

Coordination between Electron Transfer and Molecule Diffusion through Bioinspired Amorphous Titania Nanoshell for Photocatalytic Nicotinamide Cofactor Regeneration

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4 **Coordination between Electron Transfer and Molecule**
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6 **Diffusion through Bioinspired Amorphous Titania**
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8 **Nanoshell for Photocatalytic Nicotinamide Cofactor**
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10 **Regeneration**
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Abstract

In-depth understanding and rational manipulation of the electron transfer process and molecule diffusion process are critical to promote the overall photocatalytic efficiency. In our study, core@shell photocatalysts that embody graphitic carbon nitride (GCN) core and amorphous titania (a-TiO₂) nanoshell are prepared to elucidate and coordinate the electron transfer and molecule diffusion for the regeneration of nicotinamide adenine dinucleotide (NADH) with [Cp*Rh(bpy)H₂O]²⁺ as the redox mediator. The GCN core absorbs visible light to generate electron-hole pairs, whereas the a-TiO₂ nanoshell facilitates the transfer of photo-generated electrons from GCN to the a-TiO₂ surface for NADH regeneration, which also enables the diffusion of electron donor molecules (TEOA) from the a-TiO₂ surface to GCN for consuming the holes left on GCN. The transfer of photo-generated electrons and the diffusion of electron donor molecules are coordinated by finely tuning the thickness of a-TiO₂ nanoshell. Under the optimized nanoshell thickness of ~2.1 nm, the GCN@a-TiO₂ photocatalyst exhibits the highest NADH regeneration yield of 82.1% after 10-min reaction under LED light (405 nm), over 200% higher than GCN photocatalyst. Combined with the highly controllable and mild features of the bioinspired mineralization method, our study may offer a facile and generic strategy to design high-performance photocatalysts through rational coordination of different substances/species transport processes.

Keyword: Coordination; Electron Transfer; Molecule Diffusion; Amorphous Titania Nanoshell; Bioinspired mineralization; Photocatalytic NADH Regeneration

Introduction

Coordinating the transfer/diffusion processes of multiple substances, e.g., electrons, ions, molecules, is critical for the efficient energy conversion in nature. Understanding and exploring the coordinated mechanism would help to elevate the energy conversion efficiency and sustain the development of modern society. Photocatalysis is a green energy conversion process, which can directly convert solar energy into fuels and valuable chemicals.¹⁻⁴ A photocatalytic reaction commonly includes three primary processes: photo-generated charge carrier transfer, reactant molecule diffusion, as well as reaction between carriers and molecules.⁵⁻⁷ Individual intensification of each process and coordinated optimization of all three processes are both essential for enhancing the overall process efficiency.

The intensification of photo-generated charge transfer is frequently used to promote photocatalytic performance, which can be realized by engineering the band structures, physical structures or hetero-structures of photocatalyst.⁸⁻¹² Amongst, core@shell hetero-structures have sparked numerous interests because of the flexible and independent regulation of crystal faces/forms/materials of the core and the shell.¹³⁻¹⁶ In this regard, tremendous efforts have been devoted to prepare core@shell hetero-structured photocatalysts, including transition metal oxyhydroxides@SrTiO₃, CoO_x@Ta₃N₅, MOFs@COFs, CdS@TiO₂, Au@TiO₂, *etc.*^{14, 17-20} As for a core@shell hetero-structured photocatalyst, the core material generally acts as a light harvester to induce electron-hole separation, while the shell material facilitates the electron transfer from the core material surface.²¹⁻²³ The incorporation of shell material can

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4 effectively inhibit electron-hole recombination and elevate photocatalytic conversion
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6 efficiency up to 99% for hydrogenation reaction.¹⁸ It is also reported that shell
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8 thickness is closely related electron transfer behavior, where a thicker shell with intact
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10 structure and transparent feature can lead to higher interfacial electron transfer rate.²⁰
11
12 Albeit significant progresses have been achieved in designing core@shell
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14 photocatalysts to facilitate the charge transfer, the coordination between charge
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16 transfer and molecule diffusion was rarely reported, which also seriously restricted the
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18 photocatalytic performance since the shell would frequently inhibit the diffusion of
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20 reactant molecules. In theory, a thicker shell is expected to cause decreased diffusion
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22 rate of electron donor molecules, which further results in lower photocatalytic
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24 efficiency. Then, the shell thickness of a core@shell hetero-structured photocatalyst
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26 shows a “trade-off” effect in balancing the two processes of photo-generated electron
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28 transfer and electron donor molecule diffusion, which may determine the final
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30 photocatalytic efficiency. Currently, rare investigations concern such a topic probably
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32 due to the difficulty in acquiring intact, porous and conductive shell on a core
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34 material. Therefore, developing a method to prepare core@shell photocatalyst with
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36 intact, porous and conductive shell would construct a platform to coordinate
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38 photo-generated electron transfer and electron donor molecule diffusion, and finally
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40 promote the overall photocatalytic efficiency.
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53 Herein, for the first time, a bioinspired mineralization method is developed to form
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55 an intact, porous and conductive of amorphous titania (α -TiO₂) nanoshell on graphitic
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57 carbon nitride (GCN) core for the preparation of GCN@ α -TiO₂ core@shell
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4 photocatalyst.²⁴ GCN, a metal-free semiconductor that has been successfully used for
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6 NADH regeneration, was chosen as the model core to harvest visible light and
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8 generate electron-hole pairs.²⁵ The a-TiO₂ nanoshell facilitates the transfer of
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10 photo-generated electrons from GCN to the a-TiO₂ surface for the regeneration of
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12 nicotinamide adenine dinucleotide (NADH) with [Cp*Rh(bpy)H₂O]²⁺, and at the
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14 same time enables the diffusion of electron donor molecules (TEOA) from the a-TiO₂
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16 surface to GCN for consuming the holes left on GCN.²⁶ The thickness of the a-TiO₂
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18 nanoshell could be controlled through changing the mineralization-inducer
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20 concentration and the mineralization cycles, which could optimally coordinate the
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22 photo-generated electron transfer process and electron donor molecule diffusion to
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24 pursue high photocatalytic efficiency. Our GCN@a-TiO₂ core@shell photocatalyst
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26 with controlled nanoshell thickness could establish a platform for understanding the
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28 coordination mechanism between molecule diffusion and electron transfer. When
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30 coupling the photocatalytic NADH regeneration with enzyme catalysis, the nanoshell
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32 could effectively isolate photo-generated holes and enzyme, which remarkably
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34 improved the compatibility between photocatalyst and enzyme. Moreover, the
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36 bioinspired mineralization method is performed under ambient conditions, e.g., room
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38 temperature, aqueous solution and neutral pH value,²⁷ which is applicable for
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40 producing a broad range of core@shell photocatalysts based on different core
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42 materials.
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55 **Experimental section**

56 **Materials**

Melamine, triethanolamine (TEOA) and pentamethylcyclopentadienyl rhodium (III) chloridedimer ($(\text{Cp}^*\text{RhCl}_2)_2$) were purchased from Shanghai Aladdin Co., Ltd. β -nicotinamide adenine dinucleotide phosphate sodium salt hydrate (NAD^+), protamine sulfate salt from salmon, titanium (IV) bis (ammonium lactato) dihydroxide solution (Ti-BALDH, 50 wt% aqueous solution) and yeast alcohol dehydrogenase (YADH) from *Saccharomyces cerevisiae* were purchased from Sigma-Aldrich Co., Ltd. Methanol was obtained from Tianjin Kemiou Chemical Reagent Co., Ltd. Potassium sodium tartrate was got from Shanghai Yuanye Biotechnology Co., Ltd. All other reagents were used without further purification.

