Infrared (808 nm). The photodetector achieves a responsivity of up to 8.8×10^4 A/W with an external quantum efficiency of 1.81×10^7 values over five orders of magnitude higher than of bare VP/MoS₂ device (~ 0.52 A/W, 7.68%, respectively). It also exhibits a fast photoresponse (rise time ~ 2.3 ms) and notable polarization sensitivity (dichroic ratio ~ 1.075). Moreover, the device maintains stable performance over at least 100 on-off cycles, highlighting its excellent operational durability. First-principles (DFT) calculations further corroborate the strong interlayer coupling and structural stability of the device interface, providing theoretical support for the observed durability and photodetection mechanism. Collectively, this work positions the FLG/VP/MoS₂ vdWs heterostructure as a promising platform for high-performance, broadband, and polarization-sensitive photodetection in advanced optoelectronic applications.

Keywords: Graphene contact engineering; 2D materials; Violet phosphorus; Heterostructures; Photodetectors; Responsivity

1. Introduction

Photodetectors are central to modern optical sensing and communications; however, many traditional designs struggle to deliver the required sensitivity and speed across broad wavelength ranges [1, 2]. Conventional photodetectors which are exemplified by silicon-based devices are suffers from limited responsivity (R), slow response times, and a narrow range of detectable wavelengths [3]. These inherent limitations impede their use in advanced applications that demand simultaneous ultraviolet (UV) to near infrared (NIR) detection and rapid signal processing [4]. As a result, there is a strong impetus to develop new photodetector architectures that overcome these shortcomings, achieving higher sensitivity over multi-spectral bands while maintaining fast response and long-term stability [5].

One promising approach is to leverage emerging nanoscale materials, particularly two-dimensional (2D) materials and their van der Waals (vdWs) heterostructures which are important to engineer photodetectors with superior performance [6, 7]. Atomically thin 2D semiconductors such as graphene and transition metal dichalcogenides (TMDCs) such as MoS₂, WS₂, MoSe₂, WSe₂ etc. exhibit exceptional photoelectronic properties, including high carrier mobilities and tunable bandgaps, which can be tailored to enhance photodetection efficiency [7, 8]. By vertically stacking 2D materials without the constraints of lattice matching, vdWs heterostructures enable the combination of multiple functional materials in a single device, therefore which can enhance the photodetection sensitivity and also realize broaden photoresponse across multiple spectrums [9, 10]. For example, a group of researchers explored violet phosphorus

(VP)/MoS₂ vdWs heterostructure device shows high R of 10⁵ A/W, detectivity (D*) of 10¹³ Jones and an external quantum efficiency (EQE) of 10⁷% at the wavelength of 532 nm. They noticed that, their VP/MoS₂ vdWs heterostructure device's performance are better than their individual VP and MoS₂ photodetector device [11]. Another, a group of researchers demonstrated vdWs heterostructure device based on the VP/InSe which is showing broadband spectrum response from visible (VIS) to NIR including a high R of 182.8 A/W, an excellent D* of 10¹² Jones and an EQE of 11939% [12]. Recently, a combination of 2D materials such as TaSe₂/WS₂/TaSe₂ in the form of vdWs heterostructure enables excellent photoelectronic characteristics such as a high R of 292 mA/W, an outstanding D* of 10¹¹ Jones and large optical switching ratio of 10⁵ at the wavelength of 633 nm. In addition, demonstrated device shows excellent stability over 3 months [13]. Such examples underscore how 2D materials stacks can deliver detection capabilities far beyond those of any single bulk semiconductor, offering a clear pathway toward high-performance, multispectral photodetection [14, 15].

Among the family of 2D materials, VP, a lesser-known layered allotrope of phosphorus, has recently emerged as a highly promising alternative [16]. As the most stable phosphorus allotrope, VP exhibits significantly improved resistance to oxidation and degradation compared to black phosphorus (BP) [17]. At the same time, it retains the desirable anisotropic optical response of phosphorene [18]. VP has a unique tubular crystal structure that gives rise to strongly direction-dependent (polarization-sensitive) light matter interactions and a relatively large bandgap [19]. For example, bulk and few-layer of VP feature a direct bandgap of 1.8-2.5 eV (reaching ~2.5 eV in the monolayer

limit) and can support high hole mobilities up to $7000 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$ [20]. These properties indicate that VP can serve as an excellent 2D semiconductor for photodetectors, offering broad spectral absorption in the broadband spectrum range, polarization sensitivity, and robust air stability. On the other hand, MoS₂ is a well-explored 2D material that belongs to the TMDCs family and is widely used for optoelectronic devices [21-23]. Integrating VP and MoS₂ in a single device as a vertically vdWs structure can be expected to deliver not only broadband photodetection but also reliable performance under ambient conditions. Such a synergistic combination paves the way for developing next-generation photodetectors with enhanced efficiency and stability. Furthermore, device performance can be optimized by contact engineering, for instance by using a graphene electrode. Such contact engineering can reduce contact resistance and energy barriers at the interface, thereby facilitating more efficient charge injection and collection. For example, InSe/WSe₂ with graphene electrodes can enhance photodetection performance including a high R of 829.7 A/W, an excellent D* of upto 10^{14} Jones and rapid photoresponse of 10 μs [24]. In addition, another group of researchers demonstrated Gr/InSe/MoTe₂ vdWs heterostructure device which can enable to reduce dark current upto 10^{-14} A and quick photoresponse of 14 μs . These results attributed to the narrow bandgap of the graphene as compared to gold (Au), resulting in a nearly negligible Schottky barrier at the graphene-InSe interface [25]. Such an optimized interface enables enhanced photodetection performance by reducing recombination losses and boosting carrier transport. Building on this concept, the device architecture can be further extended to achieve broadband response and

multifunctional applications in advanced optoelectronic systems.