Preparation of Bulk GCN

Bulk GCN was prepared through a one-step thermal condensation of melamine. The heating temperature was increased to 550 °C from room temperature with a ramp rate of 5 °C min^{-1} , and then kept at 550 °C for 4 hours.²⁸ After cooling down to the room temperature, bulk GCN (denoted as GCN) was obtained. Note that the yield of GCN was about 58.1%.

Preparation of GCN@a-TiO₂ Core@Shell Photocatalyst

The GCN@a-TiO₂ core@shell photocatalyst was prepared through a bioinspired mineralization method under mild conditions.²⁴ First, the as-prepared GCN was added into 2 mg mL^{-1} protamine solution and shaken for 10 min. After centrifugation and water washing, the protamine-absorbed GCN was dispersed into Ti-BALDH solution (1.25 wt%) and shaken for 10 min. After centrifugation and water washing, GCN coated with one layer of a-TiO₂ was obtained, which was denoted as GCN@a-TiO₂-1.

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4 Notably, the lower mineralization-inducer concentration (0.5 mg mL^{-1}) was also used
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6 to prepare the GCN@a-TiO₂ core@shell photocatalyst, as well, the process of
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8 bioinspired mineralization with standard mineralization-inducer concentration (2 mg
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10 mL^{-1}) was repeated one to two more times to control the amount of a-TiO₂ on GCN.
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12 The obtained samples were denoted as GCN@a-TiO₂-1*, GCN@a-TiO₂-2 and
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14 GCN@a-TiO₂-3, respectively.
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20 CdS@a-TiO₂, ZnO@a-TiO₂ and SrTiO₃@a-TiO₂ were also prepared to validate the
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22 universality of our bioinspired mineralization method. In brief, CdS was firstly
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24 prepared by directly mixing the same volume of 10 mM CdSO_4 and $10 \text{ mM Na}_2\text{S}$.
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26 After centrifugation, water washing and lyophilization, CdS was obtained.
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28 CdS@a-TiO₂, ZnO@a-TiO₂ and SrTiO₃@a-TiO₂ were prepared with the same
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30 procedure to GCN@a-TiO₂, just substituting GCN with CdS, ZnO and SrTiO₃.
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35 **Photocatalytic NADH Regeneration**

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37 The photocatalytic NADH regeneration was performed in a quartz reactor (light
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39 path: 1 cm) with 2 ml phosphate buffer saline (PBS) buffer (100 mM , $\text{pH } 8.0$)
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41 containing TEOA (15 w/v\%), $[\text{Cp}^*\text{Rh}(\text{bpy})\text{H}_2\text{O}]^{2+}$ (denoted as M, 0.25 mM), NAD^+
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43 (1 mM) and photocatalyst (1 mg mL^{-1}) at room temperature under 405 nm LED lamp
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45 illumination (300 mW cm^{-2}). The reaction system was firstly incubated in darkness
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47 for 10 min . During illumination, the regenerated NADH was detected by a UV-vis
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49 spectrophotometer (U-3010, Hitachi). The concentration of NADH was calculated
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51 according the absorbance at 340 nm with an extinction coefficient of $6220 \text{ M}^{-1} \text{ cm}^{-1}$.²⁵
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56 Moreover, the NADH regeneration experiments were also performed under different
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4 concentration of TEOA (3-15 w/v%), different light intensity (50-300 mW cm⁻²) and
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6 different pH values (7.0-9.0).
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9 The initial reaction rate (r , mmol g⁻¹ min⁻¹) of NADH regeneration was calculated
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11 using equation (1):
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$$13 \quad r = \frac{n_t}{m \times t} \quad (1)$$

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16 where n_t was the amount of regenerated NADH (mmol), m was the mass of
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18 photocatalyst (g), and t was the reaction time (min).
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23 The apparent quantum yield (%) of NADH regeneration was calculated using
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25 equation (2):
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$$27 \quad \text{Apparent quantum yield (\%)} = \frac{\mu\text{mol of electron consumed by NADH regeneration}}{\mu\text{mol of incident photons}} \times 100 \quad (2)$$

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30 Notably, both the initial reaction rate and apparent quantum yield were calculated
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32 according to the initial 2-4 min reaction. Since the regeneration of one mole of NADH
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34 would consume two moles of electrons, the amount of the electrons consumed by
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36 NADH regeneration was twice of the amount of regenerated NADH. The amount of
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38 the incident photons were calculated according to the light intensity on the front
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40 surface of quartz reactor (300 mW cm⁻²) and wavelength of LED lamp (405 nm).
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48 **Photobiocatalytic Methanol Production**

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50 The reaction solution composed of PBS buffer (4 mL, 100 mM, pH 8.0), NAD⁺ (10
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52 mM), M (0.25 mM), TEOA (15 w/v%), photocatalyst (1 mg mL⁻¹), YADH (0.2 mg
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54 mL⁻¹) and formaldehyde (40 mM) was incubated in darkness for 10 min. Then, the
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56 reaction solution was illuminated by 405 nm LED lamp at a light intensity of 300 mW
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4 cm⁻². The concentration of methanol was determined by Agilent 7820B gas
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6 chromatography.
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8 9 **Characterizations**

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11 Transmission electron microscopy (TEM) images were collected on a
12 field-emission transmission electron microscopy (Tecnai G2 F20). Elemental analysis
13 was conducted by electron energy loss spectroscopy (EELS) elemental mappings
14 attached to transmission electron microscopy (TEM) and energy dispersive
15 spectroscopy (EDS) attached to scanning electron microscopy (SEM). Sample
16 preparation for EELS mappings was similar to the TEM. The samples were dropped
17 on the microgrid and examined after drying. X-ray photoelectron spectroscopy (XPS)
18 was performed in a Perkin-Elmer PHI 1600 ESCA system with a monochromatic Mg
19 K α source. X-ray diffractometer (XRD) was measured on a Rigaku D/max 2500 V/PC
20 instrument with the graphite filtered Cu K α radiation ($\lambda = 1.54056 \text{ \AA}$), and the data
21 was acquired in the range of 10-60 $^\circ$ (2 θ) at a rate of 5 $^\circ \text{ min}^{-1}$. Ultraviolet-visible
22 diffuse reflectance spectra were performed for the dry-pressed disk samples with an
23 UV-vis spectrophotometer (U-3010, Hitachi) using BaSO₄ as the reflectance standard.
24 Fourier transform infrared spectroscopy (FTIR) spectra were collected on a
25 Nicolet-560 spectrometer. Atomic force microscopy (AFM) measurement of the
26 a-TiO₂ coating on silicon wafer was performed under tapping mode in a closed fluid
27 cell filled with deionized water using BRUKER Dimension Icon. Photoluminescence
28 (PL) spectra was recorded on Jobin Yvon Fluorolog 3-21 fluorescence spectrometer
29 with excitation at 350 nm. Time-resolved transient PL decay curves were obtained on
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Jobin Yvon Fluorolog 3-21 fluorescence spectrometer under the excitation of 390 nm and probed at 460 nm. Transient absorption (TA) spectroscopy was performed on Edinburgh LP980-KS transient absorption spectrometer equip with a Nd:YAG flash pump laser (an excitation source of 355 nm laser with ~ 7 ns pulse width) and an CCD camera array detector.

Results and Discussion

Preparation and Characterizations of GCN@a-TiO₂ Core@Shell Photocatalyst

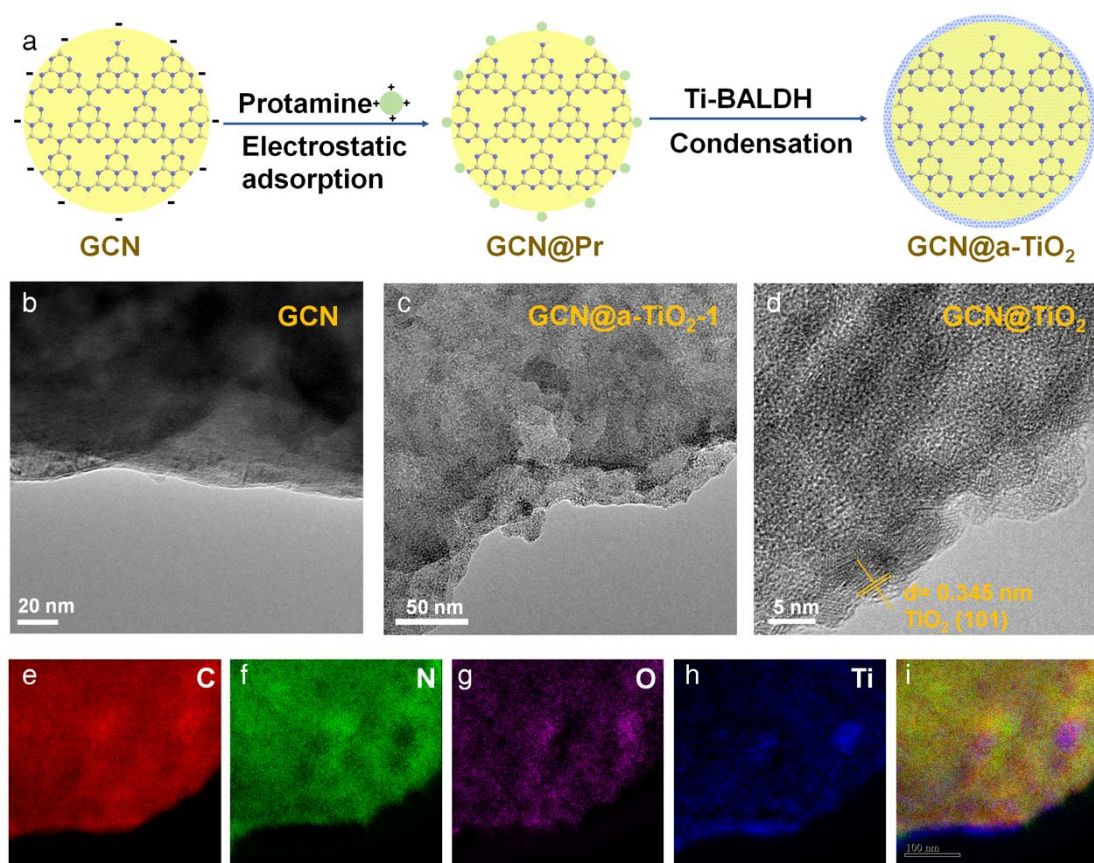


Figure 1. (a) Schematic preparation of GCN@a-TiO₂ core@shell photocatalyst. (b-d) High-resolution TEM images of (b) GCN, (c) GCN@a-TiO₂-1 and (d) GCN@TiO₂-1 calcined under 480 °C in air for 2 h. (e-i) Electron energy loss spectroscopy (EELS) elemental mappings of GCN@a-TiO₂-1. Red for carbon; green for nitrogen; purple for oxygen; blue for titanium. (i) The merged image of (e-h).

GCN@a-TiO₂ core@shell photocatalyst was prepared through protamine-induced mineralization of amorphous titania (a-TiO₂) on graphitic carbon nitrate (GCN) under room temperature in aqueous solution (**Figure 1a**). Specifically, positive charged protamine molecules were first adsorbed on the negatively charged surface of GCN through electrostatic interaction.²⁹ The adsorbed protamine then catalyzed the hydrolysis and condensation of titanium (IV) bis (ammonium lactato) dihydroxide (Ti-BALDH) and *in situ* deposition of a-TiO₂ nanoshell on GCN (GCN@a-TiO₂-1).²⁴ To validate the deposition of the a-TiO₂ nanoshell, the topological and chemical structures of GCN@a-TiO₂-1 were examined by high-resolution transmission electron microscopy (HR-TEM) and electron energy loss spectroscopy (EELS). Compared with GCN (**Figure 1b**), GCN@a-TiO₂-1 exhibited similar topological structure (**Figure 1c**), indicating amorphous nature of the mineralized titania. Subsequently, GCN@a-TiO₂-1 was calcined under 480 °C in air to examine the crystalline structure of deposited titania. As shown in **Figure 1d**, the crystal lattice corresponding to the (101) plane of anatase became rather apparent after calcination, verifying the successful deposition of amorphous titania on GCN. Moreover, EELS mappings revealed the well distribution of all elements, including O, Ti, C and N (**Figure 1e-i**). Amongst, C and N elements should be originated from GCN, while O and Ti elements should be assigned to a-TiO₂. This indicated a-TiO₂ was deposited on the entire surface of GCN.

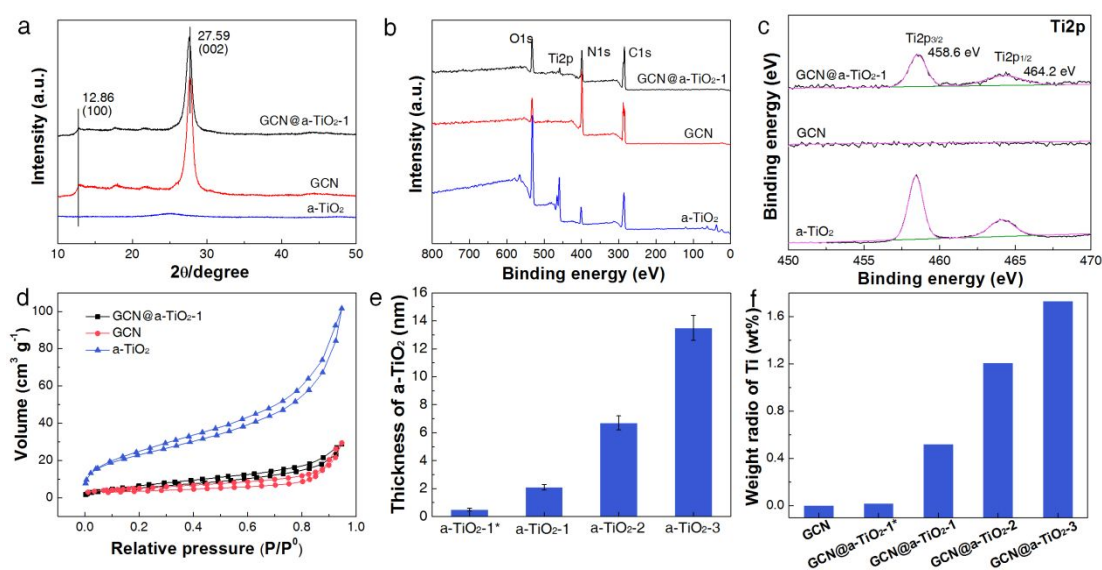
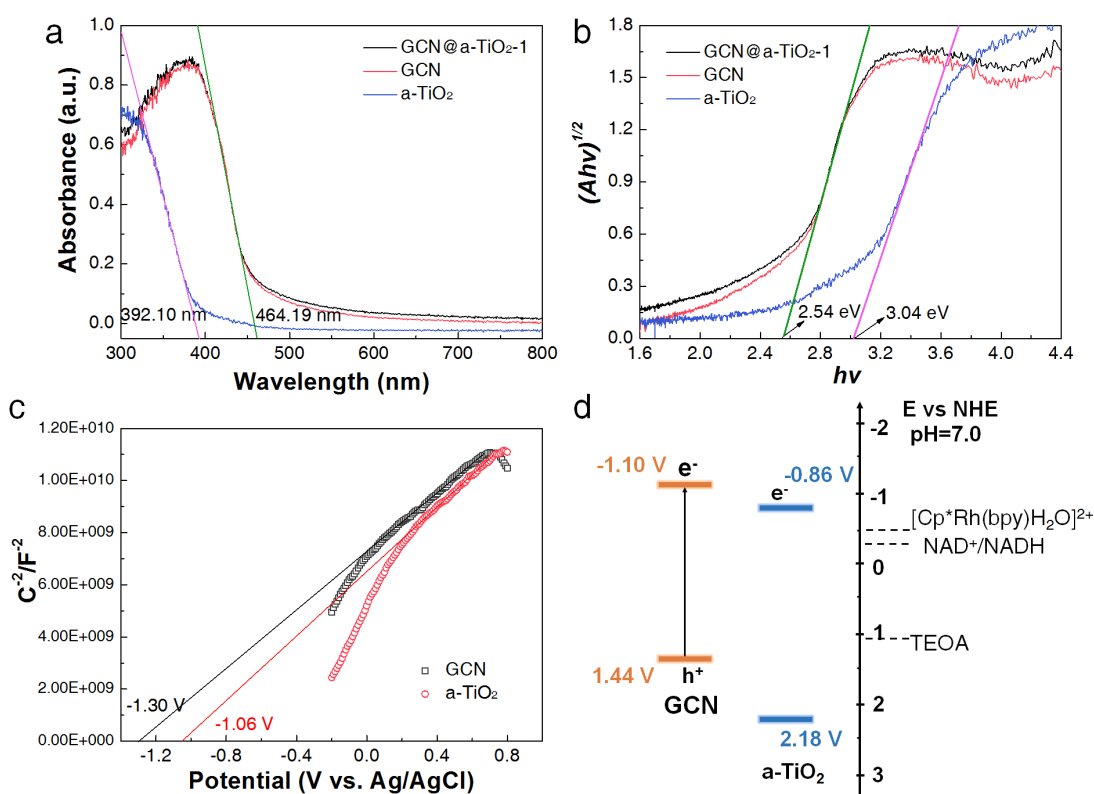