In this work, we present a photodetector based on a vertically stacked FLG/VP/MoS₂ vdWs heterostructure, designed to significantly enhance photodetection performance. In this device geometry, FLG was incorporated on the VP side, which not only improves the stability of VP but also reduces the interface barrier between VP and Au contacts. As a result of this engineering strategy, the device demonstrates an excellent broadband photodetection response from the 275 nm to 808 nm, achieving a high R of 8.8×10^4 A/W and an outstanding EQE of $1.81 \times 10^7\%$, which are more than five orders of magnitude higher than the bare VP/MoS₂ photodetector (0.52 A/W and 7.68%). In addition, the FLG/VP/MoS₂ photodetector exhibits a rapid photoresponse of ~ 2.3 ms along with a measurable polarization sensitivity of 1.075, underscoring its strong potential for multifunctional use in advanced photodetection and polarization-sensitive optoelectronic systems. Furthermore, DFT calculations confirm robust interlayer coupling and structural stability at the FLG/VP/MoS₂ interface, and molecular dynamics simulations at 600 K further verify its thermal durability. Moreover, the calculated bandgap of the VP/MoS₂ heterostructure is ~ 1 eV, indicating intrinsic sensitivity in the near-infrared region, which is consistent with our experimental findings. Overall, this research provides a promising pathway toward the development of next-generation high-performance photodetectors.

2. Experimental section

2.1 Device fabrication

Few-layer VP and MoS₂ flakes were obtained through mechanical exfoliation from

high-purity bulk crystals (~99.999%, 2D Semiconductors). For device fabrication of VP/MoS₂, VP flakes were first exfoliated onto pre-cleaned p-type SiO₂/Si substrates using the conventional scotch tape method. Prior to exfoliation, the SiO₂/Si substrates were cleaned by sequential sonication in acetone and isopropyl alcohol for 15 minutes each, and then dried with a gentle nitrogen stream. Subsequently, few-layer MoS₂ flakes were exfoliated onto a PDMS stamp and precisely transferred onto the VP flakes using a dry-transfer method, forming the VP/MoS₂ vdWs heterostructure. Graphene flakes were also prepared via the scotch tape method and transferred onto the VP side with the assistance of a PDMS stamp, completing the Gr/VP/MoS₂ heterostructure. Finally, maskless photolithography (TuoTuo Technologies, Suzhou Co., Ltd.) was applied to define the electrode regions, and Au (50 nm) contacts were deposited by thermal evaporation to finalize the device assembly.

2.2. characterization

The thickness of the Gr/VP/MoS₂ vdWs heterostructure was determined using an Atomic Force Microscope (Bruker Dimension Icon, Germany). Surface potential mapping was carried out on the same instrument operating in scanning probe mode. Raman and PL spectroscopy was performed with a Horiba LabRAM HR Evolution system (Japan) using a 532 nm excitation laser. Photoelectrical characterizations, including broadband photodetection, were conducted using a wavelength-tunable monochromatic light source (<https://bjnbet.com/>) coupled with a Keithley 2638B source meter. Polarization-resolved photocurrent measurements were performed on a customized setup from TuoTuo Technologies (Suzhou, China), integrated with a

Keithley 2612B source meter. The optical power of the incident light was continuously monitored with a Thorlabs PM101 power meter. All fabrication and measurement processes were carried out under ambient conditions at room temperature.

3. Results and discussion

3.1. Device fabrication and characterization

The schematic of the FLG/VP/MoS₂ vdWs heterostructure device is shown in **Fig. 1a**, with a magnified view of the material interfaces. **Fig. 1b** exhibits the optical image of the FLG/VP/MoS₂ vdWs heterostructure device. In this device structure, FLG, VP and MoS₂ obtained by mechanical exfoliation (details in experimental section). Atomic force microscopy (AFM) was used to examine the device surface morphology (**Fig. 1c**). Thickness profiles taken along the red (VP), green (MoS₂) and blue (FLG) lines yield layer thicknesses of approximately 8 nm (VP), 8 nm (MoS₂), and 7 nm (FLG) as summarized in Figure **S1a-c (Supporting Information)**. These measurements confirm the successful assembly of the vdWs heterostructure device of FLG/VP/MoS₂. Raman measurements were performed to assess material quality for the heterostructure device of VP/MoS₂. The Raman peaks for VP observed at 353, 373, and 471 cm⁻¹, which are attributed to the A_g and B_g modes as depicted in Figure **S2a (Supporting Information)**. For the MoS₂ nanoflakes, the Raman spectra observed at 383 and 408 cm⁻¹ correspond to the E_{2g}¹ and A_{1g} vibrational modes as illustrated in Figure **S2b (Supporting Information)**. Raman spectra of VP/MoS₂ are well consistent with previous reports [11, 26]. The coexistence of these Raman features in the overlapping region confirms the successful formation of the VP/MoS₂ heterostructure as shown in Figure **S2c**

(Supporting Information). To further explore the interfacial optoelectronic properties, photoluminescence (PL) measurements were carried out. As shown in Fig. 1d, the PL intensity of MoS₂ is significantly quenched in the heterostructure compared to that of bare MoS₂. This quenching effect evidences efficient charge transfer and electron-hole separation at the interface, which plays a crucial role in enhancing the performance of photodetectors.

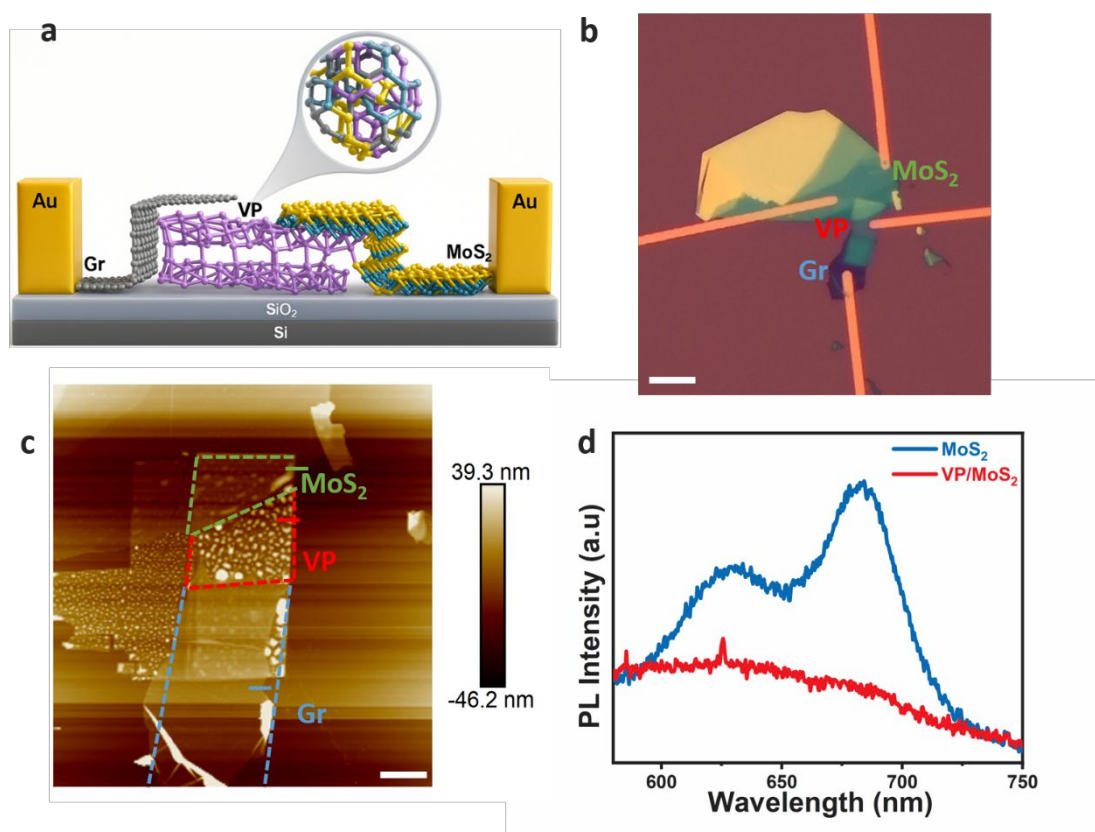


Fig. 1. Device fabrication and characterization. (a, b) Schematic illustration of FLG/VP/MoS₂ based vdWs heterostructure and optical image of the corresponding device, scale bar = 10 μ m. (c) AFM image of the device of FLG/VP/MoS₂ based vdWs heterostructure. PL spectra of MoS₂ and VP/MoS₂ vdWs heterostructure.

3.2 Photodetection characteristics

Before evaluating the photodetection characteristics of the FLG/VP/MoS₂ photodetector device, we first characterized the photoresponse of the bare VP/MoS₂ photodetector device. As shown in **Fig. 2a**, the VP/MoS₂ device exhibits a broadband spectral response from 275 nm to 808 nm, with a pronounced peak at 532 nm. Notably, clear photoresponse signals are observed at the UV (275 nm) and NIR (808 nm) edges **Figure S3a-d (Supporting Information)**. However, this bare VP/MoS₂ device produces only a very low photocurrent on the order of 10⁻¹¹ A. To improve the photocurrent, we integrated FLG on the VP side of the heterostructure to reduce the Schottky barrier and contact resistance, thereby enhancing the device performance [27]. For the FLG/VP/MoS₂ photodetector device, the output characteristics (**Fig. 2b**) reveal a pronounced photoconductive effect in this vdWs heterostructure of FLG/VP/MoS₂. This indicates that an external bias is required to operate the device, and it confirms that the device functions as a photodetector under 532 nm illumination. The FLG/VP/MoS₂ photodetector device can even operate under a very low incident power density of 0.19 mW/cm², demonstrating its high sensitivity (**Fig. 2c**). Furthermore, the FLG/VP/MoS₂ device retains a broadband spectral response similar to the bare device (**Fig. S4a,b Supporting Information**). **Fig. 2d** demonstrates the device's time-resolved photoresponse under chopped 532 nm laser illumination at various power levels, showing stable on/off switching behavior. When the light is turned on, the photocurrent rises rapidly to a steady value. Once the light is switched off, it quickly returns to the baseline, reflecting the device's high sensitivity and reversibility. Similarly, the bare