Figure 2. (a) XRD patterns and (b) XPS spectra of GCN@a-TiO₂-1, GCN and a-TiO₂. (c) High-resolution Ti 2p XPS spectra of GCN@a-TiO₂-1, GCN and a-TiO₂. (d) N₂ adsorption-desorption isotherms of GCN@a-TiO₂-1, GCN and a-TiO₂. (e) The thickness changes of a-TiO₂ coating on silicon wafer measured by AFM. (f) The weight ratio of Ti element of GCN, GCN@a-TiO₂-1* (0.5 mg mL⁻¹ protamine), GCN@a-TiO₂-1, GCN@a-TiO₂-2, GCN@a-TiO₂-3 and a-TiO₂ calculated from the EDS results.

The existence and content of a-TiO₂ on GCN were further determined by X-ray powder diffraction (XRD), Fourier transform infrared spectroscopy (FTIR) and X-ray photoelectron spectroscopy (XPS). As shown in **Figure 2a**, GCN@a-TiO₂-1 exhibited similar peaks with GCN at 12.86° (100) and 27.59° (002), which corresponded to the in-planar packing of tri-*s*-triazine and the interlayer stacking of the conjugated aromatic system.³⁰ The interlayer distance was calculated to be 0.323 nm, indicating the negligible influence of the mineralization process on the GCN

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4 structure, in line with the FTIR results (**Figure S1**). None of other diffraction peak
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6 representing titania was found, again evidencing the amorphous nature of a-TiO₂
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8 nanoshell. The elemental composition of GCN@a-TiO₂-1, GCN and a-TiO₂ was then
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10 characterized by XPS. As shown in **Figure 2b**, GCN@a-TiO₂-1 exhibited typical
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12 peaks of C, N, O and Ti elements. In deconvoluted Ti2p spectra, two characteristic
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14 peaks corresponding to Ti2p_{3/2} and Ti2p_{1/2} at 458.6 and 464.2 eV were observed
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19 (**Figure 2c**).²⁴ High-resolution C1s XPS spectra (**Figure S2a**) could be deconvoluted
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21 into three peaks at 284.81, 286.23 and 288.19 eV. Two main peaks at 284.81 and
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23 288.19 eV were assigned to C-C bond and N-C=N bond of GCN, respectively. The
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25 peak of 286.23 eV corresponded to C-O bond of protamine and GCN calcined in air.¹⁵
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29 For N1s spectra, the peaks were assigned to C-N-H at ~400.92 eV, C-(N)₃ at ~399.81
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31 eV and C-N=C at ~398.53 eV, respectively (**Figure S2b**).³¹ The texture structure
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33 examined by N₂ adsorption-desorption isotherms (**Figure 2d** and **Table S1**) showed
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35 that GCN@a-TiO₂-1 (20.9 m² g⁻¹) possessed higher surface area than GCN (13.9 m²
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37 g⁻¹), which was probably due to the deposition of a-TiO₂ nanoshell (a-TiO₂, 84.2 m²
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39 g⁻¹). Similar surface area, pore volume and pore size distribution of GCN@a-TiO₂-1,
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41 GCN@a-TiO₂-2 and GCN@a-TiO₂-3 suggested that the increase of mineralization
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43 cycles may only thicken the a-TiO₂ nanoshell rather than interfering its internal
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45 structure (**Table S1** and **Figure S3**). Since the nanoshell structure of a core@shell
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47 photocatalyst would have great impact on both behaviors of electron transfer and
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49 molecule diffusion during the photocatalytic reaction, the thickness of the a-TiO₂
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51 nanoshell was regulated by changing the concentration of protamine and the
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4 mineralization cycles. To reveal the thickness changes of a-TiO₂ nanoshell prepared
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6 under different conditions, silicon wafer was used instead of GCN as substrate for
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8 a-TiO₂ deposition, and atomic force microscopy (AFM) was employed to measure the
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10 thickness of a-TiO₂ coating. As shown in **Figure S4a**, the a-TiO₂-1 coating on silicon
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12 wafer was generally continuous and compact, which showed the thickness of 2.1±0.2
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14 nm obtained by measuring the height change of silicon wafer with and without a-TiO₂
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16 (**Figure S4b**). Using the same method, the thickness of a-TiO₂ deposited under
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18 different conditions was acquired (**Figure 2e**), which increased gradually with the
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20 increase of protamine concentration and mineralization cycle, in line with the
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22 previous literature.²⁴ The weight ratio of Ti on the surface region of GCN@a-TiO₂ got
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24 increased from 0.02% to 1.73% with the increase of protamine concentration and
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26 mineralization cycle as indicated by the EDS results (**Figure 2f**).



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4 **Figure 3. (a)** UV-Vis diffuse reflectance spectra and **(b)** the corresponding tauc plots
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6 of GCN@a-TiO₂-1, GCN and a-TiO₂. **(c)** Mott-Schottky plot for GCN and a-TiO₂
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8 electrodes measured in 0.1 mol L⁻¹ Na₂SO₄ solution at 1 kHz in the dark. **(d)**
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10 Schematic band structure of GCN and a-TiO₂, and the redox potential of
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12 [Cp*Rh(bpy)H₂O]²⁺, NAD⁺ and TEOA.
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20 The band structure of GCN and a-TiO₂ in GCN@a-TiO₂ was then characterized
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22 based on several techniques, including UV-Vis diffuse reflection spectroscopy (DRS),
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24 XPS valence band (VB) spectra and Mott-Schottky plot. As shown in **Figure 3a**,
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26 GCN@a-TiO₂-1 and GCN exhibited the same absorption edge of 464.19 nm,
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28 suggesting that the deposition of a-TiO₂ did not have impact on the light absorption of
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30 GCN. GCN and GCN@a-TiO₂ exhibited similar bandgap of ~2.54 eV from the tauc
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32 plots (**Figure 3b**), while a-TiO₂ showed a higher bandgap of 3.04 eV, which allowed
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34 visible light transmission through the a-TiO₂ nanoshell and did not affect the visible
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36 light absorption of GCN.³² Mott-Schottky analysis (**Figure 3c**) further indicated the
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38 flat-band potential of GCN and a-TiO₂ being located at -1.10 and -0.86 V (*vs* NHE,
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40 pH=7.0), which could be viewed as the CB values of GCN and a-TiO₂, respectively.³³
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42 Combined with the analysis of bandgap derived from the DRS spectra, the valence
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44 bands (VB) potentials of GCN and a-TiO₂ were calculated to be 1.44 and 2.18 V (*vs*
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46 NHE, pH=7.0), respectively (**Figure 3d**). The CB offset of 0.24 V between GCN and
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48 a-TiO₂ may facilitate the transfer of electrons from GCN to a-TiO₂.
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58 All above results indicated the formation of porous, conductive a-TiO₂ nanoshell
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with controllable thickness on the surface of GCN. Besides, such a-TiO₂ nanoshell could be further deposited on other semiconductor cores, such as CdS, ZnO and SrTiO₃, (Figure S6 and S7) by other organic inducers, such as PAH, lysozyme and arginine (Figure S6). Therefore, the semiconductor@a-TiO₂ core@shell photocatalyst prepared through bioinspired mineralization could set a good sample to judge the coordinated optimization between photo-generated electron transfer and electron donor molecule diffusion.

Coordination between Photo-Generated Electron Transfer and Electron Donor

Molecule Diffusion through a-TiO₂ Nanoshell

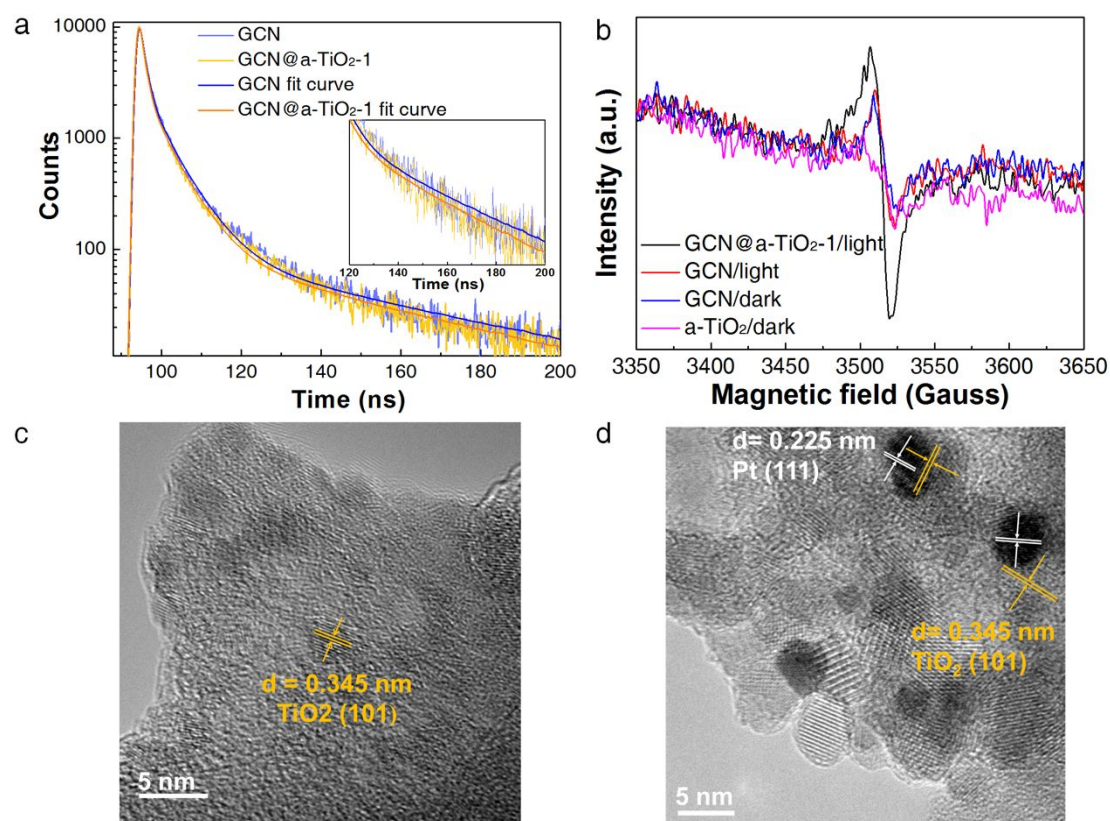


Figure 4. (a) Time-resolved transient PL spectra of GCN@a-TiO₂-1 and GCN. (b) EPR spectra of GCN@a-TiO₂, GCN and a-TiO₂ under light and dark condition. (c-d) HRTEM images of GCN@a-TiO₂ treated by H₂PtCl₆ under (c) dark and (d) light