VP/MoS₂ photodetector device exhibits the same stable on/off photocurrent behavior (Fig. S5a,b Supporting Information). In addition, we evaluated the photodetector's performance using key figures of merit, specifically R and EQE. The R is defined as:

$$R = I_{ph}/(P_{in}A) \quad (1)$$

where I_{ph} denoted the photocurrent, calculated as the difference between the illuminated current ($I_{light}-I_{dark}$). P_{in} expresses as the incident light power density, while A corresponds to the effective active area of the photodetector.

$$EQE = \frac{I_{ph}/q}{P_{in}/h\nu} \quad (2)$$

In the above relation, where h denotes Planck's constant, q is the elementary charge, and ν represents the frequency corresponding to the wavelength of the incident light.

Using these relations, the FLG/VP/MoS₂ photodetector device achieved a maximum R 8.8×10^4 A/W and $1.81 \times 10^7\%$, which are more than five orders of magnitude higher than those of the VP/MoS₂ photodetector device (5.2×10^{-1} A/W and 7.68%, respectively) as shown in Fig. 2e,f. This exceptional improvement is attributed to the reduced Schottky barrier and greatly enhanced interfacial charge transfer afforded by the graphene contact.

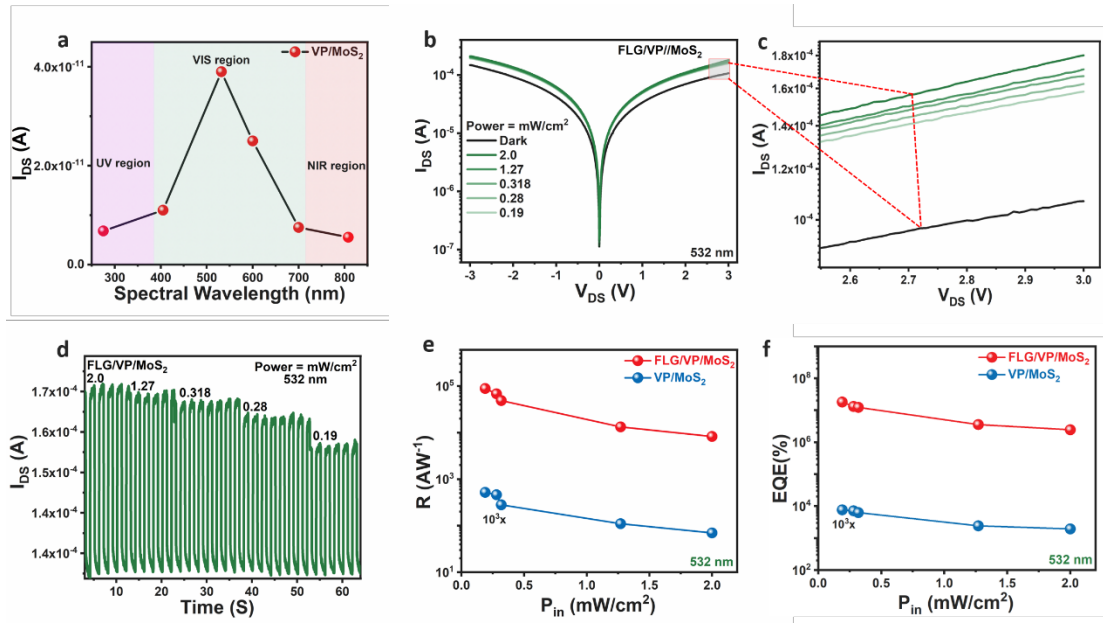


Fig. 2. Photodetection characterizations of FLG/VP/MoS₂ photodetector device. (a) Spectral photoresponse of the VP/MoS₂ photodetector device measured across 275 nm - 808 nm. (b) Typical output characteristics of the VP/MoS₂ photodetector device under dark and different illuminated conditions at the wavelength of 532 nm. (c) Enlarged view of the photocurrent evolution with increasing optical power density. (d) Time-resolved photoresponse of the VP/MoS₂ photodetector device under different optical power density at the wavelength of 532 nm. (e) Responsivity of VP/MoS₂ and Gr/VP/MoS₂ photodetector devices as a function of different incident optical power density. (f) External quantum efficiency comparison of VP/MoS₂ and Gr/VP/MoS₂ photodetector devices at 532 nm under different optical excitation power densities.