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4 conditions followed by heating under 480 °C in air to crystallize the titania.
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9 Since it was the first time to adopt bioinspired a-TiO₂ nanoshell to coordinate the
10 photo-generated electron transfer and the electron donor molecule diffusion, the
11 electron transfer behavior of the a-TiO₂ nanoshell on the GCN core should be
12 in-depth elucidated. As shown in the steady-state PL (**Figure S8**), all samples of
13 GCN@a-TiO₂ exhibited weakened luminescence intensities by contrasting with GCN.
14 This indicated a-TiO₂ in the nanoshell could extract electrons from GCN.^{34,35} Then,
15 time-resolved transient PL was conducted to show the electron transfer behavior
16 through the a-TiO₂ nanoshell. As shown in **Figure 4a**, GCN@a-TiO₂-1 exhibited a
17 shorter life time than GCN (30.7 vs 33.8 ns), further evidencing the electron transfer
18 from GCN to a-TiO₂ (**Table S2**).³⁶ Decreased steady-state photoluminescence
19 intensity and intensity-average lifetime (τ_A) of the emission decay further suggested
20 that electron-hole recombination was suppressed by the rapid electron transfer from
21 GCN to a-TiO₂.³⁷ Since the trap states at the GCN surface would also cause shorter
22 PL lifetimes, EPR spectra of GCN@a-TiO₂-1, GCN and a-TiO₂ under light and dark
23 condition were further performed to validate the electron transfer process. As shown
24 in **Figure 4b**, GCN exhibited a weak Lorentzian line, which corresponded to the
25 unpaired electrons on the polymeric heptazine rings of GCN.³⁸ After illuminated by a
26 300 W xenon lamp (≥ 420 nm, 100 mW cm⁻²), minor changes were observed for GCN,
27 while a stronger EPR signal was observed for GCN@a-TiO₂. This should be arisen
28 from the formation of Ti³⁺ due to the electron transfer from GCN to a-TiO₂. It was
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4 also reported Pt^{4+} could be reduced into metallic Pt^0 and further deposited on the
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6 position bearing enriched photo-generated electrons for a photocatalyst.³⁹ The
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8 position could reflect the electron transfer direction during photocatlytic reaction. In
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10 this context, photochemical reduction of Pt^{4+} was conducted by $\text{GCN}@a\text{-TiO}_2$ under
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12 visible light (AM 1.5G). To identify the deposition position of metallic Pt^0 , the
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14 as-synthesized $\text{Pt}@GCN@a\text{-TiO}_2$ was calcined in air to crystallize the amorphous
15
16 titania. For comparison, $\text{GCN}@a\text{-TiO}_2$ was also treated by Pt^{4+} under dark condtion,
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18 followed by water washing and calcination in air (**Figure 4c**). By contrast with
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20 **Figure 4c**, **Figure 4d** showed the typical crystal lattice of Pt (111) on the top of the
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22 crystal lattice of titania (101), validating the electron enrichment on a-TiO₂ and the
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24 oriented electron transfer from GCN to a-TiO₂.
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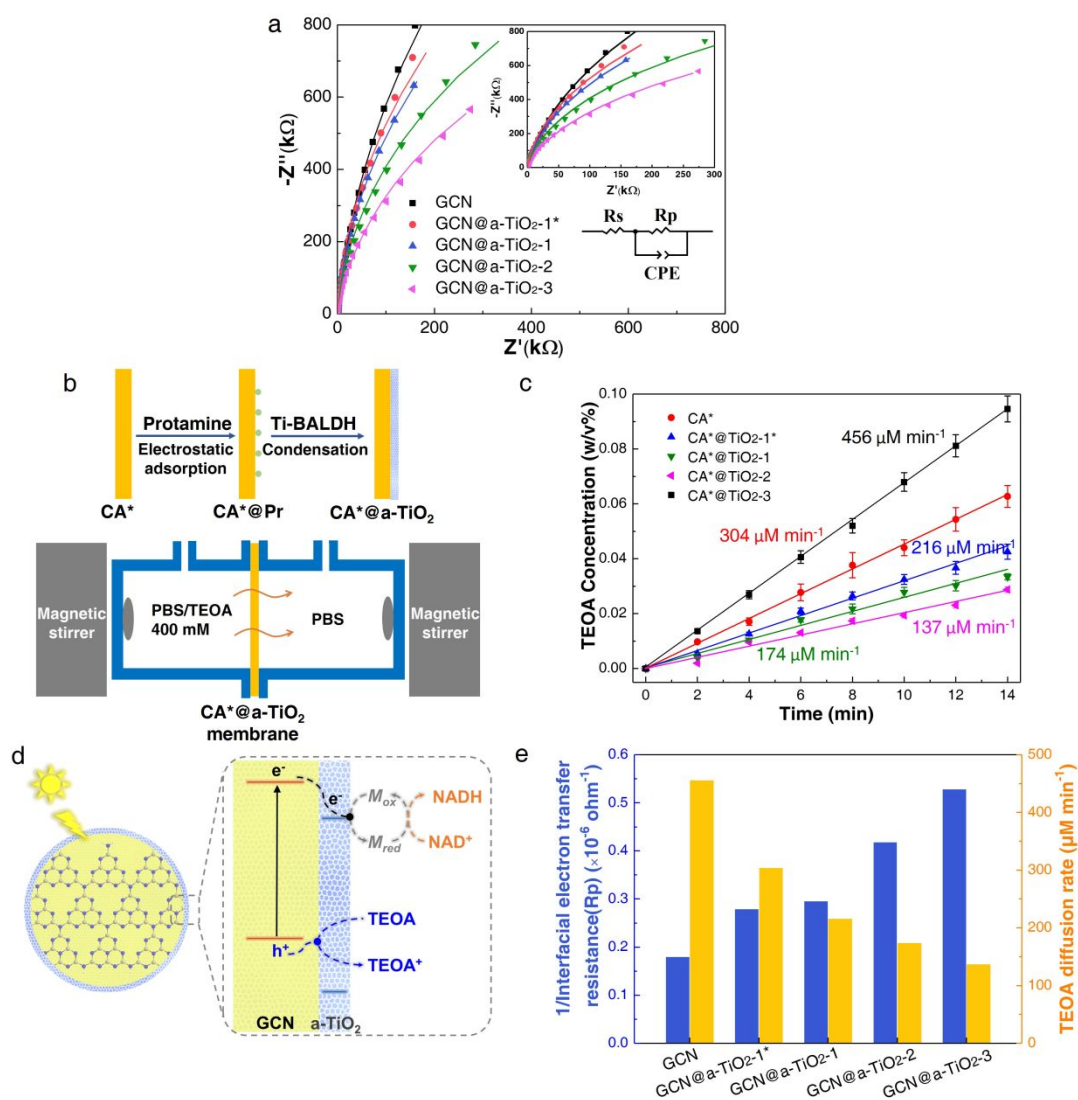


Figure 5. (a) EIS Nyquist plots. (b) Schematic preparation of CA*@a-TiO₂ membranes and a home-made diffusion device. (c) Diffusion rate of TEOA through CA*@a-TiO₂ membranes. Note that CA referred to cellulose acetate membrane, which was pre-treated with dopamine (2 mg mL⁻¹) tris-HCl solution for 2 h to enhance the surface hydrophilicity and narrow the pore size. The pre-treated CA membrane was denoted as CA* membrane. (d) Schematic interfacial electron transfer and TEOA diffusion through the nanoshell of GCN@a-TiO₂ photocatalyst. (e) Interfacial electron transfer resistance of GCN, GCN@a-TiO₂-1*, GCN@a-TiO₂-1,

GCN@a-TiO₂-2 and GCN@a-TiO₂-3, as well as TEOA diffusion rate of CA*, CA*@a-TiO₂-1*, CA*@a-TiO₂-1, CA*@a-TiO₂-2 and CA*@a-TiO₂-3.

The coordination between electron transfer and molecule diffusion through the a-TiO₂ nanoshell was then elucidated by altering the nanoshell thickness. The interfacial electron transfer resistance was evaluated by EIS Nyquist plots. As shown in **Figure 5a**, with the increase of a-TiO₂ nanoshell thickness, the radius of the curve decreased, suggesting the lower interfacial electron transfer resistance for thicker a-TiO₂ nanoshell. The interfacial electron transfer resistance could be calculated through the equivalent circuit (**Figure 5a**), where R_p corresponded to the interfacial electron transfer resistance.²⁸ According to the fitting results, R_p of GCN, GCN@a-TiO₂-1*, GCN@a-TiO₂-1, GCN@a-TiO₂-2 and GCN@a-TiO₂-3 were calculated to be 5.55×10⁶, 3.58×10⁶, 3.38×10⁶, 2.39×10⁶ and 1.89×10⁶ ohm, respectively. GCN@a-TiO₂-3 exhibited the smallest semi-diameter and the lowest R_p value, in line with the results from steady-state PL spectra (**Figure S8**). This further demonstrated that thicker a-TiO₂ nanoshell could decrease the interfacial electron transfer resistance and promote charge separation. The more efficient electron transfer for thicker a-TiO₂ nanoshell should be arisen from the larger interfacial area between GCN and a-TiO₂, and the inhibited charge recombination by separating electrons and holes over a larger distance.

In addition to extracting electrons from GCN, the a-TiO₂ nanoshell should also allow the free diffusion of TEOA to consume the photo-induced holes on the surface

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4 of GCN. As shown in **Figure S3**, the pore diameter of a-TiO₂ nanoshell was ~3.3 nm,
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6 which was ~5 times larger than TEOA (0.59×0.61×0.63 nm³). Hence, TEOA could
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8 diffuse through a-TiO₂ nanoshell to the surface of GCN. To prove the diffusion of
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10 TEOA, we designed a home-made device with a-TiO₂ nanolayer coated on pre-treated
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12 cellulose acetate (denoted as CA*) membrane as the support (**Figure 5b**). For CA*,
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14 the TEOA diffusion rate was 456 μM min⁻¹ (**Figure 5c**). After coating one layer of
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16 a-TiO₂ under 0.5 mg mL⁻¹ protamine (to mimic GCN@a-TiO₂-1*), the TEOA
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18 diffusion rate decreased to 304 μM min⁻¹. This suggested that a-TiO₂ nanoshell
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20 exerted diffusion resistance to TEOA but still allow its penetration. With the increase
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22 of a-TiO₂ thickness on CA* membrane to mimic GCN@a-TiO₂-1, GCN@a-TiO₂-2
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24 and GCN@a-TiO₂-3, the TEOA diffusion rate further decreased to 216, 174 and 137
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26 μM min⁻¹. This suggested that although thicker a-TiO₂ nanoshell reduced the electron
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28 transfer resistance, it also increased the molecule diffusion resistance (**Figure 5d and**
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30 **5e**). The performance of GCN@a-TiO₂ core@shell photocatalyst should be governed
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32 by both behaviors of electron transfer and molecule (TEOA) diffusion. Thus, an
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34 optimal nanoshell thickness exists toward the highest photocatalytic efficiency.
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45 **Photocatalytic Nicotinamide Regeneration.**