The photoresponse speed is a critical metric for evaluating photodetector performance. It is commonly quantified by the rise time and fall time, defined as the intervals for photocurrent to increase from 10% to 90% of its peak value and decay from 90% down

to 10%, respectively. In our devices, both the VP/MoS₂ vdWs heterostructure and the FLG/VP/MoS₂ vdWs heterostructure exhibit ultrafast response speeds. Specifically, the VP/MoS₂ photodetector device shows a rise time of ~5.1 ms and fall time of ~8.7 ms, while the FLG/VP/MoS₂ photodetector device achieves an even faster rise/fall time of ~2.3 ms/~7.5 ms (**Fig. 3a,b**). The significantly faster response of the FLG/VP/MoS₂ photodetector device is attributed to the integration of the FLG as a contact engineering, which improves charge transport and extraction. Graphene forms a clean, conductive contact that reduces interfacial trap states and lowers the Schottky barrier, allowing photogenerated carriers to be collected more rapidly. In essence, graphene's high mobility and Fermi-level alignment help sweep out photo-excited carriers quickly, minimizing charge trapping and recombination at the interfaces. This leads to a markedly quicker on/off switching of photocurrent in the FLG/VP/MoS₂ photodetector device compared to the VP/MoS₂ photodetector device alone. We further investigated the power-dependent photocurrent characteristics of both devices (**Fig. S6a,b Supporting Information**). The relationship between photocurrent (I_{ph}) and incident optical power (P_{in}) can be expressed as $I_{ph} \propto P_{in}^\alpha$ where the exponent α ($0 < \alpha \leq 1$) provides insight into carrier dynamics under varying illumination intensities. An ideal linear photodetector would have $\alpha \approx 1$ (meaning photocurrent scales linearly with light power), whereas $\alpha < 1$ indicates sublinear behaviour often caused by effects like carrier recombination, trap-assisted photogating, or saturation of trap states at higher illumination. For our VP/MoS₂ photodetector device, we obtained a fitted $\alpha \approx 0.70$ at 532 nm, indicating a moderately sublinear response. In contrast, the FLG/VP/MoS₂

device shows a much lower exponent, $\alpha \approx 0.13$ at 532 nm, revealing a strongly sublinear dependence on power.

To explore polarization sensitivity, we measured the photocurrent of the FLG/VP/MoS₂ photodetector device as a function of incident light polarization angle. (Fig. S7, **Supporting Information**) shows the schematic of our polarization-dependent photodetection setup. The polarization angle θ is defined as the angle between the incident light's polarization direction and the crystallographic a-axis of the photodetector [28]. Using a half-wave plate, θ was varied continuously from 0° to 360°, and we recorded the photocurrent at 15° increments to ensure comprehensive and reliable measurements. FLG/VP/MoS₂ photodetector device demonstrates clear angle-dependent photoresponse under 532 nm illumination (**Fig. 3c**), at a fixed incident power density of 2.0 mW/cm². The measured polarization-dependent photocurrent follows a sinusoidal pattern accurately described by the equation below [29]:

$$I_{ph}(\alpha) = I_a \cos^2(\alpha + \beta) + I_b \cos^2(\alpha + \beta) \sin^2(\alpha + \beta) + I_c \sin^2(\alpha + \beta) \quad (3)$$

In the above relation, I_a , I_b and I_c are related to the photocurrent, while α represents the polarization angle and β is a constant offset angle. **Fig. 3d** shows polar plots that further emphasize the polarization nature of photodetection at each angle. FLG/VP/MoS₂ photodetector device, we extracted a dichroic ratio of ~ 1.076 , meaning the photocurrent in the preferred polarization is about 7.6% higher than in the orthogonal polarization. This dichroic ratio is somewhat lower than those reported for photodetectors made from other highly anisotropic 2D materials or their heterostructures (in many cases, dichroic ratios of 2-3 or higher have been achieved with materials like black phosphorus, ReS₂,

etc.) [30-32]. The comparatively low polarization contrast in our device can be explained by the influence of MoS₂ and graphene. In the FLG/VP/MoS₂ photodetector device, the isotropic MoS₂ layer dilutes the anisotropic response of the VP. Essentially, some fraction of the photocurrent is generated or conducted through MoS₂ (and graphene contacts) in a polarization-insensitive manner, thereby reducing the overall contrast between different polarization angles. Additionally, the stacking and coupling of layers may slightly reduce the anisotropy of VP's absorption (for instance, if the VP's absorption peak is not extremely strong at 532 nm or if multi-layer VP is used, the intrinsic anisotropy might not be as pronounced). As a result, the device shows only a mild polarization dependence (dichroic ratio ~1.076), which is lower than devices using purely anisotropic crystals, but still demonstrates the ability to discern polarization to a degree. Despite the modest dichroic ratio, our FLG/VP/MoS₂ photodetector device achieves a well-balanced performance when compared with other state-of-the-art 2D material and their vdWs heterostructures photodetector devices at VIS spectrum (**Fig. 3e,f**) [33-50]. Notably, the FLG/VP/MoS₂ photodetector device simultaneously delivers high R, fast response speed and moderate dichroic ratio.