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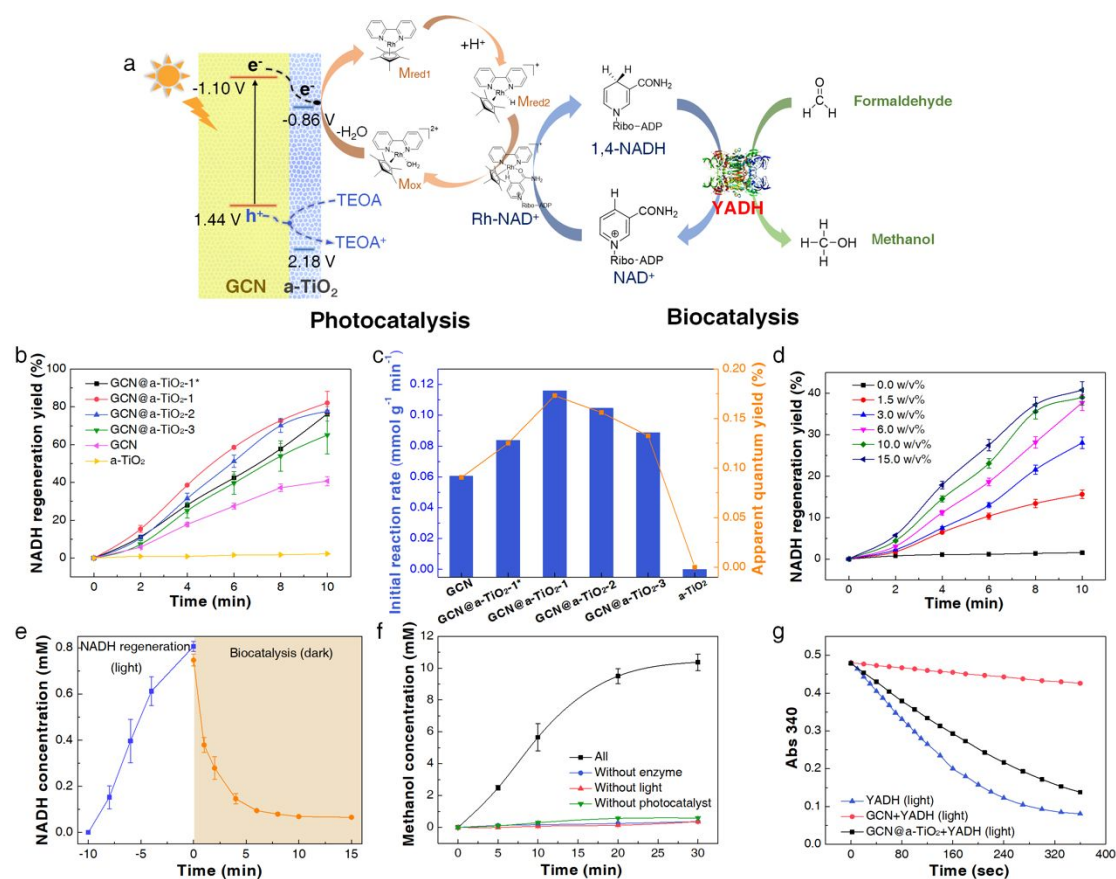


Figure 6. (a) Schematic photocatalytic NADH regeneration and photo-bio-coupled catalytic conversion of formaldehyde into methanol. (b) Photocatalytic NADH regeneration by GCN@a-TiO₂, GCN and a-TiO₂, and (c) the corresponding initial reaction rate and apparent quantum yield (2-4 min). (d) Photocatalytic NADH regeneration by GCN with different TEOA concentration. (e) NADH regeneration and enzymatic degradation enabled by GCN@a-TiO₂-1 and YADH. (f) Photo-bio-coupled catalytic production of methanol. (g) The activity of YADH after incubated with GCN and GCN@a-TiO₂ for 1 h under LED light illumination ($\lambda = 405$ nm).

In the photo-bio-coupled catalytic system, the expensive nicotinamide cofactor NADH activates oxidoreductase and provided hydrides for biocatalytic reaction.

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4 Therefore, efficient and selective regeneration of NADH is an important process for
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6 assessing photo-bio-coupled catalytic system.⁴⁰⁻⁴¹ Based on the above investigation,
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8 the NADH regeneration performance was evaluated by utilizing $[\text{Cp}^*\text{Rh}(\text{bpy})\text{H}_2\text{O}]^{2+}$
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10 as an electron mediator (M) and TEOA as a hole scavenger (also called electron
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12 donor).^{25,42} Upon the absorption of visible light by GCN, electrons would be excited
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14 to the CB of GCN. The electrons then transferred to the CB of a-TiO₂ nanoshell. The
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16 holes left on the GCN surface were consumed by TEOA that diffused through the
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18 a-TiO₂ nanoshell. Finally, M accepted the electrons, accomplishing the selective
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20 reduction of NAD⁺ (**Figure 6a**). Notably, the regenerated NADH was detected by
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22 measuring its characteristic absorption at 340 nm and confirmed by enzymatic
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24 conversion (**Figure S9**). And, we could confirm that the oxidized product of TEOA
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26 have little impact on the reaction system (**Figure S10**).

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35 As shown in **Figure 6b** and **6c**, all samples of GCN@a-TiO₂ exhibited higher
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37 NADH regeneration efficiency than GCN, while the NADH regeneration efficiency
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39 exerted a volcano-shape curve as the a-TiO₂ nanoshell thickness increased. The
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41 phenomenon of nanoshell thickness-dominated NADH regeneration efficiency should
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43 be arisen from the competing between electron transfer from GCN to a-TiO₂ and
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45 diffusion of TEOA through a-TiO₂ nanoshell to consume holes. It should be noted
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47 that the influence of electron transfer efficiency from the photocatalyst to M was
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49 excluded through using an excessive amount of M (**Figure S11**). As demonstrated in
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51 previous part, thicker a-TiO₂ nanoshell exhibited higher electron transfer rate but
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53 larger TEOA diffusion resistance. For GCN, GCN@a-TiO₂-1* and GCN@a-TiO₂-1,
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4 the a-TiO₂ nanoshell thickness was increased, the initial NADH regeneration rate and
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6 apparent quantum yield increased from 0.06 to 0.12 mmol g⁻¹ min⁻¹ and 0.09% to
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8 0.17%, respectively, due to the dominant function of facilitated electron transfer
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10 arisen from the a-TiO₂. Further increasing the nanoshell thickness, the initial NADH
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12 regeneration rate and apparent quantum yield decreased from 0.12 to 0.09 mmol g⁻¹
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14 min⁻¹ and 0.17% to 0.13%, which was probably owing to larger TEOA diffusion
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16 resistance. The increase of TEOA diffusion resistance would inhibit the diffusion of
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18 TEOA through the nanoshell and further reduced the TEOA concentration near the
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20 surface of GCN. The lower TEOA concentration would decrease the consumption rate
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22 of the photo-generated holes and then increase the possibility of electron-hole
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24 recombination, thus lowering the NADH regeneration rate. The assumption was
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26 further validated by examining the effect of TEOA concentration on the NADH
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28 regeneration rate of GCN, where the NADH regeneration rate got increased with the
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30 increase of TEOA concentration on the surface of GCN (**Figure 6d**). As a result, after
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32 coordination between the electron transfer and TEOA diffusion through regulating the
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34 a-TiO₂ nanoshell thickness, GCN@a-TiO₂-1 exhibited the highest NADH
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36 regeneration yield of ~82.1% after reaction equilibrium, which also exerted the
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38 highest initial reaction rate (~0.12 mmol g⁻¹ min⁻¹) and apparent quantum efficiency
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40 (~0.17%) (**Figure 6c**). Importantly, the optimized NADH regeneration yield was over
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42 200% higher than GCN. Meanwhile, GCN@a-TiO₂-1 photocatalyst also exerted
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44 excellent operational stability and retained its original structure after the
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46 photocatalytic NADH regeneration (**Figure S12**). As a matter of fact, during the
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4 photocatalytic NADH regeneration, the photo-generated charge carriers (electrons and
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6 holes) were probably involved in the oxidation of water and TEOA by holes and the
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8 reduction of oxygen by electrons to form reactive oxygen species (ROS), which may
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10 destroy the structure of NADH. Detailed discussion about the generated ROS ($\bullet\text{OH}$,
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12 $\bullet\text{O}_2^-$ and other radicals), and their influence on the structure of NADH were presented
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14 in the supporting information (**Page S3 and Figure S13**).