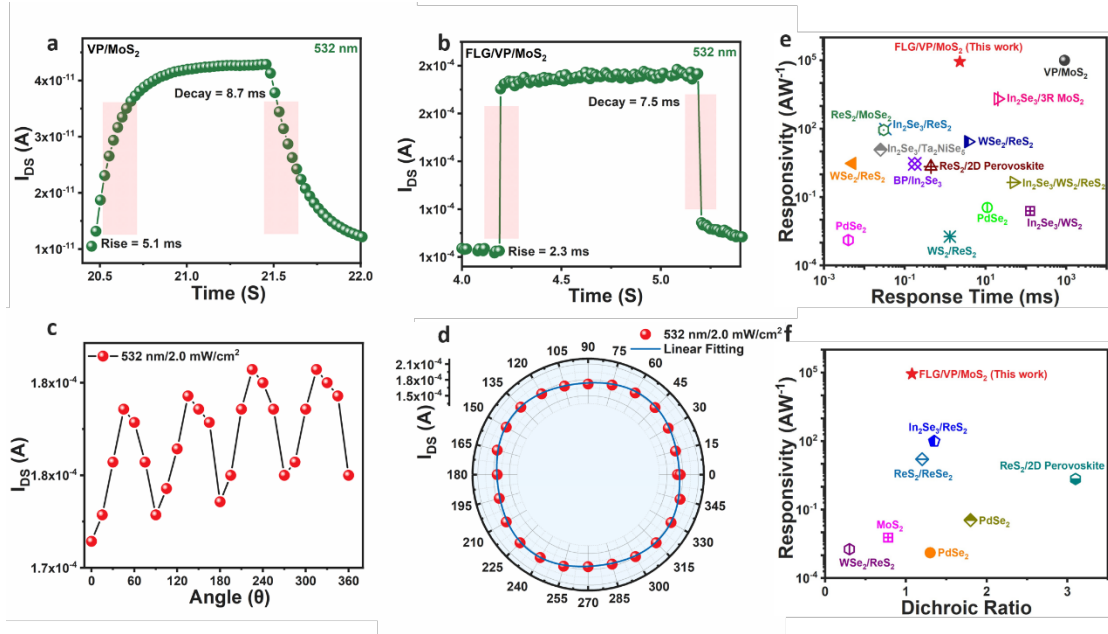


Fig. 3. Photoresponse and polarization characterization. (a, b) Time resolved photoresponse of the VP/MoS₂ and FLG/VP/MoS₂ photodetector devices at the wavelength of 532 nm, respectively, $V_{ds} = 1V$, $P_{in} = 2.0 \text{ mW/cm}^2$. (c, d) Photoresponse as a function of polarization angle (0° - 360°) at wavelength of the 532 nm for the FLG/VP/MoS₂ photodetector device, respectively, $V_{ds} = 1V$, $P_{in} = 2.0 \text{ mW/cm}^2$. (e, f) Performance comparison of R Vs. Response time and R Vs. Dichroic ratio FLG/VP/MoS₂ photodetector device with others 2D materials and their vdWs photodetectors technologies.

3.3 Durability and stability

To assess the theoretical durability of the FLG/VP/MoS₂ vdWs heterostructure, DFT calculation performed at 600 K for a 5 ps trajectory. The potential energy remained nearly constant throughout the simulation (**Fig. 4a**), fluctuating only minimally around an equilibrium value. This stable energy profile indicates excellent thermal stability of the heterostructure at elevated temperature, with no signs of structural degradation or

bond breaking over the simulation timeframe. Such robust atomic-scale stability suggests that the layered FLG/VP/MoS₂ system can withstand significant thermal stress, providing a strong foundation for reliable device operation. Importantly, the experimental results corroborate this theoretical prediction of stability. **Fig. 4b,d** present the time-resolved photoresponse of VP/MoS₂ and FLG/VP/MoS₂ photodetector devices, respectively, under 532 nm illumination ($V_{ds} = 1$ V) over 100 on-off cycles. In both devices, the photocurrent response remains strong and repeatable from the first cycle to the hundredth, with no appreciable decay in intensity or alteration of response speed. **Fig. 4c,e** zoom into the last five switching cycles of each device, highlighting that the photogenerated signal amplitude and timing are consistent even at the end of the 100-cycle test. Notably, no encapsulation was used in these measurements, the devices were exposed to ambient conditions yet they maintained robust performance throughout. This enduring stability under repetitive optical cycling can be attributed to the intrinsic resilience of the VP/MoS₂ vdWs heterostructure. In particular, integrating a FLG capping layer (FLG/VP/MoS₂) appears to further preserve device integrity by passivating the VP and providing an efficient charge transport pathway. The excellent agreement between the DFT calculations (predicting a thermally stable structure) and the sustained experimental photoresponse confirms that the VP/MoS₂ and FLG/VP/MoS₂ vdWs heterostructures are highly durable.

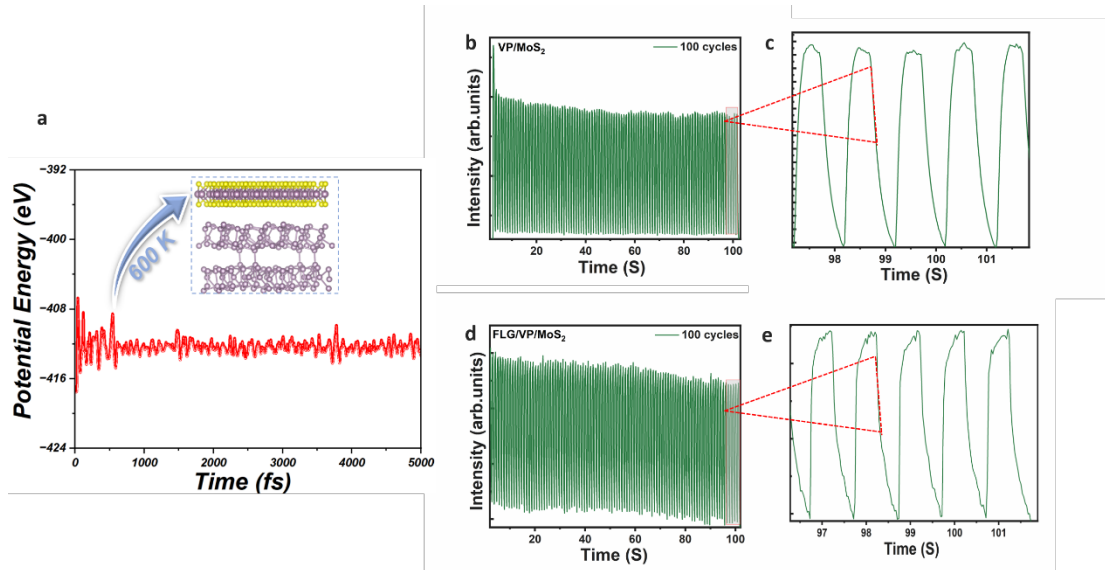


Fig. 4. Durability and stability. (a) Theoretical calculations of VP/MoS₂ vdWs heterostructure at 600 K for 5 ps, showing a stable potential energy trace over time and indicating excellent thermal stability of the structure. (b, d) Time-resolved photoresponse of the VP/MoS₂ and FLG/VP/MoS₂ photodetector device under 532 nm light ($V_{ds} = 1$ V) respectively for 100 on-off cycles, demonstrating stable and repeatable photocurrent with no significant degradation over the cycling period. (c, e) Enlarged view of the last five cycles from (b, d), highlighting consistent signal amplitude and switching behavior up to the 100th cycle.

3.4 Photodetection mechanism

We then systematically investigated the photodetection mechanism between the FLG, VP and MoS₂. The VP/MoS₂ vdWs heterostructure (**Fig. 5a**) was first analyzed via DFT. The hybrid HSE06 functional yields an electronic band structure for the VP/MoS₂ bilayer with a predicted bandgap of approximately 1.0 eV (**Fig. 5b**). This moderate gap lies in the NIR range, indicating that the coupled material system can absorb and respond to long wavelengths. Indeed, a 1 eV gap corresponds to a photon wavelength

of ~1240 nm, so the heterostructure device of VP/MoS₂ is expected to be sensitive well into the NIR which is consistent with our experimental observation of photodetection up to 808 nm (**Fig. 2a**). The DFT band structure also suggests a type-II band alignment. The valence band maximum (VBM) and conduction band minimum (CBM) reside predominantly on different layers (VP and MoS₂, respectively).

To further elucidate the electronic contribution of individual atoms, the projected density of states (PDOS) was calculated (**Fig. S8, Supporting Information**). The PDOS shows that the states VBM are dominated by P orbitals, whereas the conduction CBM primarily originates from Mo d-orbitals with partial contributions from S. This orbital-resolved picture corroborates the type-II alignment deduced from the band structure. For example, holes preferentially localize in VP while electrons are stabilized in MoS₂. Such orbital distribution not only confirms the staggered band alignment but also highlights the efficient charge-transfer pathway across the interface, which is essential for high-performance photodetection.

To validate the predicted band alignment and its role in photodetection, we employed Kelvin probe force microscopy (KPFM) to map the surface potential of the FLG/VP/MoS₂ vdWs heterostructure. As shown in **Fig. 5c**, a line scan along the white dashed path sequentially traverses Gr-VP-MoS₂, enabling extraction of local surface-potential variations across each interface (**Fig. 5c**). The contact potential difference (CPD) between the probe tip and the sample is mathematically expressed as follows:

$$V_{\text{CPD}} = (W_{\text{tip}} - W_{\text{sample}})/e \quad (4)$$

where e represents the elementary charge, W_{tip} denotes the work function of the KPFM

tip, and W_{sample} corresponds to the work function of the sample. Using this relation, KPFM measurements reveal a contact potential difference (V_{CPD}) of approximately 264 mV at the VP/MoS₂ vdWs heterostructure interface (**Fig. 5d**). Based on the KPFM calculations, the energy band diagram of Gr/VP/MoS₂ vdWs heterostructure before contact and after contact were plotted as shown in **Fig. 5e,f.** respectively. According to the band alignment, upon illumination, electron-hole pairs generated in both VP and MoS₂ are separated by the built-in field at the VP-MoS₂ heterojunction. Electrons transfer from VP into the lower-lying MoS₂ conduction band and drift through the channel toward the Au contact, while holes remain in VP; a fraction becomes trapped at VP/MoS₂ or interfacial states. These trapped holes act as a local positive gate on MoS₂ increasing the electron density and amplifying the channel current. The combination of type-II bands offsets, cooperative interfacial fields at VP/MoS₂ and MoS₂/Au that assist carrier separation and extraction. The theoretical band structure analysis (**Fig. 5b**) and the KPFM/band-alignment measurements (**Fig. 5c-f**) thus converge on a consistent picture that the VP/MoS₂ vdW heterostructure forms a type-II heterojunction with an ~ 1 eV gap, whose internal electric field and strong interfacial trapping enable efficient separation of electron-hole pairs and allow the device to detect NIR spectrum.