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19 To evaluate the enzymatic activity of regenerated NADH, yeast alcohol
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21 dehydrogenase (YADH) was coupled with GCN@a-TiO₂ for photo-bio-coupled
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23 catalytic hydrogenation of formaldehyde to methanol.^{15,43} First, photocatalytic NADH
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25 regeneration and enzymatic hydrogenation were conducted successively. As shown in
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27 **Figure 6e**, NADH was successfully accumulated during the visible light illumination.
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29 After adding YADH and formaldehyde in the regeneration solution, NADH was
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31 completely oxidized in 10 min (**Figure S15**), indicating enzymatically active feature
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33 of the regenerated NADH. Moreover, the methanol product was measured under
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35 continuous visible light illumination. As shown in **Figure 6f**, the concentration of
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37 methanol got gradually increased and ~10.4 mM methanol was detected after 30 min
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39 illumination. By contrast, no methanol was detected in the absence of light, YADH or
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41 photocatalyst, verifying the reaction scheme in **Figure 6a**.

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50 The compatibility of photocatalyst and enzyme is crucial when the photocatalytic
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52 NADH regeneration is coupled with the enzymatic hydrogenation of formaldehyde by
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54 YADH. During the light illumination, GCN and YADH were incubated for 1 h, and
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56 YADH was almost completely inactivated (**Figure 6g**). We found that the
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4 photo-generated holes of GCN damaged the active site and structure of YADH. The
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6 active site of YADH is formed by the complexing of Zn ions with two cysteine
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8 residues (Cys 43 and 153) and one histidine residue (His 66). Cysteine residues are
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10 very susceptible to be oxidized by the photo-generated holes, leading to the loss of
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12 active site (Zn ions) and the inactivation of YADH. After deposition of the titania
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14 nanoshell on the surface of the GCN core, YADH retained over 67% of its original
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16 activity (**Figure 6g**). The core@shell structure could then avoid the unfavorable
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18 contact between photo-generated holes of GCN and YADH, and then improve the
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20 compatibility of photocatalyst and YADH under light illumination. In short, the
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22 core@shell structure could not only coordinate the process of electron transfer and
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24 molecule diffusion, but also isolate the photo-generated holes and YADH, improving
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26 the compatibility between these two kinds of catalysts.
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34 35 **Conclusions**

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38 In summary, GCN@a-TiO₂ core@shell photocatalyst was prepared by coating an
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40 a-TiO₂ nanoshell on the GCN core through a facile and green bioinspired
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42 mineralization method. The GCN core could be excited to generate electrons and
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44 holes under visible light illumination, whereas the porous, conductive a-TiO₂
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46 nanoshell could rapidly extract the photo-generated electrons from GCN and allow
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48 the free diffusion of electron donor molecules to consume the photo-generated holes
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50 left on the GCN surface. The coordination between photo-generated electron transfer
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52 and electron donor molecule diffusion was realized by regulating the thickness of the
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54 a-TiO₂ nanoshell. With the optimal nanoshell thickness, GCN@a-TiO₂-1 exerted the
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4 highest photocatalytic NADH regeneration yield of ~82.1%, over two folds higher
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6 than GCN, which could be coupled with enzymatic reaction for stably and sustainable
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8 production of methanol. Hopefully, this study could provide a generic strategy for the
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10 design of high performance photocatalysts through the coordinated optimization of
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12 the transfer/diffusion of multiple substances/species.
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16 **Associated content**

17 **Supporting Information**

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22 FTIR spectra of GCN@a-TiO₂-1, GCN and a-TiO₂; High-resolution XPS spectra of
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24 C1s and N1s; The pore size distribution curves of GCN@a-TiO₂-1, GCN@a-TiO₂-2,
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26 GCN@a-TiO₂-3, GCN and a-TiO₂; AFM images and corresponding height profiles of
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28 a-TiO₂-3, a-TiO₂-2, a-TiO₂-1 and a-TiO₂-1*; XPS valence band spectra of GCN,
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30 GCN@a-TiO₂-1*, GCN@a-TiO₂-1, GCN@a-TiO₂-2, GCN@a-TiO₂-3 and a-TiO₂;
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32 PL spectra at the excitation wavelength of 350 nm of GCN@a-TiO₂-1,
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34 GCN@PAH/Ti-1, GCN@Lys/Ti-1, GCN@Arg/Ti-1 and GCN; Energy Dispersive
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36 Spectroscopy (EDS) analysis of CdS@a-TiO₂-1; The initial reaction rate (2 min) and
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38 balance yield of NADH regeneration with GCN@a-TiO₂-1, GCN@PAH/Ti-1,
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40 GCN@Lys/Ti-1, GCN@Arg/Ti-1, GCN, CdS@a-TiO₂-1 and CdS; EDS spectra of
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42 ZnO@a-TiO₂ and SrTiO₃@a-TiO₂; Photocatalytic NADH regeneration with ZnO and
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44 ZnO@a-TiO₂, and SrTiO₃ and SrTiO₃@a-TiO₂; Steady-state PL spectra of
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46 GCN@a-TiO₂-1, GCN@a-TiO₂-2, GCN@a-TiO₂-3 and GCN; UV-vis absorption of
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48 0.05 w/v% TEOA; UV-vis absorption of NADH regeneration solution before and
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50 after light illumination; Photocatalytic NADH regeneration and enzymatic
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4 degradation of the regenerated NADH; Photocatalytic NADH regeneration by
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6 GCN@a-TiO₂ under light-dark conditions; Transient absorption decay kinetics
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8 spectra of GCN@a-TiO₂ and [M] in PBS/TEOA (100 mM/15 w/v%, pH=8.0) buffer
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10 with different ratio of M and GCN@a-TiO₂; Correlation between observed TAS
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12 decay kinetic constant (k_{obs}) and ratio of M and GCN@a-TiO₂; Electron transfer
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14 efficiency and NADH regeneration yield with different ratio of M and GCN@a-TiO₂;
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16 Recycling stability of GCN@a-TiO₂ for NADH regeneration; XRD spectra of
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18 GCN@a-TiO₂ before and after eight times recycling; TEM image of GCN@a-TiO₂
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20 before and after eight times recycling; Comparison of the conduction band (CB) and
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22 valence band (VB) potential of GCN with the redox potential of •OH/H₂O and •O₂⁻/O₂;
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24 ¹H NMR spectra of NAD⁺ regeneration solution and freshly prepared NAD⁺ solution;
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26 NADH regeneration enabled by GCN@a-TiO₂-1 under different protamine
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28 concentration; NADH regeneration enabled by GCN@a-TiO₂-1 under different pH
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30 values; NADH regeneration enabled by GCN@a-TiO₂-1 under different light
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32 intensity; UV-Vis diffuse reflectance spectra of NAD⁺ and regenerated NADH;
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34 Chromatogram of methanol; The surface area and pore volume of GCN@a-TiO₂-1,
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36 GCN@a-TiO₂-2, GCN@a-TiO₂-3, GCN and a-TiO₂; The multi-exponential fitting
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38 results of time resolved photoluminescence.
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Notes

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