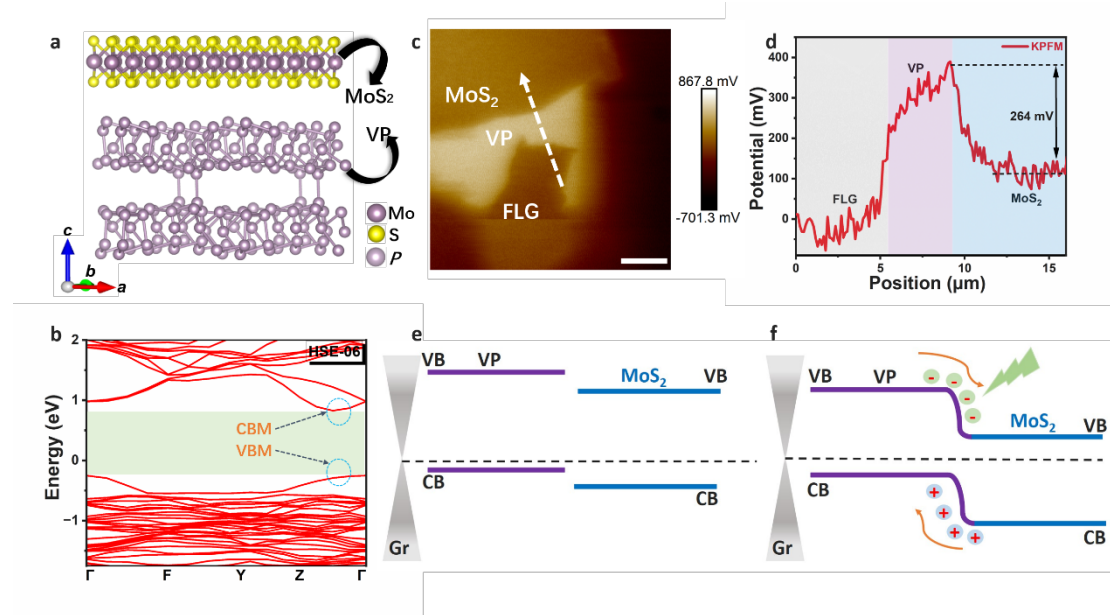


Fig. 5. Photodetection mechanism. (a) Atomistic structure of the vertically stacked VP/MoS₂ vdWs heterostructure (side view). Molybdenum (Mo) and sulfur (S) atoms of the MoS₂ layer and phosphorus (P) atoms of the VP layer are coloured to highlight the layered configuration. (b) Electronic band structure of the few-layer VP/MoS₂ bilayer calculated with the HSE06 hybrid functional. (c) KPFM surface potential map across the FLG/VP/MoS₂ vdWs heterostructure device, scale bar = 5 μm. (d) Line profile of the surface potential (contact potential difference, CPD) corresponding to the dashed arrow in (c), quantifying a ~264 mV potential step at the VP/MoS₂ junction. (e, f) Schematic energy band diagram of the FLG, VP, and MoS₂ before and after contact of illumination, illustrating the type-II band alignment.

4. Conclusions

In summary, a vertically stacked FLG/VP/MoS₂ vdWs heterostructure photodetector was fabricated, exhibiting a markedly enhanced broadband photoresponse. The device covers a spectral response range from 275 nm to 808 nm, with a high R of 8.8×10^4 A/W and an EQE of $\sim 1.81 \times 10^7\%$, exceeding the bare device of VP/MoS₂ vdWs

heterostructure photodetector device by more than five orders of magnitude. It also shows a rapid photoresponse with a rise time of 2.3 ms, a good dichroic ratio of 1.075, and stable operation over at least 100 on-off cycles. The superior performance is attributed to the graphene layer, which stabilizes VP, enhances charge transport, and reduces interfacial barriers at the FLG/VP and VP/MoS₂ junctions. Furthermore, DFT calculations confirm strong interlayer coupling, structural stability and the thermal robustness of the VP/MoS₂ vdWs heterostructure. The calculated bandgap of ~1 eV for the VP/MoS₂ interface suggests intrinsic sensitivity in the NIR region, in excellent agreement with our experimental findings. Collectively, these results establish the FLG/VP/MoS₂ vdWs heterostructure device as a robust and durable platform for high-performance, broadband, and polarization-sensitive photodetection.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

Acknowledgments

W.A and U.Y are equally contributed to this work. Y. I acknowledge the financial support from National Natural Science Foundation of China (NSFC) grant number W2432040, Guangdong Basic and Applied Basic Research Foundation 2023ZT10X010, 2024A1515010179, and Shenzhen Science and Technology Program 20231115150611001.

Appendix A. Supplementary material

Supplementary data to this article can be found online at

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